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6

### 7 Abstract

8 Lightweight cores, based on an egg-box core design, have been manufactured using a simple compression-9 moulding technique. Two types of composite prepreg were used to manufacture the core materials, these being a 10 woven carbon fibre reinforced epoxy and a woven glass fibre reinforced epoxy. The resulting cores were of a 11 high quality, exhibiting little or no wrinkling following the manufacturing procedure. Subsequent compression 12 tests at quasi-static rates of loading showed that the compression strength of the core depended strongly on the 13 level of constraint applied during testing, with sandwich panels based on composite skins bonded to an egg-box 14 core offering a load-bearing capability that was more than double that of its unconstrained counterpart. The 15 quasi-static compression strength of the carbon-based cores has been shown to be slightly higher than the glass 16 fibre systems, particularly at higher core densities. Local splitting damage at cell joining regions and crushing of 17 the cell of the egg-box structure was identified as the primary failure mechanism in the sandwich panels.

Impact tests, conducted using a drop-weight impact tower, have shown that the compression strength of the eggbox cores is higher at dynamic rates of loading than at quasi-static rates. Here again, the local splitting and crushing was the primary mode of failure in the sandwich structures. Finally, the finite element technique has been used to model the mechanical response of these core designs under both quasi-static and impact loading testing conditions. Here, agreement between the predicted and observed responses was found to be good for both extremes of loading-rate.

Keywords: A. Carbon-fibre; A. Glass fibre; B. Impact behaviour; C. finite element analysis (FEA); composite
 egg-box

27

### 28 1. Introduction

29 Sandwich structures consisting of a low density core material bonded to strong, stiff outer skins are finding 30 increasing deployment across a broad range of engineering applications [1,2]. Such structures offer many 31 unique advantages, most particularly when subjected to out-of-plane loading, such as that associated with the 32 application of bending or flexural loads. Although sandwich technology is now well-established, there have 33 been many attempts in recent years to develop new and novel core designs that can greatly expand the design envelope. Examples include the development of advanced lattice designs that seek to exploit the tensile 34 35 deformation modes when loaded in compression as well as corrugated structures that offer increased levels 36 of ventilation in humid environments [3-8]. Kazemahvazi et al. [9] investigated the compression behaviour 37 of a corrugated system based on a carbon fibre reinforced epoxy resin. The resulting panels exhibited a 38 number of different failure modes as the geometry of the structure was varied. More recently, corrugated 39 core materials, based on both glass and carbon fibre reinforced epoxy composites, have been developed and 40 tested [10]. Here, the compression moulding technique, employing a steel mould with a triangular profile, 41 was used to produce a range of systems with differing wall thicknesses. The mechanical response of the 42 composite sandwich structures were compared to that offered by an all-aluminium system, where it was 43 shown that the specific compression strength of a carbon fibre-based core exceeded that of its metallic 44 counterpart [10].

45 Found et al. [11] performed quasi-static compression tests to investigate the energy absorption properties of 46 a polyurethane foam sandwich panel with four fibre-reinforced plastic tubular inserts incorporated within the 47 core. They reported that by ensuring progressive brittle failure of the structure, higher specific energy absorption values were obtained. As a result of variations in the fibre distribution within the inserts, the 48 49 sandwich tended to collapse in a catastrophic failure mode, leading to lower specific energy values. The 50 energy-absorbing characteristics of hierarchical woven lattice composites were evaluated by Zheng et al. 51 [12]. The square interlocking structures were composed of a woven lattice to form the sandwich cell walls. They concluded that these novel cell walls successfully restricted rib buckling. As a result, the structure had 52 53 a high compressive strength and a stable plateau region, thereby enhancing the specific energy absorption of 54 the cellular material.

55 A potential new class of energy-absorbing structure based on aluminium egg box was introduced by Zupan 56 et al. [13]. Experiments suggested that egg-box structures deform by either the rotation of a stationary plastic 57 hinge or by a travelling plastic knuckle, depending upon the in-plane kinematic constraints imposed upon the 58 egg-box. Chung et al. [14] fabricated composite egg-box structures and stated that its density, boundary 59 conditions and geometry affected the energy absorption capability of the structure. Fibre reinforced 60 composite structures were manufactured using vacuum bagging and autoclave curing techniques. The 61 production of foam-filled egg-box sandwiches, via autoclave curing, was investigated by Yoo et al. [15]. It 62 was found that such structures offered an impressive energy absorption capacity, involving a stable collapse 63 response, resembling that of an ideal energy-absorbing material.

Although extensive work has been carried out to understand the effect of various sandwich geometries on energy absorption, there is limited work relating to the mechanical properties of contoured core sandwich panels (or the egg-box structure) based on composite materials. The present study investigates the properties of contoured core sandwich panels based on both carbon and glass fibre composites. The study initially focuses on the quasi-static and impact response of these panels as a function of the cell wall thicknesses and core density. Following this, a series of finite element models are developed to predict the mechanical response of these structures under compression loading.

### 72 2. Experimental procedure

73 The egg-box composite cores investigated in this study were manufactured using either a woven glass 74 fibre reinforced epoxy (GFRP) or a woven carbon fibre reinforced epoxy (CFRP). The nominal 75 thicknesses of the GFRP and CFRP prepregs were 0.10 and 0.25 mm respectively. Details of physical 76 properties of these two prepreg materials are given in Table 1. Prepreg sheets were cut to the required 77 dimensions and placed between the two contoured aluminium moulds shown in Figure 1a. Geometrical 78 details of the mould design are given in Figure 1b. The GFRP cores were manufactured by stacking 5, 79 10 and 15 prepreg sheets in the mould, and the thicknesses of the resulting cores were 0.5, 1.0 and 1.5 80 mm respectively. CFRP cores having similar thicknesses were produced by stacking 2, 4 and 6 prepreg 81 sheets in the mould. A release agent (CIL Release 1711E, from Cilchem) was sprayed on both sides of 82 the mould to ensure easy demoulding at the end of the cure cycle. The aluminium moulds were then 83 placed in a hot press and the structure cured according to the processing parameters given in Table 1. 84 Here, the panels were heated to 125 °C at a heating rate of 1.5 °C/minute. This temperature was then maintained for 90 minutes, before switching off the press and allowing the samples to cool to room 85 temperature. The panels were then removed from the press and cut into 100 x 100 mm test samples, as 86 87 shown in Figure 2(a).

To manufacture the bonded samples, skins were bonded to the core using a two-part epoxy resin (Araldite 420 A/B) in the ratio 10:4. All of the cores were bonded to 0.50 mm thick skins based on either CFRP or GFRP. The adhesive was applied to the core using a syringe. After bonding, the panels were cured in an oven at 120 °C for one hour. The manufactured sandwich panels are shown in Figure 2(b). An examination of the panels showed that they were free of defects, such as wrinkling or warping, suggesting that the weaves offered sufficient drapability to cope with the relatively complex mould design.

In the initial part of this investigation, unbonded plain core specimens (i.e. without skins) were subjected to quasi-static compression using an Instron 4505 universal test machine. Tests were conducted on two by two (100x100 mm) egg-box panels. Following this, a series of compression tests were performed on sandwich panels with similarly-sized cores and corresponding skins. In a number of tests, the lateral movement of the base of the cores was restricted to investigate the influence of boundary conditions on the compression response. All of the quasi-static compression tests were undertaken at a crosshead

displacement rate of 1 mm/minute. The crosshead movement was interrupted when the panel was fully
 crushed between the loading platens. The load-displacements response was converted to nominal stress strain curves by normalising the applied load by the planar area of the specimen and dividing the
 crosshead displacement by the original specimen height, respectively. Table 2 summaries the sandwich
 structures investigated under quasi-static compression loading, which includes ply number, sample
 dimensions and core density. Here, in specimen ID 'GF' represents glass fibre and 'CF' represents
 carbon fibre.

108 The compression tests were repeated at dynamic rates of loading using a drop-weight impact tower. A 109 flat square impacter (100 mm x 100 mm) with a mass up to 15 kg was dropped onto panels supported 110 on a steel base. The resulting impact force was recorded using a 10 kN piezo-electric load cell (Kistler 9321A) positioned under the steel base. The cell was connected to a charge amplifier (Kistler 5011) 111 112 using an insulated coaxial cable in order to amplify the resulting voltage signal. The recorded signal was then converted from an analogue to a digital format using a DAQ device (Measurement 113 114 Computing, USB 1208HS) and then converted to a force. A high speed camera (MotionPro X4, model 115 X4CU-U-4) was used to capture the displacement and velocity of the impactor. The camera was placed in the front of the impact rig to track the impactor and record displacement during the dynamic event, 116 117 as shown in Figure 3. Table 3 summarises the key parameters used in this part of the study, which include number of ply, cell wall thickness, sample dimensions, core density, drop height and impactor 118 119 mass.

120

### 121 **3.** Numerical procedure

122 Numerical models were developed to simulate the compression response of the sandwich structures under quasi-static and dynamic loading. The composite was modelled using user-defined Hashin's 3D 123 124 failure criteria for an anisotropic composite material. Figure 4 shows the finite element mesh of an egg-125 box core with the top skin removed. Here, the core was meshed using six-noded triangular solid 126 elements, while the composite skins were modelled using eight-noded brick elements, with an interface 127 defined between the former and the latter. The loading platens above and below the panel were meshed using discrete rigid elements. The size of the core corresponds to that used in the experimental study (i.e. 128 129 100 x100 mm). Mesh sensitivity was investigated by varying the mesh density within the plane and

130 through-thickness directions of the composite sheet. Following this study, a mesh size based on element 131 with a size of 1 mm within the plane and two elements through-the-thickness of the composite layer was 132 used. A number of interfaces were considered in the model, including those between the face sheets and 133 the loading platen, those between the composite core and the face sheets, as well as possible self-contact 134 between the inclined faces of the egg-box core. A modified 3D failure criteria [16, 17] was used to simulate the response of sandwich panels in a Cartesian coordinate system. The failure criteria, together 135 136 with the related constitutive model, were then implemented in ABAQUS/Explicit using a subroutine [18, 137 19]. The failure criteria can be expressed as follows:

138 Fibre tension:  $(\sigma_{11} > 0)$ 

139 
$$F_{f}^{t} = \left(\frac{\sigma_{11}}{X_{1t}}\right)^{2} + \left(\frac{\sigma_{12}}{S_{12}}\right)^{2} + \left(\frac{\sigma_{13}}{S_{13}}\right)^{2}, \ d_{ft} = 1$$
(1)

140 Fibre compression:  $(\sigma_{11} < 0)$ 

$$\frac{|\sigma_{11}|}{X_{11}}, d_{fc} = 1, \ d_{fc} = 1$$
(2)

142

141

143 Matrix tension:  $(\sigma_{22} + \sigma_{33} > 0)$ 

$$\frac{(\sigma_{22} + \sigma_{33})^2}{X_{2t}^2} + \frac{\sigma_{23}^2 - \sigma_{22}\sigma_{33}}{X_{23}^2} + \frac{\sigma_{12}^2 + \sigma_{13}^2}{X_{12}^2} = 1, \ d_{mt} = 1$$
(3)

145

144

146 Matrix compression: 
$$(\sigma_{22} + \sigma_{33} < 0)$$

147 
$$\left[ \left( \frac{X_{2c}}{2S_{23}} \right)^2 - 1 \right] \frac{(\sigma_{22} + \sigma_{33})}{X_{2c}^2} + \frac{(\sigma_{22} + \sigma_{33})^2}{4S_{23}^2} + \frac{\sigma_{23}^2 - \sigma_{22}\sigma_{33}}{X_{23}^2} + \frac{\sigma_{12}^2 + \sigma_{13}^2}{X_{12}^2} = 1, \ d_{mc} = 1$$
(4)

148

where  $X_{1t}$ ,  $X_{1c}$ ,  $X_{2t}$ ,  $X_{2c}$ ,  $S_{12}$ ,  $S_{13}$  and  $S_{23}$  are the various strength components and  $d_{ft}$ ,  $d_{fc}$ ,  $d_{mt}$  and  $d_{mc}$  are the damage variables associated with the four failure modes. A series of numerical studies, with different durations, was conducted in order to identify the appropriate time-step that gave negligible dynamic effects. This time-step was found to be 0.1 seconds. The response of the material after damage initiation, which describes the rate of degradation of the material stiffness once the initiation criterion is satisfied, is defined by the equation:

155  $\sigma_{ii} = C_{ii}(d) \cdot \varepsilon_{ii}$ 

(5)

where  $C_{ij}(d)$  is the degradation matrix. The instant damage criteria are used here, i.e. the damage variables are either taken as zero (virgin state) or unity (damaged state). Therefore, the degradation matrix components are computed in terms of undamaged elastic constants,  $C_{ij}^{o}$ , and the damage variables as follows:

160 
$$C_{ij}(d) = (1 - d_{ij})C_{ij}^{o}$$
 (6)

Here, the damage variable  $d_{ij}$  is related to fibre and matrix damage in tension and compression, as well as shear failure in matrix caused by tension and compression, which can have various forms.

The response of the sandwich structures under dynamic loading was modelled using the same elements that were employed in the quasi-static models. The impactor was modelled as a flat plate using a discrete rigid surface. A point mass, equal to that of the experimental impactor, was assigned to a reference point located at the centre of the flat plate. The reference point was also used to record the displacement from this model. An initial velocity was prescribed to the rigid plate, which was set equal to the impact velocity used in the experiments. A surface-to-surface contact condition was used to define contact between the impactor and the skin (so as the core if the skin is damaged).

The input data for the elastic properties and for progressive damage development in this model were based on the properties given in Table 4. A numbers of studies have shown that increased strain-rates, can result in enhanced mechanical properties of composite materials [20-23]. It is generally accepted that the sensitivity of mechanical properties at high-strain rate is dependent on the composite type and polymer matrix.

175

### 176 4. Results and discussion

### 177

### 4.1 The effect of local constraint on the compression response of the cores

The mechanical properties of composite cores similar to those under investigation in this study clearly depend on the level of constraint applied to their boundaries, including the upper and lower surfaces as well as at their edges. Figure 5 shows typical stress-strain plots following compression tests on egg-box cores subjected to three different boundary conditions. Here, the stress and strain are nominal ones, which are defined in Section 2 Experimental Procedure. As expected, the plain unbonded core exhibits a

relatively low compressive modulus, as well as a modest compressive strength. Following the initial peak 183 184 in the curve, the stress drops before rising slowly and dropping on a number of subsequent occasions. An 185 examination of the samples during failure highlighted local splitting at cell joining regions and crushing of 186 the cell, general flattening of the core, delamination between the layers of the composite and finally 187 fracture across the fibres had high levels of compressive strain. Constraining the lateral movement of the 188 edges of the samples yielded a 40% increase in the average peak stress from the unbonded sample. The 189 ensuing collapse and crushing processes resulted in a much higher value of average stress and greater 190 energy absorption, defined by the area under the stress-strain curve, than in the unbonded (unconstrained) 191 sample. Here, mid-way through the crushing process, the nominal stress reached a value similar to that of 192 the initial peak. The failure mechanisms in the constrained samples again involved local splitting at cell 193 joining regions and crushing of the cell prior to complete collapse. Finally, the sandwich panel with 194 composite skins bonded to the composite core offered the highest compression strength of the three 195 conditions investigated here, with the peak value being approximately 2.3 times of that measured on the 196 plain, unconstrained core. Following the peak value, the stress dropped rapidly to values that were 197 significantly lower than those associated with the constrained (bonded) core. The failure modes observed 198 during the damage process in the sandwich panels included local crushing of the core, fibre fracture, 199 delamination between the plies in the core material and debonding at the skin-core interface.

200

### 0 4.2 Compression properties of the sandwich panels

201 Figures 6(a) and 6(b) show typical stress-strain traces for the GFRP and CFRP sandwich panels 202 respectively. In Figure 6(a), all three traces exhibit an initial linear response up to the peak stress. The peak 203 stress increases with web thickness, ranging from 0.44 MPa for the 0.5 mm web to 1.60 MPa for the 1.5 204 mm thick web. Following the peak in the trace, a crack initiated in core cell wall, which propagated under 205 continued loading, resulting in steady load drops as the cells collapsed and subsequently crushed. Following this, the core cell wall started to buckle, leading to a sudden drop in stress at strains between 0.1 206 207 and 0.2 mm/mm. Beyond a strain of approximately 0.2 mm/mm, the curves plateaued, as the cell walls 208 debonded from the skin, core flattened between the platens. Finally, the stress begins to increase at high 209 strains as the core begins to densify between the platens.

In Figure 6(b), there is again a linear increase over the initial portion of the stress-strain trace of CFRP
panels. The maximum stress increases from 0.46 MPa for the 0.5 mm thick web, to 1.61 MPa for its 1.5

212 mm thick counterpart. A comparison with Fig 6(a) indicates that the strength of the CFRP core is slightly 213 higher than that measured on the GFRP core. Following the peak in trace, the drop in stress is smoother 214 than for the GFRP core. The drop in stress for the 1.0 and 1.5 mm thick systems is continuous until the 215 densification threshold is reached.

Figure 7 summarises the variation of the quasi-static compression stress with core density for the glass and carbon-based sandwich structures, where it is clear that the stress of both materials increases in a roughly linear manner. Here, the core density is defined as the mass of the core divided by the core volume (including cavities). For the lowest density, the strengths are similar, however as density is increased, the superior properties of the CFRP core become apparent.

221 Figures 8(a) and 8(b) compare the results of the finite element predictions with the data from the 222 experimental traces. Agreement between the two sets of curves is generally good for the GFRP panels, 223 with the model accurately predicting the trends in the experimental results. The model accurately predicts 224 the initial slope, i.e. the elastic modulus of the core, but slightly over-predicts the subsequent softening 225 phase following the peak stress. The oscillations in the predicted traces are due to the unstable response 226 during local collapse in the FE models. Figure 9 presents a comparison of the predictions of the finite 227 element model with the experimentally-measured compression strengths for the CFRP panels. Agreement 228 between the two sets of data is good, suggesting that the numerical model can be used to predict the 229 compression response of structures similar to those tested here. Similar levels of agreement were observed 230 following comparisons between the predicted and measured properties for the GFRP system.

Figure 10 compares the failure mechanisms in the GFRP samples with the predictions offered by the FE model. Agreement between the predictions and experimental data is generally good, with the model predicting local crushing in the core as well as flattening of the core against the upper and lower skins. The model also predicts the final increase in stress due to densification. Agreement between the FE predictions and the experimental curves is also very good for the CFRP panels. Once again, the model accurately captures the initial elastic response, as well as the post-peak softening response of the core. Here, the softening portion of the stress-strain trace is somewhat smoother than observed for the GFRP core.

Figure 11 compares the quasi-static compressive strengths of the CFRP or GFRP egg-box cores with corresponding data for a number of plain foams. From the figure it is evident that the properties of the eggbox cores lie between those associated with linear PVC foams and crosslinked PVC foams. This overall

performance is somewhat disappointing given that the current systems are based on composite materials
rather than foamed polymers. This relatively modest performance can be attributed to the fact that the
composite systems fail at relatively low load levels, due to local crushing in the composite cores.

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- 245

### 5 4.3 The Impact response of the core materials

246 Figures 12(a) and 12(b) show typical impact traces for the GFRP and CFRP sandwich panels respectively. 247 The GFRP curves exhibit oscillations, due to ringing effects in the load-cell. The curves are similar in form 248 to those following quasi-static testing. As expected, the maximum stress increases with web thickness, passing from approximately 0.44 MPa for the thinnest web to 1.94 MPa for the 1.5 mm thick system. 249 250 There is evidence to suggest that the onset of densification occurs at lower strains under impact than at 251 quasi-static rates. The dynamic response of the CFRP sandwich structures is similar to the quasi-static 252 traces. Once again, the stress in the thickest sample drops steadily from the peak value. Here, the peak stress for the 1.5mm thick CFRP panel is 2.20 MPa, compared to 0.77 MPa for the 0.5 mm thick CFRP 253 254 panel.

Figure 13 compares the dynamic and quasi-static compression strengths of the GFRP sandwich structures. It is interesting to note that for the two thinner webs, the compression strength of the dynamically-loaded samples is similar to those of their quasi-static counterparts. In contrast, loading-rate effects are evident in the thickest system, with the dynamic value being 35% higher than the quasi-static system. Figure 13(b) presents the dynamic and quasi-static properties of the CFRP sandwich panels. Here, there is evidence of loading-rate sensitivity, with the impact strength of both the thinnest and thickest webs being up to 25% higher than at quasi-static rates.

Figure 14 compares the stress-strain responses predicted by the FE models with the experimental traces for the CFRP panels. As before, the FE model exhibits a highly oscillatory response for all web thicknesses. A comparison of the numerical and experimental traces indicates that the finite element model captures the fundamental features of the experimental stress-strain traces. However, following the initial peak in the stress-strain traces the FE model tends to over-predict the softening phase of the curves.

Figure 15 compares the FE predictions of the compression strength of the GFRP sandwich panels with the experimental data where good agreement between the two sets of data is apparent. Similar trends were

269 observed when the predictions and experimental data for the CFRP samples are compared, although the 270 model tended to slightly over-estimate the measured values. This evidence further supports the argument 271 that the finite element model is suitable for predicting the dynamic behaviour of these relatively complex 272 sandwich structures.

273

### **5.** Conclusions

A range of egg-box composite cores have been manufactured via a compression moulding process in a hot 275 press. Compression tests at quasi-static and dynamic loading-rates have identified a range of failure modes, 276 277 including flattening of the core webs, local crushing of the cell and delamination within the cell walls, 278 followed by the debonding between core and skins. The influence of edge constraint has been studied, where it was noted that sandwich structures based on composite skins bonded to an egg-box core offered 279 280 the highest compression strength. Increasing the thickness of the web in the egg-box core served to 281 increase the compression strength, although failure always occurred as a result of delamination and local crushing in the relatively thin inclined faces of the inclined core members. The compression properties of 282 both materials exhibit a degree of loading-rate sensitivity, with GFRP being slightly more loading-rate 283 284 sensitive than its carbon-based counterpart. Finite element models, developed to predict the quasi-static 285 and dynamic compression behaviour of the egg-box cores, have shown a reasonably good agreement with 286 the experimental data.

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Prepreg	GFRP	CFRP
Fibre type	E-Glass	3k HTA
Weave style	Satin	Plain
Resin content (% wt)	$40 \pm 3$	53 ± 3
Curing temperature (°C)	125	125
Dwell time (minutes)	90	90
Laminate density (kg/m <sup>3</sup> )	1780	1300
Nominal thickness of ply (mm)	0.10	0.25

Table 1. Details of the glass fibre and carbon fibre reinforced epoxy composites

Table 2. Summary of the dimensions of the sandwich structures investigated at quasi-static rates of strain.

Specimen ID	No.	Thickness	Specimen	Specimen	Specimen	Core
	of	't' (mm)	Length	Width	Height	density
	plies		(mm)	(mm)	(mm)	$(kg/m^3)$
GF1	5	0.5	100	100	20.5	54.5
GF2	10	1.0	100	100	21.0	104.3
GF3	15	1.5	100	100	21.5	141.3
CF1	2	0.5	100	100	20.5	50.1
CF2	4	1.0	100	100	21.0	97.4
CF3	6	1.5	100	100	21.5	130.3
			(			

Table 3. Summary of the dimensions of the sandwich structures and test parameters for low velocity impact

testing.

Specimen ID	No.	Thickness	Specimen	Specimen	Specimen	Core	Drop	Impactor
	of	't' (mm)	Length	Width	Height	density	height	mass
	plies		(mm)	(mm)	(mm)	$(kg/m^3)$	(m)	(kg)
GF4	5	0.5	100	100	20.5	54.5	0.40	8.43
GF5	10	1.0	100	100	21.0	104.3	0.85	8.43
GF6	15	1.5	100	100	21.5	141.3	1.40	8.43
CF4	2	0.5	100	100	20.5	50.1	0.50	15.70
CF5	4	1.0	100	100	21.0	97.4	1.10	15.70
CF6	6	1.5	100	100	21.5	130.3	1.45	15.70

Properties	Symbol	(GFRP)	(CFRP)
Young's modulus in longitudinal direction	$E_{11}$	23 GPa	48 GPa
Young's modulus in transverse direction	E <sub>22</sub>	23 GPa	48 GPa
Young's modulus in thickness	E <sub>33</sub>	5 GPa	1 GPa
In-plane shear modulus	G <sub>12</sub>	5 GPa	9 GPa
Through-thickness shear modulus	G <sub>13</sub> , G <sub>23</sub>	5 GPa	9 GPa
In-plane Poisson's ratio	v <sub>12</sub>	0.15	0.10
Through-thickness Poisson's ratio	v <sub>13</sub> , v <sub>23</sub>	0.15	0.10
Longitudinal tensile strength	$T_L$	320 MPa	550 MPa
Longitudinal compressive strength	$C_L$	260 MPa	150 MPa
Transverse tensile strength	T <sub>T</sub>	320 MPa	550 MPa
Transverse compressive strength	CT	260 MPa	350 MPa
Transverse shear strength	ST	100 MPa	120 MPa
Longitudinal shear strength	SL	100 MPa	120 MPa

Table 4. Summary of material properties of two composites used in this study.

compressive s... shear strength st.



Figure 1. Photographs of the mould and profile of the mould showing the egg-box design.



Figure 2. Photograph of (a) a GFRP core after manufacture and (b) the resulting CFRP and GFRP sandwich structures based on composite skins bonded to the core.







Figure 4. Finite element mesh of the egg-box model.



Figure 5. The effect of edge constraint and skins (bonded) on the compression response of the GFRP system.



Figure 6. Typical stress-strain traces following quasi-static compression tests on (a) the GFRP sandwich panels and (b) the CFRP sandwich panels.



Figure 7. Variation of compression strength with core density for the GFRP and CFRP panels.



Figure 8. Comparison of the numerical and experimental stress-strain traces for the GFRP and CFRP cores.





line correspond to the FE predictions.



Figure 10. Photographs showing the compression process in the flat roof GFRP structures (a) experimental observation and (b) finite element predictions.



Figure 11. Comparison of the compression strength of the egg box cores with other core types as a function of core density.



Figure 12. Typical stress-strain traces following impact compressions tests on (a) the GFRP sandwich panels and (b) the CFRP sandwich panels.



Figure 13. Comparison of the impact and quasi-static compression strengths of the (a) GFRP and (b) CFRP panels.



Figure 14. Comparison of the dynamic numerical and experimental stress-strain traces for the CFRP cores.



Figure 15. Impact compression strength versus core densities for GFRP samples. The solid line correspond to

the FE predictions.