

1 Does an inter-flaw length control the accuracy of rupture forecasting
2 in geological materials?

3 Jérémie Vasseur¹, Fabian B. Wadsworth¹, Michael J. Heap²,
4 Ian G. Main³, Yan Lavallée⁴, Donald B. Dingwell¹

5 ¹Dept. of Earth and Environmental Sciences, Ludwig Maximilian University, Munich, Germany.

6 ²Institut de Physique de Globe de Strasbourg (UMR 7516 CNRS), Strasbourg, France.

7 ³School of Geosciences, University of Edinburgh, Edinburgh, UK.

8 ⁴Dept. of Earth, Ocean and Ecological Sciences, University of Liverpool, Liverpool, UK.

9 **Multi-scale failure of porous materials is an important phenomenon in nature and in material**
10 **physics – from controlled laboratory tests to rockbursts, landslides, volcanic eruptions and**
11 **earthquakes. A key unsolved research question is how to accurately forecast the time of system-**
12 **sized catastrophic failure, based on observations of precursory events such as acoustic emissions**
13 **(AE) in laboratory samples, or, on a larger scale, small earthquakes. Until now, the length scale**
14 **associated with precursory events has not been well quantified, resulting in forecasting tools that**
15 **are often unreliable. Here we test the hypothesis that the accuracy of the forecast failure time**
16 **depends on the inter-flaw distance in the starting material. We use new experimental datasets for**
17 **the deformation of porous materials to infer the critical crack length at failure from a static**
18 **damage mechanics model. The style of acceleration of AE rate prior to failure, and the accuracy**
19 **of forecast failure time, both depend on whether the cracks can span the inter-flaw length or not.**
20 **A smooth inverse power-law acceleration of AE rate to failure, and an accurate forecast, occurs**
21 **when the cracks are sufficiently long to bridge pore spaces. When this is not the case, the predicted**
22 **failure time is much less accurate and failure is preceded by an exponential AE rate trend. Finally,**
23 **we provide a quantitative and pragmatic correction for the systematic error in the forecast failure**
24 **time, valid for structurally isotropic porous materials, which could be tested against larger-scale**
25 **natural failure events, with suitable scaling for the relevant inter-flaw distances.**

26 **Keywords:** porous materials; inter-pore length; acoustic emission; precursors; rock failure; damage
27 mechanics.

28

29 **1. Introduction**

30 All materials contain flaws with a large range of length scales, from kilometre-sized fractures in the
31 crust (Hatton et al., 1994), to meter-sized cavities (Castro et al., 2002) and fractures in rocks and
32 synthetic materials (Allègre et al., 1982), down to micro- and nano-pores and density fluctuations in
33 thin-film glasses (Guyer and Dauskardt, 2004) and crystals. These flawed materials eventually rupture
34 in catastrophic failure events when applied stresses become sufficiently large to produce system-
35 spanning fractures (Sammis and Ashby, 1986). Recent efforts have converged and found that two
36 observations dominate the physics of failure of these systems. First, the flaws in the system concentrate
37 stress relative to the unflawed domains of the material and therefore the flaw fraction in the material
38 exerts a first-order control on the far-field stress required for macroscopic failure (Kemeny and Cook,
39 1986; Sammis and Ashby, 1986; Vasseur et al., 2013). Second, the size of flaws and the inter-flaw length
40 determine the extent to which the cracks that emanate from flaws will interfere (Bažant, 2004; Sornette
41 and Andersen, 1998). These two paradigms underpin all elastic models of rupture events in
42 heterogeneous solids and predict that, as the material approaches macroscopic failure, the rate of energy
43 released as acoustic emissions (AEs) by microscopic failure events accelerates (Kilburn, 2012; Lockner
44 et al., 1991; Scholz, 1968; Turcotte and Newman, 2003; Vasseur et al., 2015; Voight, 1989). When first
45 proposed, the finding that these bulk-material accelerations in the rate of energy release or event number
46 approaches a singularity that coincided with the failure time provided a tantalizing possibility that
47 material failure could be forecast accurately using indirect observations such as micro-earthquakes or
48 AEs prior to wholesale rupture (Voight, 1989). Indeed a large effort has been expended in assessing the
49 utility of this tool for forecasting hazardous failure phenomena in nature (Bell et al., 2011; Bell and
50 Kilburn, 2013; Hao et al., 2016; Kilburn et al., 2017; Robertson and Kilburn, 2016; Voight, 1988).
51 However, the still-limited success of these methods (Bell et al., 2013) has highlighted complexities in

52 the approach to failure of heterogeneous materials that must be addressed if forecasting tools are going
53 to be of the widest utility.

54

55 **2. Micromechanical models for the uniaxial deformation of porous materials**

56 Here we present a linear elastic model to demonstrate quantitatively how stress is distributed around a
57 circular (2D) or spherical (3D) cavity in an infinite solid and exposed to a far-field stress. Then we
58 follow previous work to scale that concept to a porous body with finite dimensions in order to predict
59 the failure stress of a porous material as a function of the porosity ϕ and the pore radius a . We focus on
60 the uniaxial case in which far-field stresses are applied in one direction only, and later we discuss how
61 our findings could be extended to more complex stress configurations in principle. Finally, we explore
62 other characteristic length scales in natural materials that may be more relevant than the pore size;
63 namely, the inter-pore and inter-particle distances.

64

65 **2.1. The concentration of uniaxial applied stress around circular and spherical pores**

66 First we use a linear elastic model for the stress distribution around a circular (2D) or spherical (3D)
67 cavity. For the 2D case we opt for the solution credited to Kirsch (1898) and to Goodier (1933) for the
68 3D case, repeated in variable completeness in subsequent work (Jaeger et al., 2009; Soutas-Little, 1999)
69 with which the stress components can be computed for each spatial position around a cavity of radius a
70 and which, for completeness, we reproduce here. We use the Cartesian coordinate system with the far-
71 field stress applied in the z -direction and the centre of the pore positioned at $(x, y, z) = 0$. A line of
72 length r away from the pore centre in any direction subtends an angle with the z -axis of θ and an angle
73 with the x - or y - axes of ψ . In what follows, we normalize each axis (x, y, z) and the radial direction r
74 by a and the individual stress components τ_{ij} by the far-field applied stress σ_1 , yielding a coordinate
75 system and stress tensor components for which a bar above the parameter denotes its normalized value.

76 We introduce the 2D and 3D stress components in the supplementary file as Eqs (S1)-(S3) and Eqs (S4)-
77 (S7).

78 In Figure 1, we present the normalized stress as a colour map around a 2D circular cavity (Fig 1a) and
79 a 3D spherical cavity using $\nu = 0.25$ (Fig 1b), which is a first-order approximation for crustal rocks
80 (assuming the two Lamé parameters are equal). The lobes of concentrated stress are compressive in the
81 region of the solid surrounding the z -axis and are tensile in the region of the solid about the x -axis (2D)
82 or the x - y plane (3D). It is in these lobes of concentrated stress that fractures would be most likely to
83 initiate. For this reason, in Figure 1c we additionally show the stress resolved along the z -axis ($\theta =$
84 $\pi/2$) and along the x -axis (2D) or the x - y plane (3D) ($\theta = 0$).

85

86 **2.2. Approximate methods for predicting the stress required for rupture**

87 The deformation of elastic porous media results in cracks that propagate from interfaces at which stress
88 is locally concentrated relative to the far-field applied load (Sammis and Ashby, 1986). Sammis and
89 Ashby (1986) present a static so-called *pore-crack* model to compute the degree to which stress is
90 concentrated around cavities (a cavity stress intensity factor K_{Ii}) and the degree to which cracks that
91 grow from those cavities interact (a crack interaction stress intensity factor K_{Iii}). Their solutions are cast
92 as simple functions of the sample porosity ϕ , rendering them easy to use and to compare with measured
93 data (Zhu et al., 2011). Where the *pore-crack* model is used, only the solution for 2D is usually compared
94 with experimental data (Baud et al., 2014; Zhu et al., 2011). Here we apply the *pore-crack* model
95 (Sammis and Ashby, 1986) in uniaxial conditions where the sum of K_{Ii} and K_{Iii} is the total stress
96 intensity K_I .

97 When a far-field stress σ_1 is applied ($\sigma_2 = \sigma_3 = 0$) onto a material rupture begins only when the local
98 stress σ exceeds σ_c . At this point a fracture can initiate to a distance c away from the pore or cavity at
99 which distance $\sigma = \sigma_c$, and beyond which $\sigma < \sigma_c$. This distance c is the equilibrium crack length for
100 the stress state at a given time and, defined in non-dimensional form as $\bar{c} = c/a$. Then \bar{c} as a function

101 of a normalized stress $\bar{\sigma} = \sigma\sqrt{\pi a}/K_{Ic}$ (where K_{Ic} is the fracture toughness or critical stress intensity
 102 required for crack propagation in the solid) for the 3D and uniaxial case, is as follows (Sammis and
 103 Ashby, 1986)

$$\bar{\sigma} = \left(\frac{0.62\sqrt{\bar{c}}}{(1 + \bar{c})^{4.1}} + \frac{\sqrt{2\phi(1 + \bar{c})}}{\pi} \right)^{-1} \quad (1)$$

104 where the first term on the right-hand side of Eq. (1) describes the growth of a crack from a single pore,
 105 while the second term is a crack-interaction term related to the porosity ϕ (see Sammis & Ashby (1986)
 106 for full description). This model neglects time-dependency and therefore it is implicitly assumed that
 107 the cracks grow more quickly than the far-field stress changes. This is similar to saying that the strain
 108 rate is sufficiently low that the damage is in equilibrium with the stress at all times.

109 Eq. (1) provides us with a tool to assess when linear elastic mechanics predicts failure for a porous
 110 material loaded uniaxially by assessing Eq. (1) when $d\bar{\sigma}/d\bar{c} = 0$. This condition clearly demarks the
 111 onset point beyond which increased crack growth will manifest as a stress drop. In practice, the sample
 112 can remain coherent for a relatively small region of crack lengths above this point, but, following Zhu
 113 et al. (2011), we approximate the failure point as described. At this point, we can define the equilibrium
 114 normalized crack length that is failure in this model as $\bar{c}_c = c_c/a$ where \bar{c}_c is a function of ϕ only.
 115 Differentiating Eq. (1) with respect to \bar{c} and setting $d\bar{\sigma}/d\bar{c} = 0$ then yields an expression for the
 116 porosity

$$\phi = 2\pi^2(1 + \bar{c}_c) \left(\frac{2.542\sqrt{\bar{c}_c}}{(1 + \bar{c}_c)^{5.1}} - \frac{0.31}{(1 + \bar{c}_c)^{4.1}\sqrt{\bar{c}_c}} \right)^2 \quad (2)$$

117 so that the equilibrium crack length at failure \bar{c}_c can be found numerically for a given ϕ .

118

119 **2.3. The inter-flaw length and pore sizes in porous materials**

120 In natural rocks as well as synthetic porous materials, the pore space is rarely an array of spherical
 121 cavities (e.g. Vasseur et al., 2013). Indeed, for sandstone, limestone, welded volcanic materials including
 122 ignimbrites, among many other lithologies, it is more relevant to think of the solid matrix as an array of
 123 near-spherical objects (grains) and the pore space as the convolute inter-sphere void (Vasseur et al.,
 124 2016). In this scenario, we can apply metrics for the characteristic length scales of the system based on
 125 theoretical models for the description of microstructure in random heterogeneous materials (Torquato,
 126 2013). This is an advance on using simple concepts of “pore sizes”, which are typically scaled to bulk
 127 porosity simply by assuming regular simple arrays of monodisperse pores in a unit volume (Zhu et al.,
 128 2011).

129 If we think of our model geological material as a packing of spherical grains with radius R and that these
 130 grains are able to freely overlap or inter-penetrate, then we can account for porosities lower than the
 131 maximum packing porosity of grains. In this case, we use a nearest-neighbour function to find the
 132 average inter-pore lengths in a heterogeneous grain pack. The nearest-neighbour function in a random
 133 system of interacting spheres can be evaluated from the probability $F(r)dr$ that an arbitrary sphere
 134 centre in the system lies at a distance between r and $r + dr$ from another sphere centre. The n th moment
 135 of $F(r)$ is given by (Torquato et al., 1990)

$$\langle \bar{r}^n \rangle = \int_0^\infty \bar{r}^n \bar{F}(\bar{r}) d\bar{r} \quad (3)$$

136 where a bar above a symbol denotes a parameter normalized by the sphere radius \mathcal{R} (i.e. $\bar{r} = r/\mathcal{R}$ and
 137 $\bar{F}(\bar{r}) = F(r)\mathcal{R}$). The first moment (i.e. $n = 1$) gives the mean nearest-neighbour distance between
 138 spheres $\bar{l} \equiv \langle \bar{r} \rangle$. In our case the spheres can either represent the pores (i.e. $\mathcal{R} = a$), yielding inter-pore
 139 distances termed l_1 , or the particles (i.e. $\mathcal{R} = R$), yielding inter-particle distances termed l_2 . In the case
 140 where the spheres are monodisperse and fully penetrable, the nearest-neighbour function for finding l_1
 141 or l_2 is (Torquato et al., 1990)

$$\bar{F}(\bar{r}) = 3\eta\bar{r}^2 \exp(-\eta\bar{r}^3) \quad (4)$$

142 where η represents the sphere reduced density (i.e. the product of sphere number density and
 143 sphere volume). Combining Eq. (4) with Eq. (3) and taking $n = 1$, results in an analytical expression

$$\bar{l}_i = \frac{\Gamma(4/3)}{\eta^{1/3}} \quad (5)$$

144 where Γ is the gamma function, and $\eta = -\ln(1 - \phi)$ when $i = 1$ (the case when $\bar{l}_1 = l_1/a$) and $\eta =$
 145 $-\ln \phi$ when $i = 2$ (the case when $\bar{l}_2 = l_2/R$). We can think of l_1 as a characteristic inter-pore distance
 146 which we will use to estimate the average distance a crack must bridge to connect two pores, and l_2 as
 147 a characteristic inter-particle distance which we can think of as a more rigorous proxy for pore size in
 148 heterogeneous random media. In Figure 2 we show how both \bar{l}_1 and \bar{l}_2 vary with ϕ for overlapping
 149 monodisperse spheres (spherical pores in the case of \bar{l}_1 and spherical particles in the case of \bar{l}_2).

150 For comparison, we can also use the model of Lu and Torquato (1992) to predict the characteristic pore
 151 radius between random heterogeneous overlapping particles. To do this, we use Eq. (4) to get a pore-
 152 size density function $P(a)$ (here $\eta = -\ln \phi$)

$$\bar{P}(\bar{a}) = \frac{\bar{F}(1 + \bar{a})}{\phi} = \frac{3\eta(1 + \bar{a})^2}{\phi} \exp(-\eta(1 + \bar{a})^3) \quad (6)$$

153 where $\bar{a} = a/R$ and $\bar{P}(\bar{a}) = P(a)R$. The n th moment of $P(a)$ is given by

$$\langle \bar{a}^n \rangle = \int_0^\infty \bar{a}^n \bar{P}(\bar{a}) d\bar{a} \quad (7)$$

154 and the first moment (i.e. $n = 1$) gives the mean pore radius $\langle \bar{a} \rangle$.

155

156 **3. Experimental materials and methods**

157 **3.1. Materials, experimental deformation and data acquisition**

158 We use experimental data from samples of a range of different porous geological media including
159 sandstone, limestone, volcanic welded debris, and synthetic analogues for quartz-rich sandstone of
160 sintered glass beads (c.f. Blair et al., 1993). While these data are associated with experiments from
161 published studies (Heap et al., 2013; Heap et al., 2015; Wadsworth et al., 2016), the acoustic data are
162 analysed here for the first time in terms of the critical crack length inferred from a micromechanical
163 model. Fig 3 shows photomicrographs of characteristic sample microstructure collected either using
164 scanning electron microscopy or optical microscopy. We selected this range of samples to encompass
165 the simplest case of a two-phase system of solid and pores (synthetic analogues for quartz-rich
166 sandstones; Fig 3a-c), and the more complex cases of multiphase natural materials relevant to crustal
167 rocks (quartz-rich sandstones, volcanic clastic rocks, and clastic limestones; Fig 3d-f).

168 The porosity of all materials was determined using helium pycnometry and the mean particle sizes $\langle R \rangle$
169 were estimated using optical microscopy. For the sandstone samples $\langle R \rangle \approx 2.5 \cdot 10^{-4}$ m (Wadsworth et
170 al., 2016), for the limestone samples $\langle R \rangle \approx 2.5 \cdot 10^{-4}$ m (Heap et al., 2013), for the welded volcanic debris
171 $\langle R \rangle \approx 2 \cdot 10^{-4}$ m (Heap et al., 2015), and for the synthetic sintered glass beads, $\langle R \rangle \approx 7.6 \cdot 10^{-5}$ m (Vasseur
172 et al., 2016, 2015). All samples were dried and deformed under uniaxial loading at a constant strain rate
173 of 10^{-5} s^{-1} . Acoustic emission data was collected continuously during deformation at acquisition rates of
174 20 MHz, synchronized with the mechanical data acquisition.

175 In uniaxial tests, the time at which the samples rupture completely, t_c , is simply the point at which the
176 measured stress drops significantly and is therefore trivial to pick. At a strain rate of 10^{-5} s^{-1} the peak
177 stress σ_c typically occurs at $t = t_c$ or just prior to t_c , consistent with the failure criterion $d\bar{\sigma}/d\bar{c} = 0$
178 assumed above in deriving Eq. (2).

179

180 **3.2. Retrospective ‘forecasting’ of the failure time**

181 Retrospective forecasting, or ‘hindcasting’ is a necessary though not sufficient step in assessing the
182 predictability of non-linear complex systems. It can provide a ‘best-case scenario’ for forecasting in

183 prospect near the failure time, but can also provide insight into phenomena not yet included in current
 184 models. Here we test a commonly-applied model for failure forecasting using precursory changes in
 185 the rate of acoustic emissions – high-frequency elastic wave packets generated by the rapid release of
 186 strain energy during local micro-crack rupture – during deformation. Specifically, we monitor the
 187 number of events per unit time $\dot{\Omega}$, the parameter most commonly used to forecast failure of a system due
 188 to its sensitivity to deformation (Lavallée et al., 2013, 2008; Vasseur et al., 2015; Voight, 1988). The
 189 variety of lithologies tested allows us to study failure forecasting in a controlled manner, and to isolate
 190 the fundamental controls on the evolution of $\dot{\Omega}$ and the accuracy of the forecast failure time.

191 One of the most common ways to relate the rate of an observable signal $\dot{\Omega}$ that is precursory to the
 192 forecast failure time t_p is the Time-Reversed Omori Law (TROL; Vasseur et al., 2015)

$$\dot{\Omega}(t) = k(t_p - t)^{-p} \quad (8)$$

193 where k is a scaling factor and p parameterizes the rate of acceleration of $\dot{\Omega}$. Here the approach of $\dot{\Omega}$ to
 194 failure is an inverse power-law, with a well-defined singularity at t_p , as expected for a system
 195 approaching a critical point defined by a system-sized event. Note that in the following we refer to t_c as
 196 the observed failure time. Following the procedure described in detail in Bell et al. (2013) we applied
 197 the TROL to catalogues of AE events in order to retrospectively forecast failure. This law has three free
 198 parameters (k , p and t_p) to adjust, which are not known *a priori*. The Maximum Likelihood (ML)
 199 method is applied to the TROL and has been shown to provide statistically stable and repeatable
 200 estimates of its parameters (Bell et al., 2013). Additionally, this method uses the timing of individual
 201 AE events rather than event rates determined in equally spaced bins (as is commonly the case when
 202 applying standard failure forecast methods). The ML solution is found by minimizing the negative log-
 203 likelihood function using a downhill simplex algorithm. In an interval (t_0, t_1) and for n number of
 204 observations, the log-likelihood function for the TROL is given by

$$\ln(L) = \sum_{i=1}^n \ln\left(k(t_p - t_i)^{-p}\right) + \frac{k}{1-p} \left((t_p - t_1)^{1-p} - (t_p - t_0)^{1-p} \right) \quad (9)$$

205 for $p \neq 1$ and

$$\ln(L) = \sum_{i=1}^n \ln \left(k(t_p - t_i)^{-1} \right) + k(\ln(t_p - t_1) - \ln(t_p - t_0)) \quad (10)$$

206 for $p = 1$. This yields a retrospective forecasted failure time t_p based on precursory signals only.

207 Alternatively, the approach of $\hat{\Omega}$ to failure may be exponential: $\hat{\Omega}(t) = h \exp(qt)$, where h is another
208 scaling parameter and q controls the evolution of $\hat{\Omega}$. The exponential model can be fit in the same way
209 using another form of the ML method but does not have the same degree of forecast power as there is
210 no unambiguous singularity in $\hat{\Omega}$ at any time. The ML solution for the exponential law is

$$\ln(L) = q \sum_{i=1}^n t_i + n \ln(h) - \frac{h}{q} (\exp(qt_1) - \exp(qt_0)) \quad (11)$$

211 The forecasting window was restricted to 90% of the known failure time t_c . In cases where the TROL
212 is an appropriate model for the underlying process, the analysis by Bell et al. (2011b) indicates a typical
213 random error (precision) of $\pm 6\%$ at 95% confidence or so when the forecast was made at 90% of t_c .
214 Hence any difference between forecast and observed t_c above $\pm 6\%$ or so is diagnostic of a systematic
215 error or bias (loss of accuracy) at this level of confidence, requiring a correction to the TROL.

216 The Bayesian Information Criterion (BIC) is a statistical tool to quantify the relative performance of
217 different models in describing a dataset (i.e. when making an inference, the preferred model is more
218 likely to have the lower BIC value). It is based on the likelihood L of the observation given the model,
219 with a weighting favouring the model with fewer parameters, and is given by $BIC = -2 \ln(L) +$
220 $N \ln(n)$ for which N is the number of free parameters. Therefore, calculating the positive difference
221 ΔBIC between the BIC value of the TROL and the exponential law respectively helps discriminate
222 which is the preferred model. As such, when the ΔBIC becomes negative it indicates a strong statistical
223 preference for the TROL over the exponential law.

224 4. Results and analysis

225 4.1. Comparing results with the mechanical model

226 Using the peak stress σ_c observed in the uniaxial compression experiments, we can test the
227 micromechanical model presented. Applying Eq. (1) allows us to compute the normalized uniaxial
228 stresses for every normalized crack length value for a given porosity (see Fig 4a *inset* for this result for
229 four porosities). We can compute the normalized critical crack length \bar{c}_c for a failure to occur in a sample
230 of given porosity using Eq. (2), and then convert that to a critical peak stress required for failure $\bar{\sigma}_c$
231 using Eq. (1). The model and observed peak stresses can then be compared directly as a hypothesis test.
232 As we know the mean particle radius for all of our experimental samples, we can compute a
233 characteristic pore radius using either Eq. (5) to find l_2 or Eq. (7) to find a . We can use this to find the
234 stress required for failure, termed the uniaxial compressive strength (UCS). In Fig 4 we show that when
235 we perform this analysis using a in the dimensional result for c_c and σ_c , the model performs poorly (Fig
236 4a). Whereas when we use δl_2 (with a calibrated $\delta = 3/2$) in the result, we find that the predicted peak
237 stress is in good agreement with the observed peak stress (Fig 4b). This validates the micromechanical
238 model used here (Sammis and Ashby, 1986), and confirms l_2 as the best metric for the characteristic
239 pore dimension. This is in contrast with previous work in which investigators use a characteristic pore
240 radius a in Eq. (2) (Zhu et al., 2011). The success of using l_2 (Fig 4b), demonstrates that the challenges
241 associated with defining and measuring l_2 in rocks can be circumvented and represents an advance on
242 previous approaches.

243

244 4.2. Predicting the rupture time

245 We show in Fig 5 that all samples exhibit apparent acceleration of $\dot{\Omega}$ toward the observed failure time
246 t_c . Here we normalize the time data so that deformation begins at -1, and t_c occurs at 0 (Fig 5). Across
247 the full range of porosities tested, these accelerations are well-fit by a power-law TROL (see Eq. (8)).
248 While we plot the cumulative number of events for the model and observed data in Fig 5, the model was
249 fitted on the rate data, so that the data points remain independent. Here we do not show explicitly the

250 best-fit p , which lie below 1 and compare favourably with previously published values for synthetic
251 tests (Bell et al., 2013) and deformation experiments (Cornelius and Scott, 1993; Voight, 1989). The
252 best-fit t_p diverges from t_c as $\phi \rightarrow 0$, indicating that the power-law extends systematically beyond t_c
253 toward its singularity at $t_p > t_c$. The time deficit between the forecast and observed failure time exceeds
254 the estimated precision of $\pm 6\%$ or so described above, and increases systematically as porosity
255 decreases: the systematic error is as high as 100% at a porosity of 3% (Fig 5).

256 The observation in Figure 5 is consistent with those of Vasseur et al. (2015; data from this study is
257 repeated here for context and comparison) and Jiang et al. (2016), whereby system-sized failure can
258 only forecast failure accurately (i.e. within the calculated precision of the ML method) in highly
259 heterogeneous, porous samples. Conversely it does not provide accurate forecasts of failure in relatively
260 homogeneous, low-porosity materials. Vasseur et al. (2015) also showed that failure of porous materials
261 is best described by an inverse power-law acceleration at high porosity, and by the exponential
262 acceleration at low porosity ($\phi < 0.3$; Fig 6 *inset*). However, the low-porosity trends are not necessarily
263 exponential in nature and this is an effect of the non-existence of a power-law singularity in these data.
264 Additionally the failure time is not defined by the dynamics underlying the exponential model and
265 failure forecasts using this model must be based on other metrics. These observations highlight a current
266 shortcoming in our ability to forecast system-sized material failure in natural and synthetic porous
267 media, which we now address.

268 The first clue to accounting for the systematic bias in the failure time is illustrated in Figure 7a. Here
269 we see a strong positive correlation between the bias, expressed as the ratio of the predicted to the
270 observed failure time, and the inter-pore distance $\bar{l}_1(\phi)$. This implies that failure is poorly resolved
271 when the distance between two pores is large and thus that the crack-length required to connect two
272 pores should also be large. If we apply the micromechanical model used to accurately predict the failure
273 stress, we would expect that $2c_c$ is the crack length required to connect two pores (given that a crack
274 grows from each pore at the same time; dashed line in Fig 7b). But as porosity decreases, there is a
275 systematic deviation from the micromechanical model result for $\bar{c}_c(\phi)$ from $\bar{l}_1(\phi)$, calculated using
276 Eq. (2) and (5) respectively. We find a correlative relation between the normalized failure forecast and

277 the normalized critical crack length, such that $t_p/t_c \approx 2\bar{c}_c/\delta$ (with $\bar{c}_c = c_c/l_2$) and hence t_p/t_c as a
278 function of \bar{l}_1 (solid line in Fig 7b). We infer that this represents a distance deficit between the crack
279 length and the length required to connect two pores, which is larger for low porosity samples than for
280 high porosity samples. We note a strong correlation between this increasing distance deficit (between
281 the dashed and solid lines in Fig 7b) and the increasing forecast bias with respect to decreasing porosity
282 previously illustrated in Figure 5. The implication is that low porosity materials have relatively large
283 distances that must be spanned by cracks in order to fail, and that this leads to late time, rapid time-
284 dependent crack growth rather than equilibrium crack growth predicted by the static model here
285 presented. This also seems to correlate with the shift from AE accelerations that are well-predicted by
286 power laws (and accurately forecast failure) to those that are better predicted by exponential
287 accelerations and which cannot accurately forecast failure.

288 Finally, we use the distance deficit of Figure 7b to correct for the bias in the forecast failure time, as
289 illustrated in Figure 7c. The agreement is very good within the remaining (random) scatter in the data
290 about the optimal line. This figure validates the modification to the TROL we have made using the
291 microstructural and micromechanical models presented, and the empirical results of Figure 7a and 7b.

292

293 **5. Discussion and conclusions**

294 Our work shows that as the nearest-neighbour distance approaches the pore size, i.e. $\bar{l}_1 \rightarrow 1$, the forecast
295 failure time becomes more accurate, and that this transition can be associated with the case where $2\bar{c}_c =$
296 \bar{l}_1 . More specifically, if the equilibrium crack length at failure approaches half the total distance between
297 pores which must be bridged to achieve failure, then the precursory AE rate indeed exhibits an inverse
298 power law approach to a critical singularity that coincides with the observation of catastrophic failure.
299 However, when the nearest-neighbour distance is much greater than the pore size $\bar{l}_1 \gg 1$, then the
300 forecast is not successful and there is a length-deficit between the equilibrium crack length \bar{c} and the
301 distance that must be bridged \bar{l}_1 . This implies that there is not a simple approach to a critical failure
302 point for systems in which there is a length-deficit: instead failure occurs suddenly and early. This is

303 consistent with the observation that the approach to failure is better described by an exponential than a
304 power-law evolution in the AE rate on a statistical basis for these samples (Fig 6 *inset*). Finally, this
305 finding suggests that it is the inter-pore length (the nearest-neighbour distance) that is more important
306 than the porosity of a medium in determining whether a crack can propagate the required half-distance
307 between two neighbouring pores to precipitate failure.

308 Although we focus on the uniaxial loading case for experimental convenience, the full triaxial
309 micromechanical model provided by Sammis and Ashby (1986) could be used to extend the results to
310 the compressional stress field relevant to the Earth's crust. There is no reason to anticipate a distance
311 deficit term would not act as a control on the early failure time in this case, though this may take a
312 different functional form to the results presented here, which itself may be dependent on confining
313 pressure. This remains to be tested in future work. We also note the materials here tested are structurally
314 isotropic, so there is no directional dependence of the inter-pore length scaling. This is not necessarily
315 true at all scales in natural and synthetic systems, so the effect of anisotropy remains to be examined.

316 There is large variability in the accuracy and reliability of forecast attempts using the classic failure
317 forecast method (De la Cruz-Reyna and Reyes-Dávila, 2001; Kilburn, 2003; Kilburn and Voight, 1998;
318 Ortiz et al., 2003; Smith et al., 2007; Smith and Kilburn, 2010; Voight and Cornelius, 1991) or the TROL
319 (Bell et al., 2013) or variations thereof (Boué et al., 2015; Salvage and Neuberg, 2016). This applies
320 even for fully retrospective forecasting of volcanic eruption time, based on precursory earthquakes at
321 different volcanoes worldwide. This demonstrates that there is a wide range of error in applying this
322 method to natural data, and that these tools are not always of the widest utility for real time monitoring.
323 While we have proposed a correction that works well in a controlled laboratory setting, it is unlikely
324 that pore-scale heterogeneity controls volcano- or fault-scale rupture. However, it is likely that there are
325 larger length scale domains of heterogeneity in those crustal systems, which control the flaw-to-flaw
326 fracture propagation events precursory to system-sized rupture. Our model therefore suggests that if
327 these larger scale flaws can be identified, then it is their inter-flaw distances that would most likely scale
328 the error in forecasts. Nevertheless, in qualitative terms, our model suggests that it is the most apparently
329 heterogeneous systems, with the lowest inter-flaw distances, that might be expected to be well forecast

330 using the variants of the failure forecasting method outlined above. One example could be the Mt St
331 Helens volcano (USA) in 1985-86, which had a systematic error in the forecast failure time of <0.1
332 expressed as a time since the start of the acceleration (Voight and Cornelius, 1991).

333 In the volcanic case, there is a crucial distinction to be drawn between events that can be interpreted to
334 be the result of magma fracturing during ascent (Kendrick et al., 2014; Neuberg et al., 2006) in an
335 established conduit and those that are likely related to the fracturing of crustal rocks during the initiation
336 of eruption and magma propagation to the surface (Kilburn et al., 2017; Lamb et al., 2017). An example
337 of the latter interpretation was made on the basis of the patterns of evolution of low-frequency events
338 preceding individual eruption episodes at Soufriere Hills volcano (Montserrat) (Neuberg et al., 2006),
339 and an example of the latter is the signal evolution without eruption at Campi Flegrei (Italy) (Kilburn et
340 al., 2017). Our experiments are explicitly suited to explain the brittle mechanics involved in the latter
341 process of crustal fracturing ahead of a vanguard magma batch on its way to the surface. But
342 additionally, our results are applicable to the highest viscosity systems in the former case of fracturing
343 of magma itself (Lavallée et al., 2008).

344 In summary this study provides a simple explanation for the substantial variability in the success of
345 forecast attempts for system-sized catastrophic failure in natural and artificial systems (Bell and Kilburn,
346 2013, 2012) and the quantitative correction we provide offers the opportunity to scale lab-forecasts to
347 natural systems, if a convincing scaling for lengths between large scale flaws can be identified.

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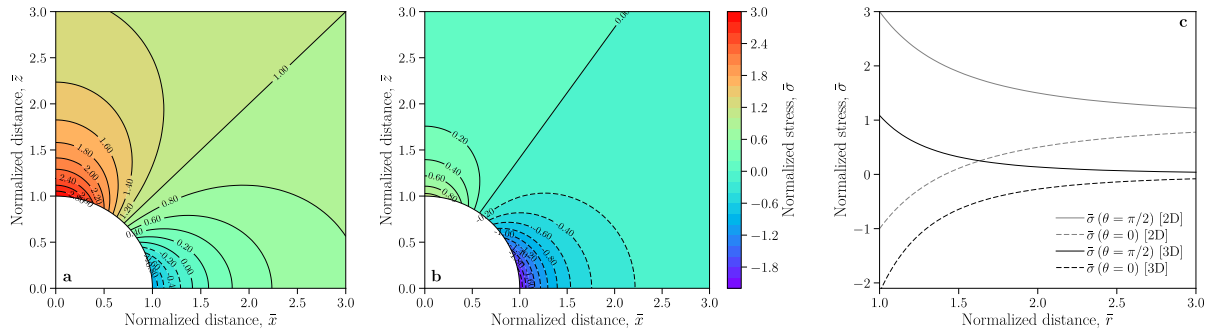
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482 **Figures**

483



484

485 Figure 1 – Stress around pores in 2D and 3D. The distances are normalized by the cavity radius a . **a.**

486 The total stress distribution around a circular pore in an infinite plate (2D) mapped out in the positive

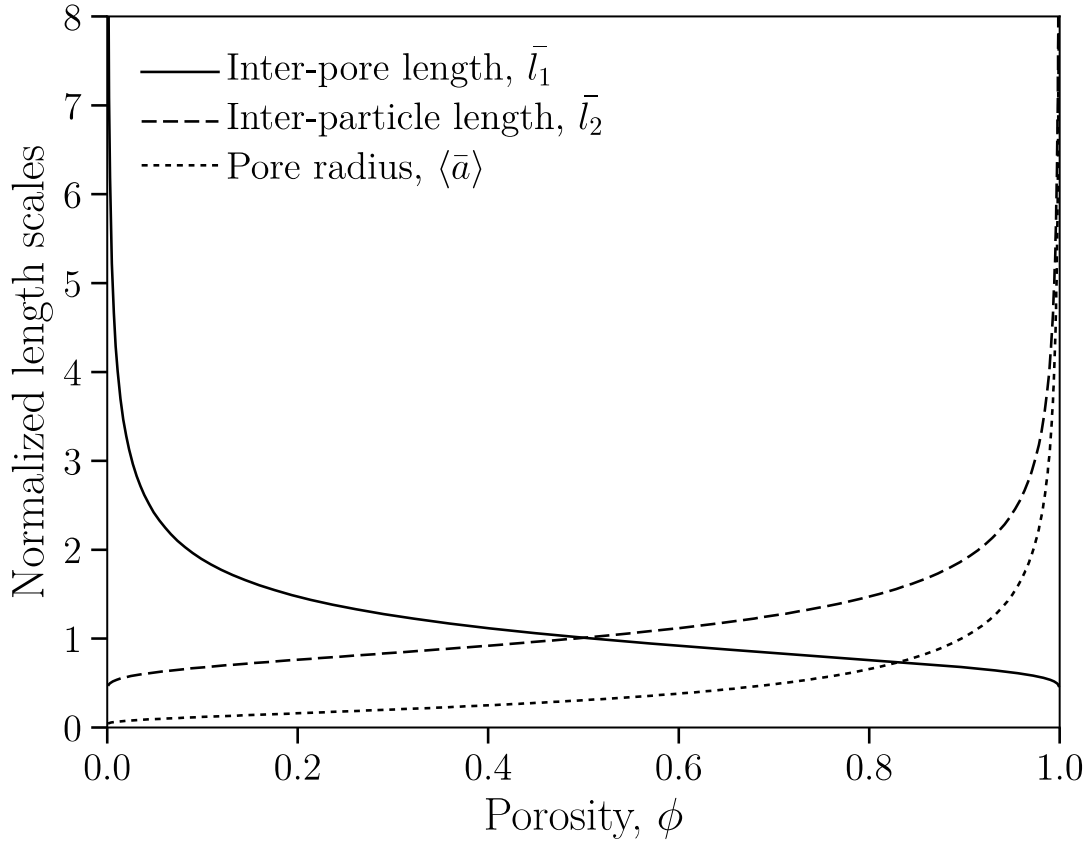
487 quadrant of the $x - z$ plane as calculated by combining Eqs (S1)-(S3). **b.** The total stress distribution

488 around a spherical pore in an infinite body (3D) mapped out in the positive quadrant of $x - z$ plane as

489 calculated by combining Eqs (S4)-(S7). **c.** The total stress resolved along the z -axis ($\theta = \pi/2$) and along

490 the x -axis (2D) or the x - y plane (3D) ($\theta = 0$).

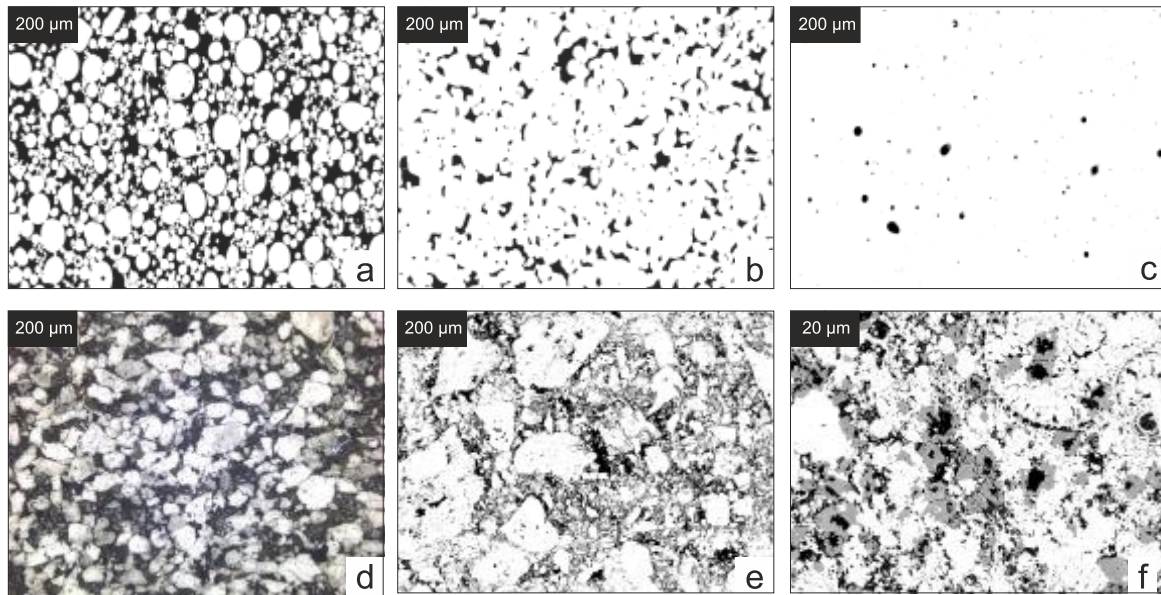
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493 Figure 2 – The calculated mean nearest-neighbour distance between overlapping spheres in a statistically
 494 random heterogeneous medium as a function of porosity using Eq. (5) and (7). l_1 corresponds to the
 495 case where the spheres are the pores (porosity is thus the sphere volume fraction) and is then an inter-
 496 pore distance. l_2 corresponds to the inverse case where the spheres are the particles (porosity is thus the
 497 volume fraction exterior to the spheres) and is then an inter-particle length. $\langle a \rangle$ corresponds to the mean
 498 pore radius between solid spheres.

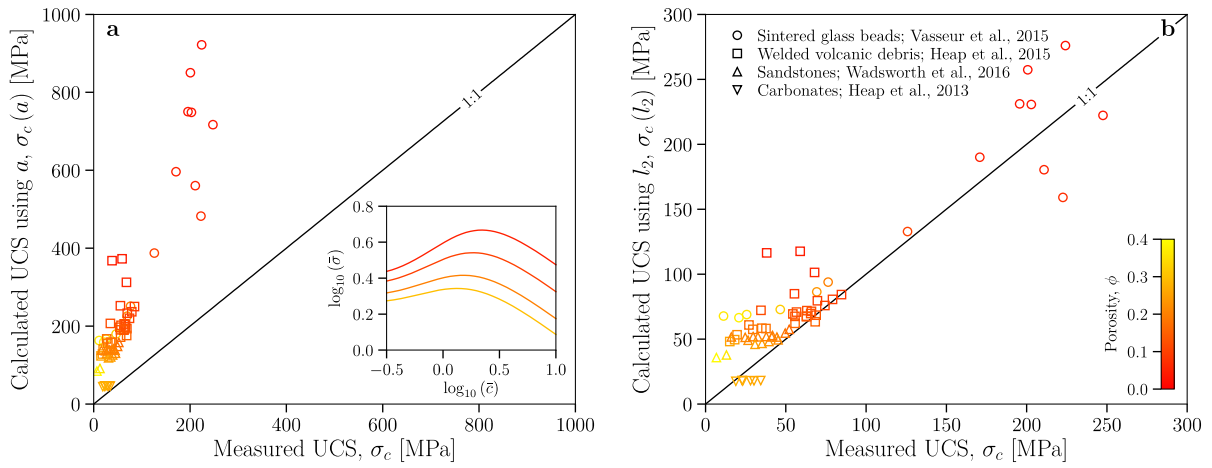
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501 Figure 3 – Characteristic photomicrographs of the samples used in this study: a suite of **a-c** synthetic
 502 porous glasses and **d-f** natural samples. Black represents the gas phase, white and shades of grey the
 503 solid phase. **a.-c.** Sintered glass beads from Vasseur et al. (2015) with varying porosity. **d.** Darley Dale
 504 (UK) sandstone from Wadsworth et al. (2016). **e.** Mt Meager (Canada) welded volcanic debris from
 505 Heap et al. (2015). **f.** Mt Climiti (Italy) carbonate from Heap et al. (2013). Note that all materials are
 506 porous, variably densified, initially granular materials with simple microstructures **a-c** or increasingly
 507 complex microstructures **d-f**. In particular, the limestone **f** is multiphase and finer grained than the other
 508 samples **a-e**.

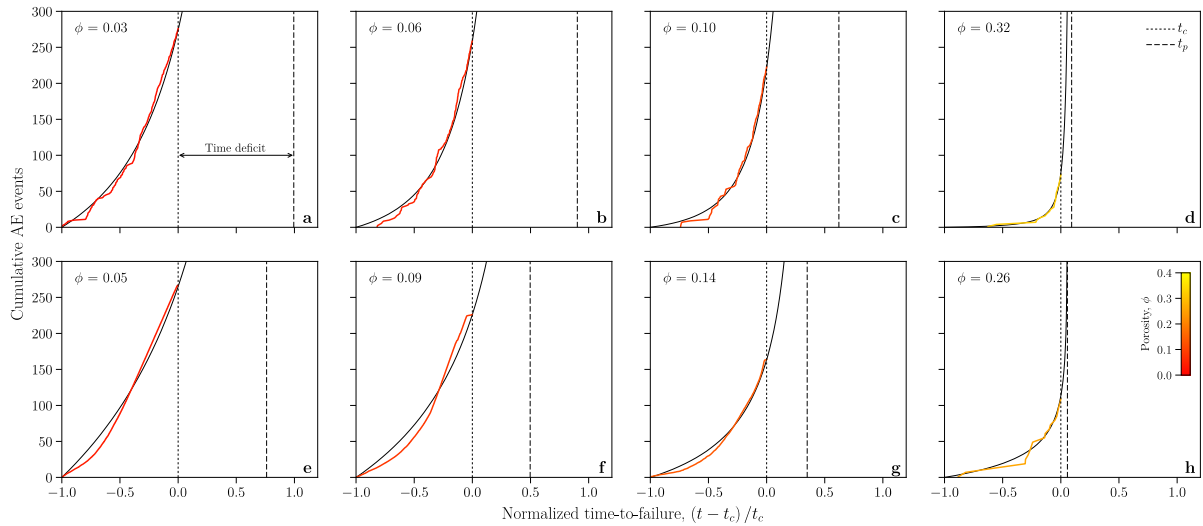
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511 Figure 4 – Calculated versus measured uniaxial compressive strength (UCS) using **a** the inferred mean
 512 pore radius $\langle a \rangle$ from Eq. (7) and **b** the inferred inter-particle distance l_2 from Eq. (5) for all the samples
 513 studied here and colour-coded for porosity. *Inset* – the evolution of stress σ with crack length c for 4
 514 different porosity values as calculated from Eq. (1) for a sample subjected to uniaxial loading.

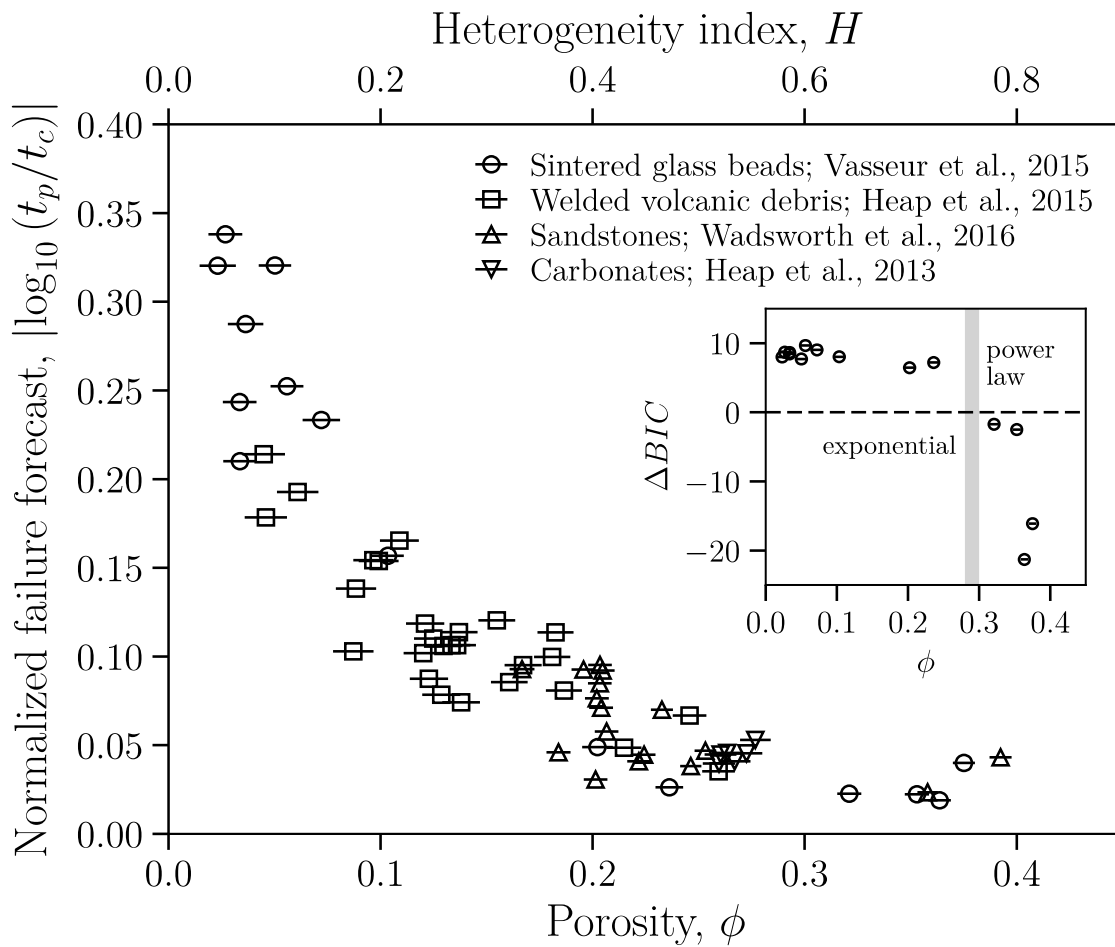
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517 Figure 5 – Examples of failure forecasting for two sample types studied herein (**a-d** sintered glass beads;
 518 Vasseur et al. (2015) and **e-h** welded volcanic debris; Heap et al. (2015)) with varying porosity. The
 519 colourful thick solid lines represent the raw data, while the black thin solid lines represent the model
 520 output. The predicted (from the model) t_p and the actual failure times t_c are marked by vertical dashed
 521 and dotted lines, respectively. One can notice how the time deficit between t_p and t_c reduces as porosity
 522 increases and how this corresponds well with a decrease in distance deficit as shown in Fig 7.

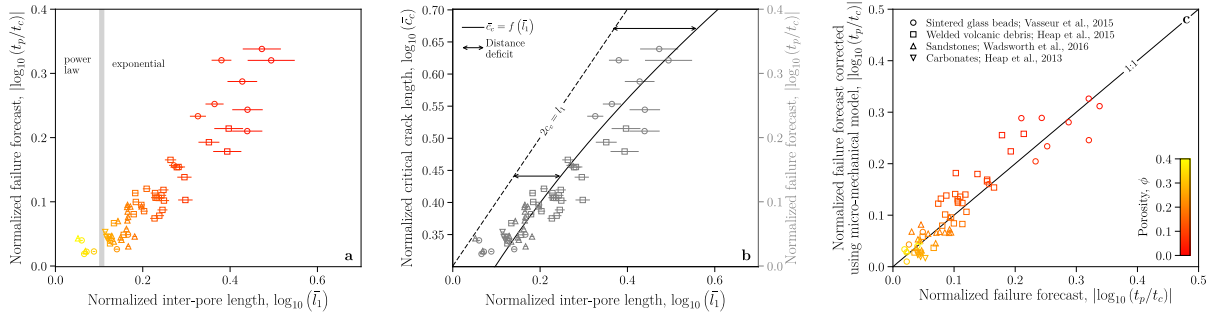
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525 Figure 6 – The dependence of the forecast error (cast as the ratio between the predicted failure time t_p
 526 from the TROL and the observed failure time t_c) on the sample porosity ϕ (or heterogeneity index H
 527 defined in Vasseur et al. (2015)) for a range of rock types and material analogues (Heap et al., 2013,
 528 2015; Vasseur et al., 2015; Wadsworth et al., 2016). *Inset* – the transition from an exponential to a power
 529 law approach of the acoustic emission rate to failure on a statistical basis (*see* text for definition of the
 530 statistical ΔBIC criterion). The vertical grey bar marks the approximate transition between a power-law
 531 and an exponential approach to failure and is the same as in Fig 7.

532



533

534 Figure 7 – Testing the micromechanical origin of errors in failure prediction. **a.** The ratio between the
 535 predicted failure time t_p from the TROL and the observed failure time t_c as a function of the normalized
 536 mean nearest-neighbour length \bar{l}_1 . The vertical grey line represents the transition between low \bar{l}_1 where
 537 the acoustic emission output as failure is approached is a power-law and high \bar{l}_1 where this approach to
 538 failure is an exponential function (see Fig 6 *inset*). **b.** The equilibrium crack lengths at failure from a
 539 micromechanical model for deformation of porous solids compared with the mean nearest-neighbour
 540 length (solid line calibrated in Fig 2; Sammis and Ashby, 1986). Shown for comparison are the data
 541 from panel **a** (grey data) showing that the failure forecast discrepancy grows as the critical crack length
 542 at failure becomes less than the half-distance between pores. **c.** The empirical correction proposed herein
 543 provides well-resolved failure forecasts.

544