



# Modeling of the nonlinear dynamic degradation characteristics of fiber-reinforced composite thin plates in thermal environment

Hui Li · Tinan Zhang · Zelin Li ·  
Bangchun Wen · Zhongwei Guan

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**Abstract** In this research, the nonlinear dynamic degradation characteristics of fiber-reinforced composite plates in thermal environment are investigated through a novel dynamic degradation model, which is established by introducing the thermal and time fitting coefficients simultaneously to express dynamic elastic moduli of such composite materials. Based on the classical laminated plate theory, the improved exponential function method, the complex modulus approach and the Ritz method, the dynamic equations are derived to solve the dynamic parameters. Besides, a particle swarm optimization algorithm is employed to iteratively calculate dynamic elastic moduli, and the nonlinear least squares technique in MATLAB software

is utilized to draw the three-dimensional fitted surfaces of dynamic elastic moduli, degradation time and temperature data, so that the concerned fitting coefficients in the model developed can be identified. In order to validate the dynamic degradation model, experimental measurements of E120 carbon fiber/FRD-YG-03 resin composite thin plates are undertaken. The first three natural frequencies, resonant responses and modal damping ratios obtained from the model are compared with experimental results at different degradation time and stabilized thermal environment, which are shown to be in good agreement. Also, the specific influences with and without consideration of the degradation behavior on the dynamic characteristics are investigated and evaluated.

H. Li (✉) · T. Zhang · Z. Li · B. Wen  
School of Mechanical Engineering and Automation,  
Northeastern University, Shenyang 110819, China  
e-mail: lh200300206@163.com

T. Zhang  
e-mail: ztinan@foxmail.com

Z. Li  
e-mail: 1299478310@qq.com

B. Wen  
e-mail: bcwen1930@vip.sina.com

H. Li · T. Zhang · Z. Li · B. Wen  
Key Laboratory of Vibration and Control of  
Aero-Propulsion System Ministry of Education,  
Northeastern University, Shenyang 110819, China

H. Li · Z. Guan  
School of Engineering, University of Liverpool, Brownlow  
Street, Liverpool L69 3GQ, UK  
e-mail: Zhongwei.Guan@liverpool.ac.uk

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## 1 Introduction

Fiber-reinforced composites are widely used in aeronautical, astronautical, naval vessel and weapon industries due to their light weight and excellent mechanical performance [1,2]. Currently, there are a large number of such composite thin plate structures that are in service in thermal environment, such as composite panels in high-speed aircraft, high-temperature turbine blades in aero engines and composite wings in

Author Proof

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the solar unmanned aircraft. Due to the effect of high-temperature environment (which may reach to hundreds or thousands of degrees Celsius), after a period of servicing time, those composite materials and structures will undergo a certain level of dynamic degradation or aging phenomenon [3–5]. This will lead to weakened stiffness and strength, excessive structural vibration, etc.; sometimes even lead to structural function loss, thus causing a possible catastrophic accident for the whole working components and systems.

Usually in the degradation process, the macroscopic dynamic properties of composite materials and structures decline. For example, the natural frequencies and dynamic stiffness decrease gradually [6,7]. Also, some complex changes occur in the vibration stress, dynamic response and damping behavior [8,9]. From the microscopic scale, the dynamic degradation is an irreversible process, which inevitably includes the matrix cracking, delamination damage, interface degradation, inelastic deformation and failure [10–14].

In the past decades, many scholars and researchers mainly focused on the static degradation characteristics of fiber-reinforced composite materials and structures, where the effects of different temperature range and oxidation environment (dry air, fuel gas, water vapor, etc.) on the mechanical degradation properties were studied [11,15–23,25]. For example, McManus [15] proposed an analytical method to calculate stresses and damages in fiber/polymer matrix composites due to the material degradation in thermal environment. Chung et al. [16] established a model by a time–temperature superposition method to predict the degradation effect on the weight of carbon fiber/epoxy composite materials. Zinchenko et al. [17] investigated the effects of thermal degradation of carbon fiber-reinforced plastics and predicted the mass loss based on the physical and mathematical models. Wang et al. [18] conducted an experimental study on the degradation of mechanical properties of carbon and glass fiber-reinforced polyester bars at elevated temperatures. Wolfrum et al. [11] proposed an empiric method that could be used to estimate the degradation of long-term mechanical properties of carbon fiber-reinforced composites in thermal environment based on the experimental data. Dawood and Rizkalla [19] conducted an accelerated environmental exposure tests to evaluate bond strength, yield strength, stiffness degradation of the CFRP systems for strengthening steel structures. Lafarie-Frenot et al. [20] studied the degradation behavior of composite beams

in a thermal environment of 150 °C by the numerical model developed. Upadyaya et al. [21] established a mechanism-based multiscale model for degradation prediction of polymer matrix composites at a thermo-oxidative aging condition. Moreover, Bojja et al. [22] predicted the stiffness degradation behavior of GFRP nanocomposites at block amplitude fatigue loads from micro-mechanics. Khalili et al. [23] investigated the effect of thermal cycling from –30 °C to +220 °C on the tensile behavior of composite laminate plate. By fitting the electrical impedance measurement results, Ndiaye et al. [24] proposed an acoustical characterization method to quantify the thermal aging behavior of composite plates and honeycomb sandwiches. Guo et al. [25] analyzed the effects of thermal-oxidative aging on the mechanical properties of glass fiber-reinforced polypropylene composites.

To improve the service life of fiber-reinforced materials and structures, the influence of dynamic loads needs to be considered, since the damage possibilities due to vibration loads or other shock loads are often greater than the static loads. Fortunately, some research progresses have been made in this field recently [26–32]. However, most of them only focused on the degradation behavior associated with the increasing environmental temperatures while ignoring the influence of degradation time. For example, Huang and Shen [26] investigated the dynamic behavior of simply supported functionally graded plates in thermal environment. The analytical results revealed that high temperature had an important effect on the reduction of structural natural frequencies. Melo and Radford [27] evaluated the temperature and frequency effects on the viscoelastic properties of fiber-reinforced composites with polymeric matrix (PEEK/IM7 composite). The results showed a decreasing trend in the storage moduli, but an increasing trend in the loss factors as the temperature increased. Based on the temperature-dependent material assumption, Duc et al. [28] established an analytical model to investigate the nonlinear dynamic behavior of the piezoelectric eccentrically stiffened FGM plates in thermal environment. Wu et al. [29] investigated the nonlinear relationships between the dynamic properties and degradation time of the metal and C/SiC plate specimens in high-temperature environment of 1200 °C by finite element simulations and experimental tests. However, only the degradation properties of natural frequencies at different heating time were studied; the changing trends of dynamic responses and damping were not

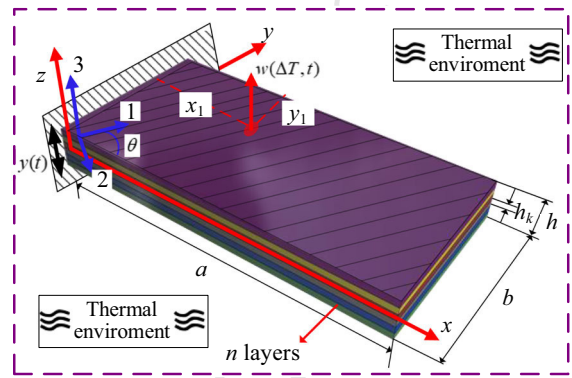
140 reported. Jakkamputi and Rajamohan [30] experimen-  
 141 tally investigated the natural frequencies and modal  
 142 damping ratios of CNT-reinforced hybrid polymer  
 143 composite beams under clamped-free and clamped-  
 144 clamped boundary conditions when the temperature  
 145 increased from 30 to 60 °C. Liu et al. [31] measured  
 146 the time-temperature-dependent elastic moduli of loss  
 147 factors of CFRP at various temperatures and found that  
 148 more elastic-like behavior would be induced by the  
 149 higher frequencies. Based on the experimental and sim-  
 150 ulation results, Bai et al. [32] studied on the temperature  
 151 effects on modal parameters of composite honeycomb  
 152 structure under the suspended boundary condition.

153 The literature survey presented here indicates that  
 154 there is limited research work on dynamic degrada-  
 155 tion modeling techniques of fiber-reinforced compos-  
 156 ite plates in thermal environment, especially with the  
 157 lack of systematic analysis and solution of structural  
 158 resonant responses and damping characteristics. Con-  
 159 sidering abovementioned issues, this research proposed  
 160 a novel dynamic degradation model by introducing the  
 161 thermal and time fitting coefficients simultaneously  
 162 to express dynamic elastic moduli of such compos-  
 163 ite materials. Moreover, a particle swarm optimization  
 164 algorithm with high convergence efficiency and accu-  
 165 racy is employed to iteratively calculate the concerned  
 166 elastic moduli, which pave a very practical way to facil-  
 167 itate the data fitting operation to identify the key fit-  
 168 ting coefficients. Also, a series of experimental tests  
 169 are conducted to validate the model. The theoretical  
 170 and experimental findings in this research provide an  
 171 insight of how the nonlinear dynamic parameters of  
 172 composite thin plate structures are affected by long-  
 173 term degradation time over a wide range of stabilized  
 174 thermal conditions.

175 **2 Modeling and solution**

176 **2.1 Description of the dynamic degradation model**

177 Assume that a fiber-reinforced composite thin plate  
 178 (FCTP), as seen in Fig. 1, which is made of fiber and  
 179 matrix materials with  $n$  layers, is in a uniform thermal  
 180 environment. First, the coordinate system  $xyz$  is set in  
 181 the mid-plane, and the length, width and thickness are  
 182 assumed to be  $a$ ,  $b$  and  $h$ , respectively. In this theoret-  
 183 ical model, the fiber direction within a certain layer is  
 184 defined as an angle  $\theta$  from the  $x$ -axis of the coordinate,



185 **Fig. 1** A theoretical model of fiber-reinforced composite thin  
 186 plate in thermal environment

185 and each layer is located between  $h_{k-1}$  and  $h_k$  along  
 186 the  $z$ -axis with an equal thickness. There is also a local  
 187 coordinate system, where “1” represents the direction  
 188 parallel to the fiber, “2” the direction perpendicular to  
 189 the fiber and “3” the direction perpendicular to the 1–2  
 190 plane. In addition, assume that FCTP is under cantilever  
 191 boundary condition and is subjected to a base excitation  
 192 load  $y(t)$ . The concerned vibration displacement  
 193  $w(\Delta T, t)$  is located at a point  $(x_1, y_1)$ , which is affected  
 194 by temperature change  $\Delta T$  relative to the room temper-  
 195 ature (20 °C) and degradation time  $t$  (or heating time)  
 196 simultaneously.

197 Based on the observed downward trend [33,34] of  
 198 elastic moduli of fiber-reinforced composite at differ-  
 199 ent temperatures with considering heating time, also  
 200 utilizing the improved exponential function approach  
 201 to introduce degradation time  $t$  and temperature change  
 202  $\Delta T$ , the dynamic elastic moduli of fiber-reinforced  
 203 composite materials,  $E'_1(\Delta T, t)$ ,  $E'_2(\Delta T, t)$ ,  $G'_{12}(\Delta T, t)$   
 204 are assumed to have the following forms

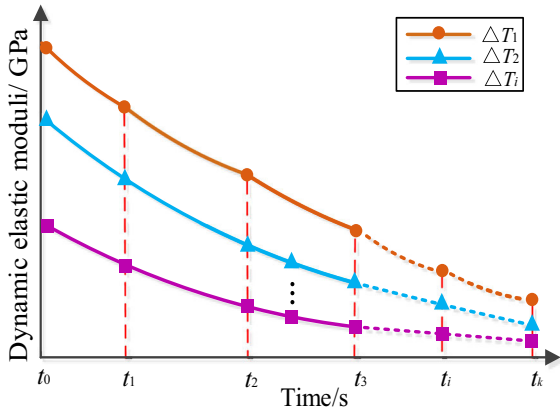
205 
$$E'_1(\Delta T, t) = E_1^R - A_1 \times \Delta T^{B_1} \times e^{-\frac{C_1}{t}} \quad (1)$$

206 
$$E'_2(\Delta T, t) = E_2^R - A_2 \times \Delta T^{B_2} \times e^{-\frac{C_2}{t}} \quad (2)$$

207 
$$G'_{12}(\Delta T, t) = G_{12}^R - A_{12} \times \Delta T^{B_{12}} \times e^{-\frac{C_{12}}{t}} \quad (3)$$

208 where  $E_1^R, E_2^R, G_{12}^R$  represent the elastic moduli in  
 209 room temperature. To consider the influence caused  
 210 by thermal environment (or temperature change  $\Delta T$ ),  
 211  $B_i (i = 1, 2, 12)$  are introduced to represent the thermal  
 212 fitting coefficients. In addition, to consider the influence  
 213 due to degradation time  $t$ ,  $C_i$  are introduced to repre-  
 214 sent the time fitting coefficients. Also,  $A_i$  are defined  
 215 as the adjustment coefficients, which can facilitate the

Author Proof



**Fig. 2** The schematic diagram of dynamic elastic moduli changed with temperature and degradation time

The expressions of thermal strains in the  $k$ th layer of FCTP are

$$\boldsymbol{\varepsilon}_{Ti}^{(k)} = \begin{bmatrix} \alpha_x \\ \alpha_y \\ \alpha_{xy} \end{bmatrix} \Delta T \quad (i = 1, 2, 12) \quad (8)$$

where  $\alpha_x, \alpha_y, \alpha_{xy}$  are the coefficients of thermal expansion along the  $x, y$  and shear directions, respectively.

Since  $\theta_k$  is the fiber angle in the  $k$ th layer of FCTP,  $\Phi$  and  $\Gamma$  are set as  $\Phi = \cos \theta_k$  and  $\Gamma = \sin \theta_k$  for the convenience of expression. Thus, the relationships between thermal expansion coefficients  $\alpha_x, \alpha_y, \alpha_{xy}$  along  $x, y$  and shear direction and  $\alpha_1, \alpha_2$  parallel and perpendicular to the fiber direction can be written as

$$\begin{bmatrix} \alpha_x \\ \alpha_y \\ \alpha_{xy} \end{bmatrix}^{(k)} = \begin{bmatrix} \Phi^2 & \Gamma^2 \\ \Gamma^2 & \Phi^2 \\ 2\Phi\Gamma & -2\Phi\Gamma \end{bmatrix}^{(k)} \begin{bmatrix} \alpha_1 \\ \alpha_2 \end{bmatrix} \quad (9)$$

The stress–strain relationship in any direction of the  $k$ th layer of FCTP affected by thermal environment and degradation time can be expressed as follows

$$\begin{bmatrix} \sigma_x \\ \sigma_y \\ \tau_{xy} \end{bmatrix}^{(k)} = \begin{bmatrix} \bar{Q}_{11} & \bar{Q}_{12} & \bar{Q}_{16} \\ \bar{Q}_{12} & \bar{Q}_{22} & \bar{Q}_{26} \\ \bar{Q}_{16} & \bar{Q}_{26} & \bar{Q}_{66} \end{bmatrix}^{(k)} \begin{bmatrix} \varepsilon_x^{(k)} - \varepsilon_{Tx}^{(k)} \\ \varepsilon_y^{(k)} - \varepsilon_{Ty}^{(k)} \\ \gamma_{xy}^{(k)} - \gamma_{Txy}^{(k)} \end{bmatrix} \quad (10)$$

where the strains are denoted by  $\varepsilon_1^{(k)} = \varepsilon_x^{(k)}, \varepsilon_2^{(k)} = \varepsilon_y^{(k)}, \varepsilon_6^{(k)} = \gamma_{xy}^{(k)}$ , while the thermal strains are denoted by  $\varepsilon_{T1}^{(k)} = \varepsilon_{Tx}^{(k)}, \varepsilon_{T2}^{(k)} = \varepsilon_{Ty}^{(k)}, \varepsilon_{T6}^{(k)} = \gamma_{Txy}^{(k)}$ .  $\bar{Q}_{ij}^{(k)}$  is the partial axis stiffness coefficient matrix in the  $k$ th layer of FCTP.

Since  $\bar{Q}_{ij}^{(k)}$  incorporates real and imaginary part, it can be expressed in the following form

$$\bar{Q}_{ij}^{(k)} = \bar{Q}'_{ij}{}^{(k)} + i\bar{Q}''_{ij}{}^{(k)} \quad (11)$$

where  $\bar{Q}'_{ij}$  and  $\bar{Q}''_{ij}$  are the real and imaginary parts of  $\bar{Q}_{ij}^{(k)}$ , respectively.

Let  $\bar{Q}_{ij}^{(k)} = \mathbf{H}^{(k)} \mathbf{Q}_{ij} \mathbf{H}^{T(k)}$  (the corresponding expression is shown in “Appendix”). Then, the partial axis stress transformation matrix  $\mathbf{H}^{(k)}$  in the  $k$ th layer of FCTP can be deduced as follows

data fitting operation when the fitting coefficients mentioned above are determined.

The schematic diagram of the change of dynamic elastic moduli with temperature and degradation time is shown in Fig. 2, where  $\Delta T_i$  ( $i = 1, 2, \dots, k$ ) is the temperature difference at each degradation time point.

The complex modulus method is utilized to consider the damping property of fiber-reinforced composites [35,36], according to which their complex dynamic elastic moduli can be further expressed as

$$E_1^*(\Delta T, t) = E'_1(\Delta T, t) \times (1 + i\eta_1) \quad (4)$$

$$E_2^*(\Delta T, t) = E'_2(\Delta T, t) \times (1 + i\eta_2) \quad (5)$$

$$G_{12}^*(\Delta T, t) = G'_{12}(\Delta T, t) \times (1 + i\eta_{12}) \quad (6)$$

where  $E_1^*(\Delta T, t), E_2^*(\Delta T, t)$  represent the complex dynamic moduli parallel and perpendicular to the fiber direction in thermal environment,  $G_{12}^*(\Delta T, t)$  represents the complex shear modulus, and  $\eta_1, \eta_2, \eta_{12}$  are the corresponding loss factors in those fiber directions.

Based on the classical laminated plate theory and the generalized Duhamel–Neumann form of Hooke’s Law [37,38], the relationships between stress and strain in the  $k$ th layer of FCTP in thermal environment can be expressed as

$$\boldsymbol{\varepsilon}_i^{(k)} = \bar{S}_{ij}^{(k)} \boldsymbol{\sigma}_i^{(k)} + \boldsymbol{\varepsilon}_{Ti}^{(k)} \quad (i, j = 1, 2, 6) \quad (7)$$

where the stresses are denoted by  $\sigma_1 = \sigma_x, \sigma_2 = \sigma_y, \sigma_6 = \tau_{xy}$ .  $\boldsymbol{\varepsilon}_i^{(k)}$  are the strains in the  $k$ th layer of FCTP,  $\boldsymbol{\varepsilon}_{Ti}^{(k)}$  are the thermal strains caused by the thermally induced internal forces, and  $\bar{S}_{ij}^{(k)}$  are the partial-axis flexibility coefficient matrix.

$$\mathbf{H}^{(k)} = \begin{bmatrix} \Phi^2 & \Gamma^2 & 2\Gamma\Phi \\ \Gamma^2 & \Phi^2 & -2\Gamma\Phi \\ -\Gamma\Phi & \Gamma\Phi & \Phi^2 - \Gamma^2 \end{bmatrix}^{(k)} \quad (12)$$

When the thermal environment and degradation time are both considered, the principal axis stiffness matrix  $\mathbf{Q}_{ij}$  in all layers of FCTP can be expressed by Poisson's ratio  $\mu_{12}, \mu_{21}$  and the complex dynamic elastic moduli as

$$\mathbf{Q}_{ij} = \begin{bmatrix} \frac{E_1^*(\Delta T, t)}{1-\mu_{12}\mu_{21}} & \frac{\mu_{12}E_1^*(\Delta T, t)}{1-\mu_{12}\mu_{21}} & 0 \\ \frac{\mu_{21}E_2^*(\Delta T, t)}{1-\mu_{12}\mu_{21}} & \frac{E_2^*(\Delta T, t)}{1-\mu_{12}\mu_{21}} & 0 \\ 0 & 0 & G_{12}^*(\Delta T, t) \end{bmatrix} \quad (13)$$

where  $\mu_{21} = \mu_{12} \frac{E_2^*(\Delta T, t)}{E_1^*(\Delta T, t)}$ .

According to the assumption of the laminated plate theory, the total strain in the  $k$ th layer of FCTP is

$$\boldsymbol{\varepsilon}_i^{(k)} = \boldsymbol{\varepsilon}_i^0(x, y, t) + z\boldsymbol{\chi}_i(x, y, t) \quad (14)$$

where  $\boldsymbol{\varepsilon}_i^0$  are the mid-plane strains, and  $\boldsymbol{\chi}_i$  are the curvatures.

Thus, the relationships between strain and displacement of FCTP can be expressed as

$$\begin{bmatrix} \varepsilon_x^0 \\ \varepsilon_y^0 \\ \gamma_{xy}^0 \end{bmatrix} = \begin{bmatrix} \frac{\partial u^0}{\partial x} \\ \frac{\partial v^0}{\partial y} \\ \frac{\partial u^0}{\partial y} + \frac{\partial v^0}{\partial x} \end{bmatrix}, \quad \begin{bmatrix} \chi_x \\ \chi_y \\ \chi_{xy} \end{bmatrix} = - \begin{bmatrix} \frac{\partial^2 w}{\partial x^2} \\ \frac{\partial^2 w}{\partial y^2} \\ \frac{\partial^2 w}{\partial x \partial y} \end{bmatrix} \quad (15)$$

where  $u^0$  and  $v^0$  are the mid-plane displacements in the  $x$  and  $y$  directions, respectively, and  $w$  is the displacement in the  $z$  direction.

The overall internal forces and moments can be obtained by integrating the stress of all layers along the  $z$ -axis

$$\begin{aligned} (N_x, N_y, N_{xy}) &= \int_{-h/2}^{h/2} (\sigma_x, \sigma_y, \tau_{xy}) dz \\ (M_x, M_y, M_{xy}) &= \int_{-h/2}^{h/2} (\sigma_x, \sigma_y, \tau_{xy}) z dz \end{aligned} \quad (16)$$

Substituting Eq. (14) into Eq. (10) and taking into account of the effects of Eq. (16), the overall internal forces and combined moments of FCTP can be obtained as

$$\begin{bmatrix} N_i \\ M_i \end{bmatrix} = \begin{bmatrix} A_{ij} & B_{ij} \\ B_{ij} & D_{ij} \end{bmatrix} \begin{bmatrix} \boldsymbol{\varepsilon}_i^0 \\ \boldsymbol{\chi}_i \end{bmatrix} - \begin{bmatrix} N_{Ti} \\ M_{Ti} \end{bmatrix} \quad (i, j = 1, 2, 6) \quad (17)$$

where  $N_{Ti}, M_{Ti}$  are the thermally induced internal forces and thermal moments, respectively.  $A_{ij}, B_{ij}, D_{ij}$  are the coefficient matrix. They can be expressed as

$$\begin{aligned} (N_{Ti}, M_{Ti}) &= \int_{-h/2}^{h/2} \bar{\mathbf{Q}}_{ij} \boldsymbol{\alpha}_i \Delta T(1, z) dz \\ \begin{cases} A_{ij} = \sum_{k=1}^N \bar{\mathbf{Q}}_{ij}^{(k)} (z_k - z_{k-1}) \\ B_{ij} = \frac{1}{2} \sum_{k=1}^N \bar{\mathbf{Q}}_{ij}^{(k)} (z_k^2 - z_{k-1}^2) \\ D_{ij} = \frac{1}{3} \sum_{k=1}^N \bar{\mathbf{Q}}_{ij}^{(k)} (z_k^3 - z_{k-1}^3) \end{cases} \end{aligned}$$

### 2.2 Solutions of the dynamic characteristics with considering degradation behavior

The composite plate is assumed to undergo a base excitation load  $y(t)$ , which can be regarded as a uniform inertial force loading  $q(t)$  with the following expression [39]

$$q(t) = -\rho h \frac{d^2 y(t)}{dt^2} = \rho h Y \omega^2 e^{i\omega t} \quad (18)$$

where  $\rho$  is the density,  $Y$  is the amplitude of base excitation amplitude, and  $\omega$  is the excitation angular frequency.

By referring to the dynamic equations listed in references [40, 41], the equation of vibration displacement of FCTP in thermal environment can be obtained and expressed as

$$\begin{aligned} 4D_{16} \frac{\partial^4 w}{\partial x^3 \partial y} + 4D_{26} \frac{\partial^4 w}{\partial x \partial y^3} + D_{22} \frac{\partial^4 w}{\partial y^4} - L_1 \frac{\partial u^0}{\partial x} \\ - L_2 \frac{\partial u^0}{\partial y} - L_3 \frac{\partial v^0}{\partial x} \\ - L_4 \frac{\partial v^0}{\partial y} + L_5 \frac{\partial^2 w}{\partial x^2} + \frac{\partial^2 M_{Tx}}{\partial x^2} + 2 \frac{\partial^2 M_{Txy}}{\partial x \partial y} \\ + \frac{\partial^2 M_{Ty}}{\partial y^2} + R \frac{\partial^2 w}{\partial t^2} \\ = q(t) - N_{Tx} \frac{\partial^2 w}{\partial x^2} - 2N_{Txy} \frac{\partial^2 w}{\partial x \partial y} - N_{Ty} \frac{\partial^2 w}{\partial y^2} \end{aligned} \quad (19)$$

where  $R$  is the integral of mass density through the plate thickness, and  $L_i$  are the operators with the following expressions

$$L_1 = B_{11} \frac{\partial^2}{\partial x^2} + (B_{12} + 2B_{66}) \frac{\partial^2}{\partial y^2}, \quad L_2 = 3B_{16} \frac{\partial^2}{\partial x^2}$$



$$\begin{aligned}
 &+ B_{26} \frac{\partial^2}{\partial y^2} \\
 L_3 = &B_{16} \frac{\partial^2}{\partial x^2} + 3B_{26} \frac{\partial^2}{\partial y^2}, L_4 = (B_{12} + 2B_{66}) \frac{\partial^2}{\partial x^2} \\
 &+ B_{22} \frac{\partial^2}{\partial y^2} \\
 L_5 = &D_{11} \frac{\partial^2}{\partial x^2} + 2(D_{12} + 2D_{66}) \frac{\partial^2}{\partial y^2}
 \end{aligned}$$

The concerned composite plate in Fig. 1 is symmetric about the mid-plane, so that the in-plane displacement and out-of-plane displacement are decoupled. Then, according to the small deflection theory of laminate plates, the principle of minimum potential energy and the Ritz method are combined to solve the strain energy  $V$  in thermal environment

$$\begin{aligned}
 V = &\frac{1}{2} \int_A \left\{ D_{11} \left( \frac{\partial^2 w(\Delta T, t)}{\partial x^2} \right)^2 \right. \\
 &+ 2D_{12} \frac{\partial^2 w(\Delta T, t)}{\partial x^2} \frac{\partial^2 w(\Delta T, t)}{\partial y^2} \\
 &+ D_{22} \left( \frac{\partial^2 w(\Delta T, t)}{\partial y^2} \right)^2 \\
 &+ 4D_{16} \frac{\partial^2 w(\Delta T, t)}{\partial x \partial y} \frac{\partial^2 w(\Delta T, t)}{\partial x^2} \\
 &+ 4D_{26} \frac{\partial^2 w(\Delta T, t)}{\partial x \partial y} \frac{\partial^2 w(\Delta T, t)}{\partial y^2} \\
 &\left. + 4D_{66} \left( \frac{\partial^2 w(\Delta T, t)}{\partial x \partial y} \right)^2 \right\} dA \quad (20)
 \end{aligned}$$

where  $A$  represents the plane area of FCTP.

Considering the influence of thermal environment, the potential energy of the system due to the thermally induced internal forces can be expressed as

$$\begin{aligned}
 U_{\text{Tem}} = &\frac{1}{2} \int_A N_{Tx} \left( \frac{\partial w(\Delta T, t)}{\partial x} \right)^2 + N_{Ty} \left( \frac{\partial w(\Delta T, t)}{\partial y} \right)^2 \\
 &+ 2N_{Txy} \frac{\partial w(\Delta T, t)}{\partial x} \frac{\partial w(\Delta T, t)}{\partial y} dA \quad (21)
 \end{aligned}$$

The function of uniform inertial force  $W_q$  is

$$W_q = \int \int_A q(t) w(\Delta T, t) dx dy \quad (22)$$

The kinetic energy  $\Lambda$  of the system in thermal environment can be expressed as

$$\Lambda = \frac{1}{2} \rho h \int_A \left( \frac{\partial w(\Delta T, t)}{\partial t} \right)^2 dA \quad (23)$$

Furthermore, the vibration displacement  $w(\Delta T, t)$  of a composite thin plate in thermal environment is assumed to be

$$w(\Delta T, t) = W(x, y, \Delta T, t) e^{i\omega t} \quad (24)$$

where  $W(x, y, \Delta T, t)$  represents the modal shape function, which can be defined as

$$W(x, y, \Delta T, t) = \sum_{m=1}^M \sum_{n=1}^N a_{mn}(\Delta T, t) X_m(x) Y_n(y) \quad (25)$$

where  $m$  and  $n$  represent the half wave number of the modal shapes along  $x$  and  $y$  directions, respectively,  $M$  and  $N$  are the maximum values of  $m$  and  $n$ .  $a_{mn}(\Delta T, t)$  is the coefficient affected by temperature and degradation time.  $X_m(x)$  and  $Y_n(y)$  are the corresponding modal functions along  $x$  and  $y$  directions, which can be expressed by the fixed-free beam function and free-free beam function.

Then, according to the Ritz method and neglecting the influence of the harmonic component  $e^{i\omega t}$ , the Lagrange energy function  $\Pi$  can be defined as

$$\Pi = V + U_{\text{Tem}} - W_q - \Lambda \quad (26)$$

In order to obtain the minimum of the Lagrangian function  $\Pi$ , Eq. (26) is partially derived with respect to  $a_{mn}$  and the following equation can be acquired

$$\frac{\partial V}{\partial a_{mn}} + \frac{\partial U_{\text{Tem}}}{\partial a_{mn}} - \frac{\partial \Lambda}{\partial a_{mn}} = \frac{\partial W_q}{\partial a_{mn}} \quad (27)$$

Substituting Eqs. (20)–(23) into Eq. (27), and letting  $\mathbf{K}^* = \frac{\partial V}{\partial a_{mn}} + \frac{\partial U_{\text{Tem}}}{\partial a_{mn}}$ ,  $\omega^2 \mathbf{M} = \frac{\partial \Lambda}{\partial a_{mn}}$ ,  $\mathbf{F} = \frac{\partial W_q}{\partial a_{mn}}$ , the following equation can be obtained

$$(\mathbf{K}^* - \omega^2 \mathbf{M}) \mathbf{a} = \mathbf{F} \quad (28)$$

where  $\mathbf{M}$  represents the mass matrix of the system,  $\mathbf{F}$  is excitation force vector,  $\mathbf{K}^*$  represents the complex stiffness matrix,  $\mathbf{K}^* = \mathbf{K} + i\mathbf{C}$ ,  $\mathbf{C}$  represents damping matrix, and  $\mathbf{a} = (a_{11}, a_{12}, \dots, a_{mn})^T$  is an eigenvector.

To solve the natural frequency and modal shape, it is only necessary to make the damping matrix  $\mathbf{C}$  and excitation force vector  $\mathbf{F}$  equal to zero

Author Proof

$$(K - \omega^2 M)q = 0 \tag{29}$$

By solving Eq. (29), the natural frequencies of FCTP affected by temperature and degradation time can be obtained. Then, by substituting the eigenvector  $q = (q_{11}, q_{12}, \dots, q_{mn})^T$  into Eq. (25), the concerned modal shapes can also be obtained.

Consequently, the coefficient matrix  $a_{mn}$  can be expressed as

$$a_{mn} = [K^* - \omega^2 M]^{-1} F \tag{30}$$

Substitute the vector  $a_{mn}$  into Eq. (25) and then into Eq. (24), and set  $e^{i\omega t} = 1$ , the concerned vibration displacement  $w(\Delta T, t)$  can be obtained.

Considering that usually the absolute vibration response  $\lambda(\Delta T, t)$  of a composite structure is obtained in experimental tests, it should include the structural vibration response  $w(\Delta T, t)$  and the base excitation displacement  $y(t)$ . Therefore,  $\lambda(\Delta T, t)$  can be further expressed as

$$\lambda(\Delta T, t) = y(t) + w(\Delta T, t) \tag{31}$$

The total strain energy  $\bar{V}_i$  in the  $i$ th mode of FCTP affected by temperature and degradation time can be expressed as

$$\bar{V}_i = \bar{V}(i, x) + \bar{V}(i, y) + \bar{V}(i, xy) + \bar{V}(i, \Delta T) \tag{32}$$

where  $\bar{V}(i, x)$ ,  $\bar{V}(i, y)$ ,  $\bar{V}(i, xy)$  represent the corresponding  $i$ th strain energy along the  $x$ ,  $y$ , and  $xy$  directions respectively and  $\bar{V}(i, \Delta T)$  is the  $i$ th potential energy caused by thermally induced internal forces.

$$\bar{V}(i, x) = \frac{1}{2} \sum_{k=1}^N \int_{h_{k-1}}^{h_k} \int_A \bar{Q}'_{ij} (\varepsilon_x - \varepsilon_{Tx}) \varepsilon_x dAdz$$

$$\bar{V}(i, y) = \frac{1}{2} \sum_{k=1}^N \int_{h_{k-1}}^{h_k} \int_A \bar{Q}'_{ij} (\varepsilon_y - \varepsilon_{Ty}) \varepsilon_y dAdz$$

$$\bar{V}(i, xy) = \frac{1}{2} \sum_{k=1}^N \int_{h_{k-1}}^{h_k} \int_A \bar{Q}'_{ij} (\gamma_{xy} - \gamma_{Txy}) \gamma_{xy} dAdz$$

$$\bar{V}(i, \Delta T) = \frac{1}{2} \int_A N'_{Tx} \left( \frac{\partial w(\Delta T, t)}{\partial x} \right)^2$$

$$+ N'_{Ty} \left( \frac{\partial w(\Delta T, t)}{\partial y} \right)^2 \tag{428}$$

$$+ 2N'_{Txy} \frac{\partial w(\Delta T, t)}{\partial x} \frac{\partial w(\Delta T, t)}{\partial y} dA \tag{429}$$

where  $N'_{Tx}$ ,  $N'_{Ty}$ ,  $N'_{Txy}$  can be obtained by solving the real part of complex stiffness matrix coefficients  $\bar{Q}'_{ij}$  in Eqs (11) and (17).

The total dissipated energy  $\Delta \bar{V}_i$  in the  $i$ th mode of FCTP affected by temperature and degradation time can be expressed as

$$\Delta \bar{V}_i = \Delta \bar{V}(i, x) + \Delta \bar{V}(i, y) + \Delta \bar{V}(i, xy) + \Delta \bar{V}(i, \Delta T) \tag{33}$$

where  $\Delta \bar{V}(i, x)$ ,  $\Delta \bar{V}(i, y)$  and  $\Delta \bar{V}(i, xy)$  represent the corresponding  $i$ th dissipated energy along the  $x$ ,  $y$  and  $xy$  directions, respectively, and  $\Delta \bar{V}(i, \Delta T)$  is the  $i$ th dissipated energy caused by thermally induced internal forces.

$$\Delta \bar{V}(i, x) = \pi \sum_{k=1}^N \int_{h_{k-1}}^{h_k} \int_A \bar{Q}''_{ij} (\varepsilon_x - \varepsilon_{Tx}) \varepsilon_x dAdz \tag{442}$$

$$\Delta \bar{V}(i, y) = \pi \sum_{k=1}^N \int_{h_{k-1}}^{h_k} \int_A \bar{Q}''_{ij} (\varepsilon_y - \varepsilon_{Ty}) \varepsilon_y dAdz \tag{443}$$

$$\Delta \bar{V}(i, xy) = \pi \sum_{k=1}^N \int_{h_{k-1}}^{h_k} \int_A \bar{Q}''_{ij} (\gamma_{xy} - \gamma_{Txy}) \gamma_{xy} dAdz \tag{444}$$

$$\Delta \bar{V}(i, \Delta T) = \pi \int_A N''_{Tx} \left( \frac{\partial w(\Delta T, t)}{\partial x} \right)^2$$

$$+ N''_{Ty} \left( \frac{\partial w(\Delta T, t)}{\partial y} \right)^2 \tag{446}$$

$$+ 2N''_{Txy} \frac{\partial w(\Delta T, t)}{\partial x} \frac{\partial w(\Delta T, t)}{\partial y} dA \tag{447}$$

where  $N''_{Tx}$ ,  $N''_{Ty}$ ,  $N''_{Txy}$  can be obtained by solving the imaginary part of complex stiffness matrix coefficients  $\bar{Q}''_{ij}$  in Eqs (11) and (17).

Finally, the  $i$ th modal damping ratio of FCTP with considering degradation behavior in thermal environment can be expressed as

$$\xi_i = \frac{\Delta \bar{V}_i}{4\pi \bar{V}_i} \tag{34}$$

Author Proof

**3 Determination of the fitting coefficients in the theoretical model**

Because thermal environment has a great influence on composite material parameters, here the identification principle of dynamic elastic moduli is described by combining theory with practice, which is a prerequisite for determining the concerned fitting coefficients at different temperatures and degradation time points.

**3.1 Iterative calculations of dynamic elastic moduli**

In this section, due to the high convergence efficiency, small computational complexity [42], a particle swarm optimization algorithm (PSOA) is employed to iteratively calculate the dynamic elastic moduli of fiber-reinforced composites, so as to obtain the theoretical natural frequency data to approach to the experimental natural frequency data. In the PSOA, firstly, a  $D$ -dimensional target search space (or  $D$  optimal value) needs to be chosen, and a group can be formed through  $I$  particles whose positions are randomly assigned (to get the random solution). Then, each particle  $X_i$ , which contains the dynamic elastic moduli of composite plate, can be expressed as

$$X_i = [E'_1, E'_2, G'_{12}] \quad (i = 1, 2, \dots, I) \tag{35}$$

where

$$E'_1 = [E_1^1, E_1^2, E_1^3, \dots, E_1^I]$$

$$E'_2 = [E_2^1, E_2^2, E_2^3, \dots, E_2^I]$$

$$G'_{12} = [G_{12}^1, G_{12}^2, G_{12}^3, \dots, G_{12}^I]$$

In each iterative calculation, the particle updates itself by tracking the two “extreme values” at velocity  $V_i$ . Here, the first extreme value is the best solution the particle itself can find, called the individual extremum point (representing its position with  $p_b$ ). In the global version of the PSOA [43], the other extreme point in the population is the best solution currently found for the entire population, called the global extreme point (with  $G_b$  for its position).

Then, some new points are chosen by adding  $V_i$  coordinates to  $X_i$ , and the algorithm operates by adjusting  $V_i$ , which can be seen as an effective step size. At

the meantime, the particle  $X_i$  updates its speed and position according to the following rules:

$$V_i^{k+1} = \varpi \times V_i^k + c_1^p \times rand_1^k(p_b[i] - X_i) + c_2^p \times rand_2^k(G_b[i] - X_i) \tag{36}$$

$$X_i^{k+1} = X_i^k + V_i^{k+1} \tag{37}$$

where  $c_1^p$  and  $c_2^p$  is the learning factors,  $rand()$  is a matrix of random numbers over  $[0, 1]$ , and  $\varpi$  is the inertia weight.

During the calculation process in the PSOA, each column element of  $X_i$  is randomly obtained within the range of  $E'_1, E'_2, G'_{12}$ . Firstly, to consider the effects of thermal environment with degradation time, elastic moduli in different fiber directions are assumed to be variable. Then, set the elastic moduli at room temperature as the base values (such as  $E_1^R, E_2^R, G_{12}^R$ ) and take into account the possible error  $R_{err}$  generated by the temperature change (usually  $R_{err} = 50\%$  is large enough). In this way, the range of  $E'_1, E'_2, G'_{12}$  at certain time point and temperature condition can be determined as follows

$$\begin{aligned} E_1^R (1 - R_{err}) &\leq E'_1(\Delta T, t) \leq E_1^R (1 + R_{err}) \\ E_2^R (1 - R_{err}) &\leq E'_2(\Delta T, t) \leq E_2^R (1 + R_{err}) \\ G_{12}^R (1 - R_{err}) &\leq G'_{12}(\Delta T, t) \leq G_{12}^R (1 + R_{err}) \end{aligned} \tag{38}$$

Consequently, the position of the particle  $X_i$  is used to represent the optimal solution of the elastic modulus parameters. The performance of each particle  $X_i$  depends on the fitness error value  $e_{fre}$  (usually should be set as  $e_{fre} \leq 5\%$ ), which can be determined by an error function between the theoretical and measured results described as follows.

$$e_{fre} = \sum_i^{R_m} \left( \frac{f_i - \hat{f}_i}{\hat{f}_i} \right) / R_m \tag{39}$$

where  $R_m$  represents the number of modes in the concerned frequency range,  $f_i$  represents the  $i$ th natural frequency obtained by the calculations,  $\hat{f}_i$  is the  $i$ th natural frequency obtained by the experimental tests.

If the minimum error requirement in Eq. (32) can be met, the iterative calculation process is terminated and the optimal solutions are obtained. By repeating the above steps, the optimal elastic moduli at different degradation time points and thermal environment can be obtained.

Author Proof



### 3.2 Identification of the fitting coefficients of dynamic elastic moduli

Since the concerned fitting coefficients of dynamic elastic moduli,  $A_i$ ,  $B_i$  and  $C_i$  are dependent to the temperature and time, the nonlinear least squares technique is used in the Curve Fitting Tool (CFtool) in MATLAB software to fit the three dimensional curves.

Firstly, the temperature, degradation time and elastic modulus data are inputted and written in matrix forms. Then, when these data are read by the CFtool interface, “X data” is set to represent temperature, “Y data” to degradation time, and “Z data” to dynamic elastic moduli in the drop-down menu. (But, Z data can only be one kind of dynamic elastic moduli each time.) Usually, CFtool has various function fitting tools, including Custom Equation, Interpolant, Polynomial, etc. Here, considering the downward trend in changing of dynamic elastic modulus, “Custom Equation” is employed to draw the 3D fitted surfaces of dynamic elastic moduli at different degradation time points and thermal environments, which could better show their nonlinear relationships. Also, this fitting tool can help to automatically calculate the initial values of  $A_i$ ,  $B_i$  and  $C_i$  when the 3D fitted surfaces are obtained. In addition, it should be noted that in order to determine the optimal fitting coefficients, “R-square” value in the CFtool interface should be calculated repeatedly, which is within the range of [0, 1]. The closer to 1 it is, the more accurate values of  $A_i$ ,  $B_i$  and  $C_i$  are obtained. The expression of “R-square” can be written as follows [44].

$$R_{\text{square}} = 1 - \frac{\sum_{i=1}^{N_{\text{data}}} (\hat{y}_i - \bar{y}_i)^2}{\sum_{i=1}^{N_{\text{data}}} (y_i - \bar{y}_i)^2} \quad (40)$$

where  $y_i$  is original data,  $\bar{y}_i$  is the average of  $y_i$ ,  $\hat{y}_i$  is the fitting data, and  $N_{\text{data}}$  represents the total number of  $y_i$ .

## 4 A case study

In this section, five E120 carbon fiber/ FRD-YG-03 resin composite thin plates were taken as the research subjects. Three plate specimens were used to measure the varied natural frequencies at different temperatures and degradation time points, so as to obtain the fitting

coefficients of dynamic elastic moduli in the degradation model. The other two plates with different sizes but the same material parameters were used to verify the theoretical model.

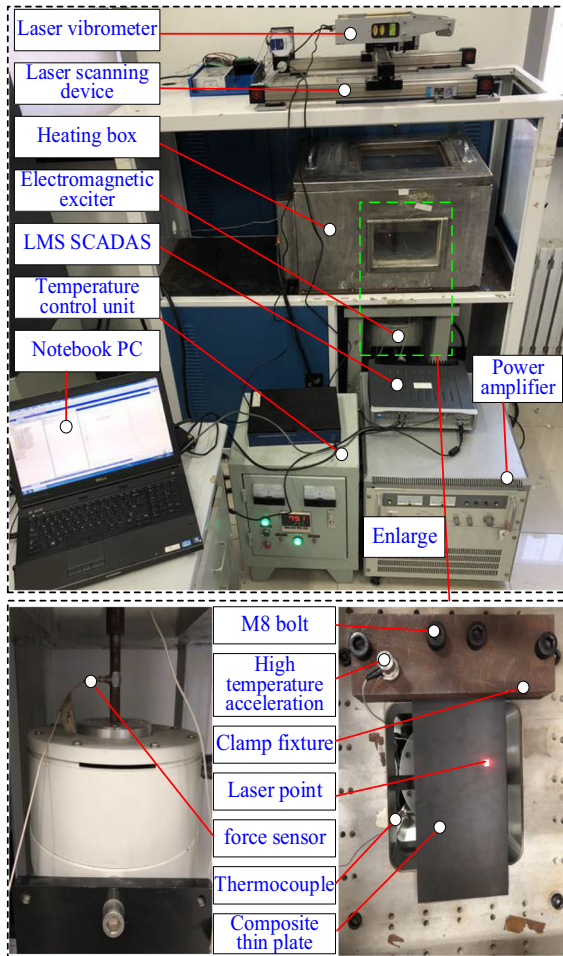
### 4.1 Test specimens and test system

In order to ensure a good test reliability, three plate specimens with the same size, namely composite *A*, *B* and *C*, were taken as test objects. They were cut from a large composite plate laminated and produced by Weisheng Xincui Composite Materials Co. Ltd. Each plate has total 15 layers with lamination configurations of  $[(0^\circ/90^\circ)_3/0^\circ/90^\circ/0^\circ/(90^\circ/0^\circ)_3]$ , which is symmetrically laid, with a longitudinal elastic modulus of 120 GPa, transverse elastic modulus of 11.32 GPa, shear modulus of 7.13 GPa, Poisson’s ratio of 0.32, a density of 1693.2 kg/m<sup>3</sup> and loss factors  $\eta_1 = 0.0063$ ,  $\eta_2 = 0.0074$ ,  $\eta_{12} = 0.0089$  in the room temperature. (Those material parameters are provided by the company.) The length is 260 mm, width is 130 mm and thickness is 2.13 mm. The thermal expansion coefficients parallel and perpendicular to the fiber direction are  $-0.15 \times 10^{-6}/^\circ\text{C}$  and  $1.1 \times 10^{-6}/^\circ\text{C}$ , respectively.

A dynamic degradation test system in thermal environment was set up, as shown in Fig. 3, to measure the varied natural frequencies, vibration responses and damping parameters in different temperatures and heating time points. In the experiment, the clamping fixture with four M8 bolts was used to clamp the plate specimen firmly (with a clamping width of 30 mm along the *x*-axis of the specimen) to simulate the cantilever boundary condition (seen in Fig. 3). The laser measuring point was 70 mm to the constraint end of the plate specimen, while the horizontal distance between this point and the right free edge was 30 mm.

The instruments and sensors used in the dynamic degradation measurements are listed in Table 1. The heating box with two thermocouple sensors was used to provide the required temperature, which can be adjusted by temperature control device. (Here, one thermocouple was used to measure the temperature within the heating box, and the other one was connected to the temperature control device for feedback purpose.)

On top of the box was an insulated glass plate through which the laser beam generated by the Polytec PDV-100 laser Doppler vibrometer can measure the vibration response on the plate specimens. The



**Fig. 3** A dynamic degradation test system of composite thin plates in thermal environment

624 motion of the laser point was powered by the two-  
 625 dimensional laser scanning device controlled by Lab-  
 626 VIEW software, which greatly improved the test effi-  
 627 ciency, especially on the modal shape measurements  
 628 in thermal condition [45]. The electromagnetic exciter  
 629 and power amplifier (which were placed outside of  
 630 the heating box) were employed to generate basic  
 631 excitation load, which can be measured with a high-  
 632 temperature accelerometer at room temperature (20 °C)  
 633 or in high-temperature environment. Besides, a force  
 634 sensor located at the middle of the rod of the elec-  
 635 tromagnetic exciter was used to determine the effec-  
 636 tiveness of exciting energy and also to avoid exces-  
 637 sive excitation energy (so that negative effects caused  
 638 by the geometric nonlinear vibration of the specimen  
 639 can be eliminated). All of the acceleration, force and

**Table 1** The instruments and sensors used in the dynamic degradation measurements

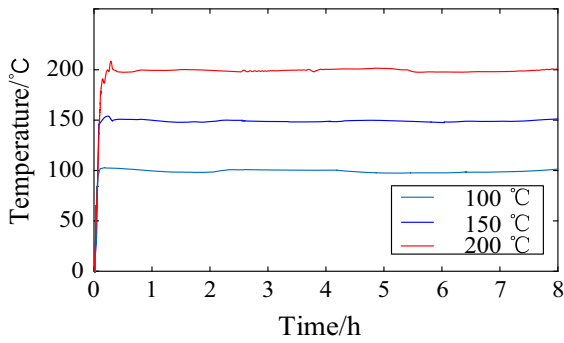
Order	Hardware type	Model
1	Laser Doppler vibrometer	Polytec PDV-100
2	Two-dimensional laser scanning device	Self-designed
3	Electromagnetic exciter	Lianneng JZK-100
4	Power amplifier	Lianneng YE5878
5	Force sensor	PCB 200B01
6	High temperature accelerometer	Lianneng CL-YD-301
7	Heating box	Changbai T-500
8	Thermocouple sensor	Yongyang PT100
9	Temperature control device	Changbai C-500
10	Modal hammer	PCB086101
11	Data acquisition instrument	LMS SCADAS 16-channel front-end
12	Mobile workstation	DELL precision M6600

640 temperature signals were recorded and stored by LMS  
 641 SCADAS 16-channel data acquisition front-end and  
 642 the notebook computer. (The output channel of LMS  
 643 SCADAS was also used to generate vibration excita-  
 644 tion signal to electromagnetic exciter.)

#### 645 4.2 Identification of fitting coefficients

646 Firstly, the composite plates *A*, *B* and *C* were installed  
 647 firmly in the clamping fixture. Based on the test system  
 648 established, sine sweeping excitation tests were per-  
 649 formed at room temperature with a frequency range of  
 650 0–550 Hz, a frequency resolution of 0.25 Hz, an exci-  
 651 tation amplitude of 1 g and a sweeping speed of 1 Hz/s.  
 652 After measuring the raw response signal, the small-  
 653 segment FFT processing technique [46] was employed  
 654 to obtain frequency spectrum of the response signal.  
 655 The natural frequencies of those specimens were deter-  
 656 mined by identifying the response peaks in these spec-  
 657 trums.

658 Next, the heating box was utilized to provide the  
 659 plate specimens mentioned above with different ther-  
 660 mal environments (i.e., plate *A* at 100 °C, plate *B* at  
 661 150 °C and plate *C* at 200 °C, respectively). Natu-  
 662 ral frequencies at different degradation time points and  
 663 temperatures were obtained by the sweeping excitation



**Fig. 4** The measured heating curve of composite plate *C* varied with the time in the heating box

technique similar as that of the room temperature test. It should be noted that each heating process was kept for more than 6 hours. Dynamic degradation experiments were performed at an interval of two hours to measure the natural frequencies. Besides, the experimental data were acquired only when the temperature was in a stabilized phase rather than a rising phase. Take the degradation test of composite plate *C* as an example. The temperature versus time curve is shown in Fig. 4. Temperature ascended at the beginning and stabilized after a certain period of time when the data recorded were valid. In this way the poor control effect in short-term heating and unwanted influence of temperature fluctuation could be excluded. It should be noted that we try to focus on the long-term degradation test of FCTP in different stabilized thermal environments. Here, the measured natural frequencies of three composite plate specimens with varied degradation time at 100 °C, 150 °C and 200 °C are listed in Tables 2, 3 and 4.

Then, according to the PSOA described in Sect. 3.1, set the following parameters in the self-written MATLAB program: (1) particle population number: 20; (2) learning factor  $c_1 = c_2 = 2$ ; (3) the maximum number of iterations: 1000; (4) inertia weight  $\omega = 0.9$ . In the iterative calculation process, if the error calculation formula [Eq. (39)] was satisfied, the execution of the program would be terminated, and the optimal particle  $X_i$ , which contained the dynamic elastic moduli would be outputted. In order to compare the theoretical predictions with the test results, the iteratively calculated natural frequencies of three composite plate specimens at different degradation time points and thermal environments (100 °C, 150 °C and 200 °C) are listed in Tables 2, 3 and 4. It can be seen that the calcu-

lation errors of natural frequencies are no more than 9.2%. Therefore, the corresponding dynamic elastic moduli of fiber-reinforced composite can be extracted in CFtool in MATLAB software. Table 5 lists the identified results of  $E'_1$ ,  $E'_2$ ,  $G'_{12}$  at different degradation time points and temperatures. (The identified results at room temperature are also provided, which are very close to the ones the manufacturer provided in Sect. 4.1.)

Then, the result data in Table 5 are used to draw the 3D fitted surfaces of dynamic elastic moduli at different degradation time points and thermal environments, as shown in Fig. 5. Through analyzing the degradation phenomenon from Fig. 5, it can be observed that the dramatic degradation change of elastic moduli of the chosen composite plate specimens in thermal environment occurs in the first 2 hours. The reason may be that there is a big change of temperature value in this stage (the heating process), so the macroscopic softening effect on composite plate is obvious, which subsequently leads to the dramatical reduction of elastic moduli. Then, as the plate structure is already within a constant thermal environment (the temperature value is keeping at a fixed level), in the following 4 h, 6 h and 8 h, the degradation performance of dynamic elastic moduli is developed relatively slowly. It should be noted due to the time-consuming and high-cost of experimental tests, the further degradation test (lasting one day or several days) were not conducted. However, based on the analysis conclusion in literatures [35, 47], the development of degradation behavior would continue for a long time.

Finally, by substituting the identified fitting coefficients (as listed in Table 6) into Eqs. (1)–(3), the expressions of dynamic elastic moduli in thermal environment with degradation time can be obtained

$$\begin{aligned}
 E'_1(\Delta T, t) &= 120.612 - 1.806 \times \Delta T^{0.6651} \times e^{-\frac{0.3338}{t}} \\
 E'_2(\Delta T, t) &= 10.882 - 0.001864 \times \Delta T^{1.432} \times e^{-\frac{0.5421}{t}} \\
 G'_{12}(\Delta T, t) &= 6.785 - 0.00147 \times \Delta T^{1.476} \times e^{-\frac{0.2996}{t}}
 \end{aligned}$$

### 4.3 Comparison and verification of the inherent characteristics

In this section, in order to verify the correctness of the theoretical model, the composites plate *D* and plate *E* were used for calculation and test. Their length, width and thickness were 330 mm × 130 mm × 2.36 mm.

**Table 2** Natural frequencies of composite plate *A* obtained by tests and iterative calculations with varied degradation time at 100 °C

Type	Degradation time/h	Mode 1	Mode 2	Mode 3	Mode 4
Experiment $\hat{f}_i$ /Hz	0 (room temperature)	39.5	97.7	238.9	328.3
	2	37.3	94.8	227.5	310.8
	4	37.0	93.5	227.4	308.1
	6	36.8	92.8	224.2	305.5
	8	36.8	92.3	224.0	302.5
Calculation $f_i$ /Hz	0 (room temperature)	41.3	99.4	253.7	339.4
	2	37.4	98.3	241.4	322.6
	4	37.1	97.2	238.5	318.1
	6	36.8	96.9	235.7	314.0
	8	36.5	96.6	232.3	310.4
Error $ \hat{f}_i - f_i /\hat{f}_i/\%$	0 (room temperature)	4.6	1.7	6.2	3.4
	2	0.3	3.7	6.1	3.8
	4	0.3	4.0	4.9	3.2
	6	0	4.4	5.1	2.8
	8	0.8	4.7	3.7	2.6

**Table 3** Natural frequencies of composite plate *B* obtained by tests and iterative calculations with varied degradation time at 150 °C

Type	Degradation time/h	Mode 1	Mode 2	Mode 3	Mode 4
Experiment $\hat{f}_i$ /Hz	0 (room temperature)	37.2	96.8	228.0	322.1
	2	35.0	93.3	211.5	307.0
	4	34.5	92.8	210.0	305.5
	6	33.3	91.7	209.8	302.0
	8	31.5	91.3	208.3	300.2
Calculation $f_i$ /Hz	0 (room temperature)	40.5	98.1	233.7	334.4
	2	35.3	97.8	221.6	316.9
	4	35.0	96.8	214.0	309.3
	6	34.6	96.5	211.6	305.8
	8	34.4	96.3	209.8	302.5
Error $ \hat{f}_i - f_i /\hat{f}_i/\%$	0 (room temperature)	8.9	1.3	2.5	3.8
	2	0.9	4.8	4.8	3.2
	4	1.4	4.3	1.9	1.2
	6	3.9	5.2	0.9	1.3
	8	9.2	5.5	0.7	0.8

742 After the plate specimen was firmly clamped, the effective  
 743 dimensions of length, width and thickness were  
 744 300 mm × 130 mm × 2.36 mm. The laser measuring  
 745 point was placed at the same position as that of composite  
 746 plates *A*, *B* and *C*.

747 In the test, the heating temperature was chosen as  
 748 120 °C for plate *D* and 180 °C for plate *E*. Four degradation  
 749 time points of 0 h, 2 h, 4 h and 6 h were selected.

750 The similar sine excitation method and signal data  
 751 processing and identification techniques described in  
 752 Sect. 4.2 were utilized to measure the first three natural  
 753 frequencies of the two plate specimens, as listed  
 754 in Tables 7 and 8. In addition, the laser linear scanning  
 755 method [47] was used to measure modal shapes. Taking  
 756 the measurement of composite plate *D* as an example,

**Table 4** Natural frequencies of composite plate *C* obtained by tests and iterative calculations with varied degradation time at 200 °C

Type	Degradation time/h	Mode 1	Mode 2	Mode 3	Mode 4
Experiment $\hat{f}_i$ /Hz	0 (room temperature)	36.8	94.5	223.9	312.5
	2	32.6	91.6	211.5	300.3
	4	30.8	89.3	208.5	299.7
	6	29.8	88.4	208.0	297.3
	8	29.3	87.5	207.5	296.0
Calculation $f_i$ /Hz	0 (room temperature)	39.8	96.1	228.5	329.4
	2	33.7	92.9	216.5	310.1
	4	32.0	91.1	212.3	307.7
	6	31.6	90.4	210.9	305.3
	8	31.3	89.8	209.2	301.6
Error $ \hat{f}_i - f_i /\hat{f}_i/\%$	0 (room temperature)	8.2	1.7	2.1	5.4
	2	3.4	1.4	2.4	3.3
	4	3.9	2.0	1.8	2.7
	6	6.0	2.3	1.4	2.7
	8	6.8	2.6	0.8	1.9

**Table 5** The identified dynamic elastic moduli of E120 carbon fiber/FRD-YG-03 resin composite at different degradation time points and thermal environments

Temperature/°C	Type	Degradation time/h			
		2	4	6	8
Room temperature	$E'_1$ /Gpa	120.612	120.612	120.612	120.612
	$E'_2$ /Gpa	10.882	10.882	10.882	10.882
	$G'_{12}$ /Gpa	6.785	6.785	6.785	6.785
100	$E'_1$ /Gpa	92.875	90.514	87.979	85.106
	$E'_2$ /Gpa	9.783	9.632	9.587	9.490
	$G'_{12}$ /Gpa	5.571	5.395	5.232	5.140
150	$E'_1$ /Gpa	72.110	70.351	67.905	66.529
	$E'_2$ /Gpa	8.987	8.794	8.612	8.523
	$G'_{12}$ /Gpa	5.020	4.904	4.830	4.762
200	$E'_1$ /Gpa	71.021	69.212	66.702	64.830
	$E'_2$ /Gpa	7.912	7.599	7.406	7.246
	$G'_{12}$ /Gpa	3.415	3.382	3.207	3.110

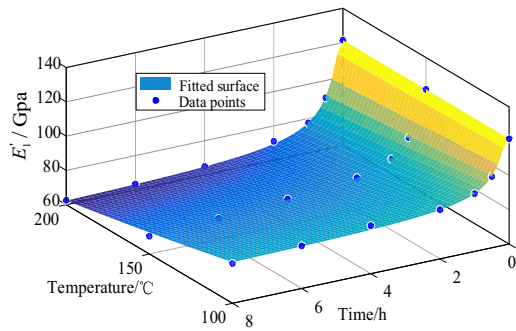
757 the first three modal shapes with varied degradation  
758 time at 120 °C were obtained, as shown in Table 9.

759 Then, set the same heating temperatures and degra-  
760 dation time points to calculate the natural frequencies  
761 based on the theoretical model established. The corre-  
762 sponding calculation results and errors of plate *D* and  
763 plate *E* are listed in Tables 7 and 8. For the conven-  
764 ience to compare, the calculated natural frequencies  
765 and the resulting errors without considering the degra-  
766 dation effect are also listed in the same tables. Besides,

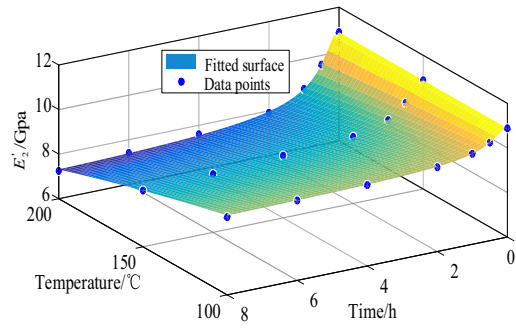
767 Table 9 shows the calculated modal shape results of  
768 composite plate *D*. Based on the work carried out by  
769 Jeyaraj et al. [8] and Li et al. [48], the modal shapes of  
770 composite plates seem to be immune to temperatures;  
771 hence, there is no need to discuss the degradation effect  
772 on modal shapes.

773 It can be seen from Tables 7, 8 and 9 that there  
774 is a good agreement between the calculated and mea-  
775 sured natural frequencies at different degradation time  
776 points in those thermal environments, which is evi-

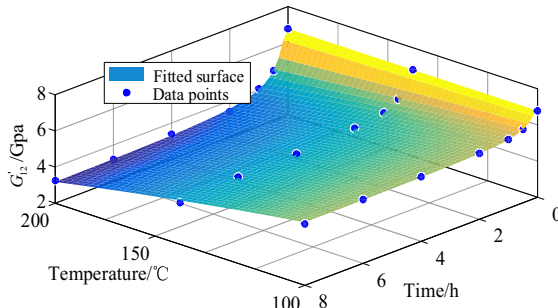




(a)  $E'_1$



(b)  $E'_2$



(c)  $G'_{12}$

**Fig. 5** The 3D fitted surfaces of dynamic elastic moduli at different degradation time points and thermal environments

777 denced by the calculation errors of the first 3 nat-  
 778 ural frequencies of FCTP at 120 °C and 180 °C being  
 779 less than 9.6% with considering the degradation effect.  
 780 The morphological characteristics of the calculated and  
 781 measured modal shapes also agree well. However, the  
 782 natural frequencies calculated without considering the  
 783 degradation effect have a larger error. What's more, the  
 784 longer the degradation time is, the greater the error. For  
 785 example, when the temperature is raised to 180 °C, the  
 786 maximum calculation error of the first 3 natural fre-  
 787 quencies at the degradation time point of 6 h reaches

**Table 6** The identified fitting coefficients of fiber-reinforced composites

Type	Fitting coefficient	Value
$E'_1$	$A_1$	1.806
	$B_1$	0.6651
	$C_1$	0.3338
$E'_2$	$A_2$	0.001864
	$B_2$	1.432
	$C_2$	0.5421
$G'_{12}$	$A_{12}$	0.00147
	$B_{12}$	1.476
	$C_{12}$	0.2996

to 21.1%, while the maximum error at the degrada-  
 788 tion time point of 2 h is 16.5%. Therefore, it is nec-  
 789 essary to introduce the material nonlinearity in the  
 790 modeling process, and the degradation model proposed  
 791 improves the calculation accuracy of natural frequen-  
 792 cies of FCTP.  
 793

Furthermore, by comparing the measured natural  
 794 frequencies of composite thin plate structures at differ-  
 795 ent thermal environments, it can be found out that the  
 796 higher the temperature is, the more severe degrada-  
 797 tion phenomenon of FCTP is induced, especially when the  
 798 environmental temperature is raised from room temper-  
 799 ature to a higher temperature through the first heating  
 800 process for 2 hours. By taking the first natural frequency  
 801 as an example, the reduced magnitude at degrada-  
 802 tion time point of 2 h at 120 °C is 17.3%, while the cor-  
 803 responding reduced magnitude reaches to 28.1% when  
 804 the temperature becomes to 180 °C. The reason for this  
 805 phenomenon may be due to a big reduction in structural  
 806 stiffness of FCTP caused by the sharp variation of ther-  
 807 mal environment. However, with the degradation time  
 808 lasting, as the thermal environment gets increasingly  
 809 stabilized, the composite plate structure shows a slowed  
 810 downward trend in changing of natural frequency, i.e.,  
 811 the stiffness softening effect due to degradation effect  
 812 becomes smaller and smaller.  
 813

#### 4.4 Comparison and verification of the dynamic response

Here, the first three resonant responses at different  
 814 degradation time points at 120 °C and 180 °C were  
 815 measured when the base excitation amplitude of 1g  
 816  
 817  
 818

**Table 7** The measured and calculated natural frequencies of composite plate *D* with varied degradation time at 120 °C as well as the calculation errors with and without considering degradation behavior

Time/h	Type	Mode 1	Mode 2	Mode 3
0 (Room temperature)	Experiment $\hat{f}_i$ /Hz	24.8	73.6	155.0
	Calculation $f_i$ /Hz	25.9	75.2	162.4
	Error $ \hat{f}_i - f_i /\hat{f}_i/\%$	4.4	2.2	4.8
2	Experiment $\hat{f}_i$ /Hz	20.5	67.2	129.2
	Calculation with considering degradation $f_i$ /Hz	22.1	70.7	135.7
	Calculation without considering degradation $\bar{f}_i$ /Hz	23.6	74.3	138.6
	Error with considering degradation $ \hat{f}_i - f_i /\hat{f}_i/\%$	7.8	5.2	5.0
	Error without considering degradation $ \hat{f}_i - \bar{f}_i /\hat{f}_i/\%$	15.1	10.6	7.3
4	Experiment $\hat{f}_i$ /Hz	20.3	67.0	125.7
	Calculation with considering degradation $f_i$ /Hz	21.3	70.6	133.5
	Calculation without considering degradation $\bar{f}_i$ /Hz	23.6	74.3	138.6
	Error with considering degradation $ \hat{f}_i - f_i /\hat{f}_i/\%$	4.9	5.4	6.2
	Error without considering degradation $ \hat{f}_i - \bar{f}_i /\hat{f}_i/\%$	16.3	10.9	10.3
6	Experiment $\hat{f}_i$ /Hz	19.3	66.7	121.6
	Calculation with considering degradation $f_i$ /Hz	20.9	70.5	132.7
	Calculation without considering degradation $\bar{f}_i$ /Hz	23.6	74.3	138.6
	Error with considering degradation $ \hat{f}_i - f_i /\hat{f}_i/\%$	8.3	5.7	9.1
	Error without considering degradation $ \hat{f}_i - \bar{f}_i /\hat{f}_i/\%$	22.3	11.4	14.0

819 was applied to the plate specimens. The theoretical  
 820 resonant responses were also calculated with the same  
 821 excitation amplitude used in the experiment. For the  
 822 convenience of comparison, Fig. 6 presents the cal-  
 823 culated and measured resonant responses of compos-  
 824 ite plate *D* at 120 °C, and Fig. 7 presents the cor-  
 825 responding resonant responses of composite plate *E*  
 826 at 180 °C. Besides, the maximum calculation error in  
 827 each mode of FCTP is listed in the same figures as  
 828 well.

829 It can be seen from Figs. 6 and 7 that the cal-  
 830 culation errors of resonant responses by considering  
 831 the degradation effect at different temperatures are no  
 832 more than 9.9%, which further verifies the degrada-  
 833 tion model. Besides, the amplitudes of the first three  
 834 resonant responses of FCTP at different degradation  
 835 time points gradually increase compared to the ones at  
 836 room temperature. Meanwhile, the higher the temper-  
 837 ature is, the more intense the resonance in each mode  
 838 of FCTP. Taking the experimental results at degrada-

839 tion time point of 2 h as an example, the first resonant  
 840 response increases from 0.0687 to 0.0797 m/s when the  
 841 temperature rises from 120 to 180 °C. The reason for  
 842 this intense vibration phenomenon is the stiffness soft-  
 843 ening behavior induced by thermal degradation effect,  
 844 which becomes more obvious when the temperature  
 845 rises.

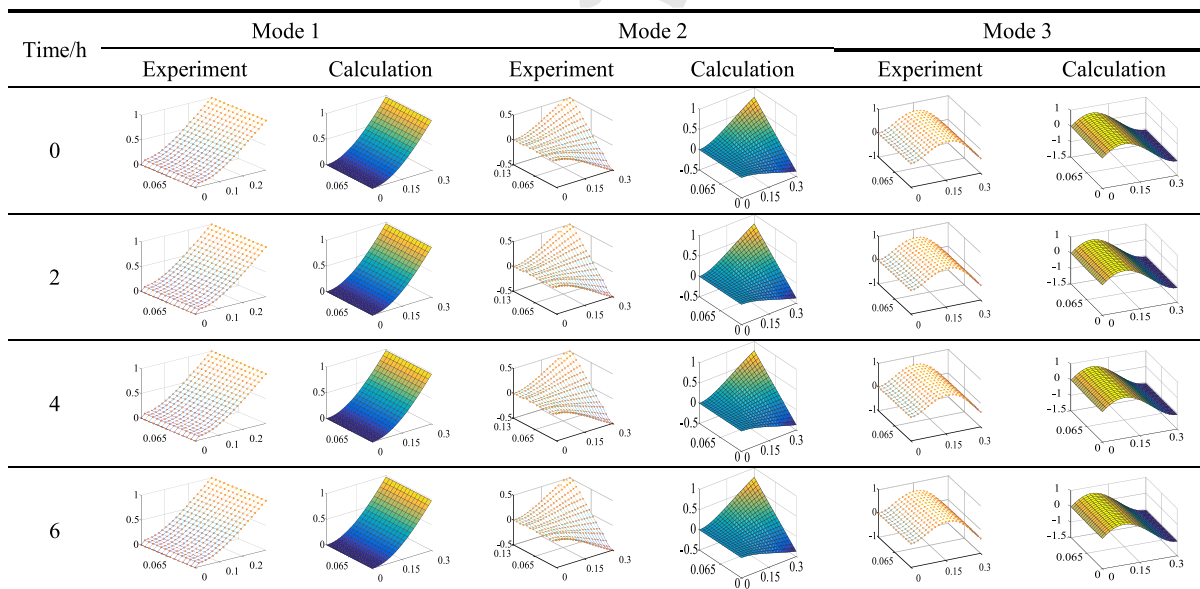
846 However, the increasing trend of resonant responses  
 847 of FCTP gradually becomes mild as the degradation  
 848 time increases, i.e., the longer the degradation time is,  
 849 the smaller influence it has on the dynamic responses.  
 850 This is also reflected by the reduction of slopes in res-  
 851 onant response curves for different modes. The reason  
 852 may be that although the softening stiffness led by  
 853 degradation effect aggravating the vibration of FCTP,  
 854 the structural damping still plays an important role. As  
 855 the degradation time goes on, the effect of the increased  
 856 damping property becomes non-negligible. Therefore,  
 857 the resonant responses of FCTP show a weakened  
 858 upward trend.

Author Proof

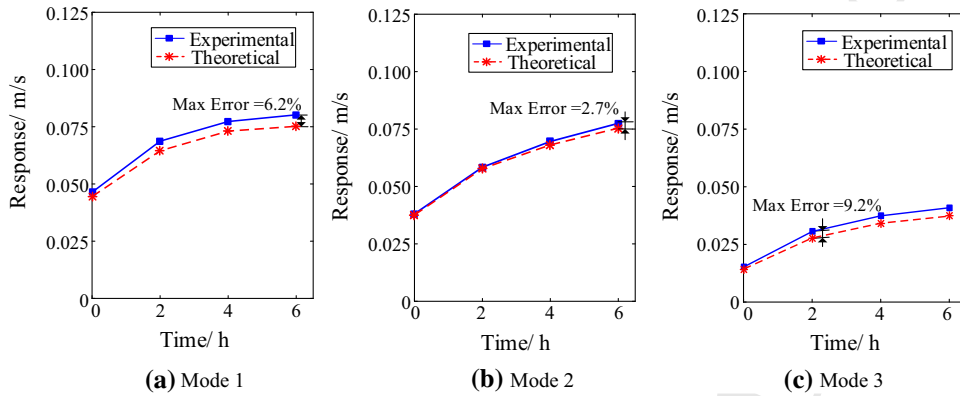
**Table 8** The measured and calculated natural frequencies of composite plate *E* with varied degradation time at 180 °C as well as the calculation errors with and without considering degradation behavior

Time/h	Type	Mode 1	Mode 2	Mode 3
0 (Room temperature)	Experiment $\hat{f}_i$ /Hz	25.3	74.0	158.3
	Calculation $f_i$ /Hz	25.9	75.2	162.4
	Error $ \hat{f}_i - f_i /\hat{f}_i/\%$	2.4	1.6	2.6
2	Experiment $\hat{f}_i$ /Hz	18.2	65.3	119.3
	Calculation with considering degradation $f_i$ /Hz	19.7	69.4	126.4
	Calculation without considering degradation $\bar{f}_i$ /Hz	21.2	72.6	131.3
	Error with considering degradation $ \hat{f}_i - f_i /\hat{f}_i/\%$	8.2	6.3	6.0
	Error without considering degradation $ \hat{f}_i - \bar{f}_i /\hat{f}_i/\%$	16.5	11.2	10.1
4	Experiment $\hat{f}_i$ /Hz	17.7	65.0	118.7
	Calculation with considering degradation $f_i$ /Hz	19.4	68.8	123.1
	Calculation without considering degradation $\bar{f}_i$ /Hz	21.2	72.6	131.3
	Error with considering degradation $ \hat{f}_i - f_i /\hat{f}_i/\%$	9.6	5.8	3.7
	Error without considering degradation $ \hat{f}_i - \bar{f}_i /\hat{f}_i/\%$	19.8	11.7	10.6
6	Experiment $\hat{f}_i$ /Hz	17.5	65.0	117.3
	Calculation with considering degradation $f_i$ /Hz	19.1	68.5	122.0
	Calculation without considering degradation $\bar{f}_i$ /Hz	21.2	72.6	131.3
	Error with considering degradation $ \hat{f}_i - f_i /\hat{f}_i/\%$	9.1	5.4	4.0
	Error without considering degradation $ \hat{f}_i - \bar{f}_i /\hat{f}_i/\%$	21.1	11.7	11.9

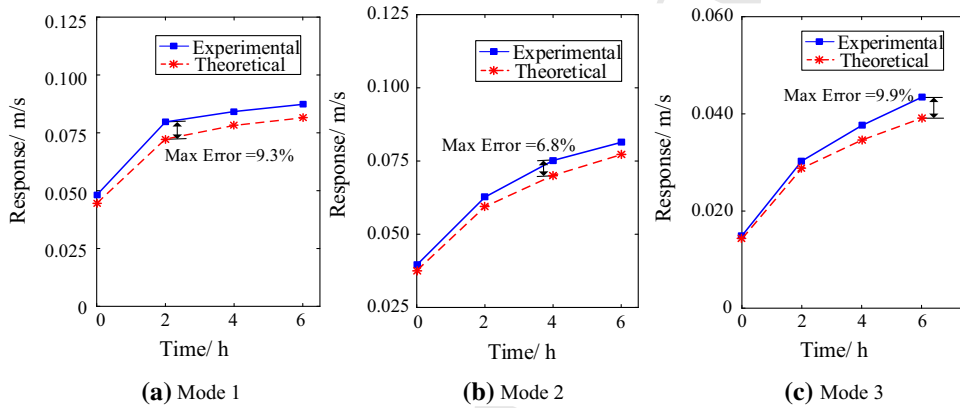
**Table 9** The measured and calculated modal shapes of the composite plate *D* with varied degradation time at 120 °C



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**Fig. 6** The first three resonant responses of composite plate *D* obtained by theoretical calculations and experimental tests with varied degradation time at 120 °C



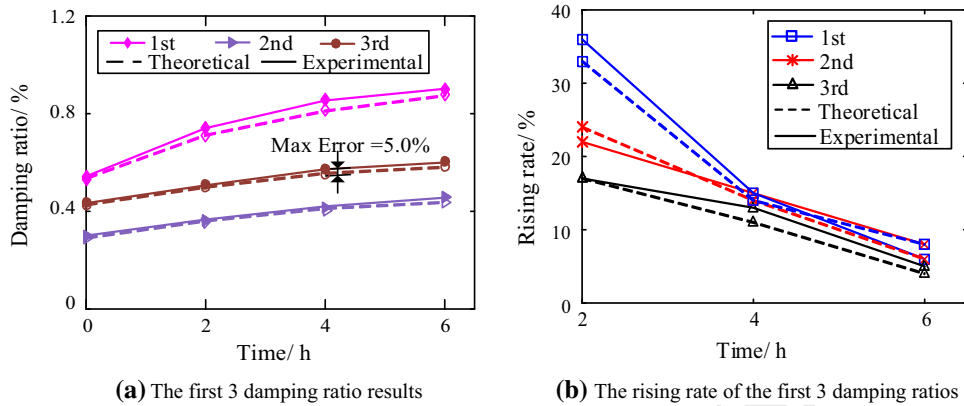
**Fig. 7** The first three resonant responses of composite plate *E* obtained by theoretical calculations and experimental tests with varied degradation time at 180 °C

4.5 Comparison and verification of the damping behavior

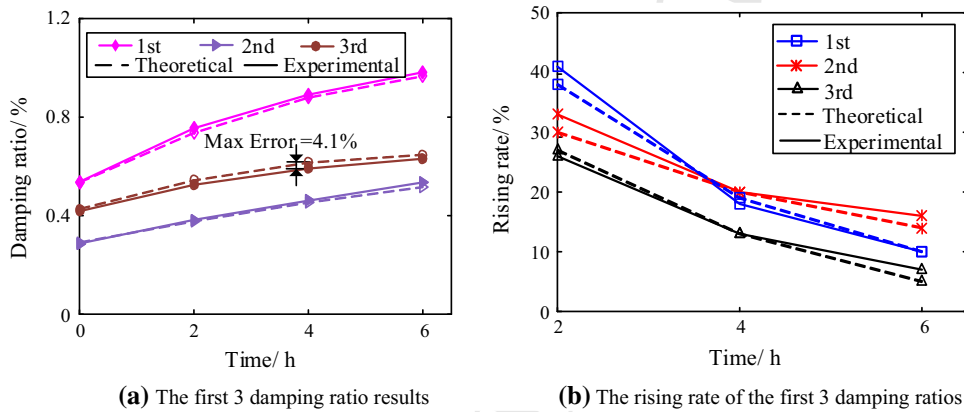
In this section, the first three modal damping ratios of composite plates *D* and *E* with varied degradation time at 120 °C and 180 °C were obtained by identifying the frequency domain data in a sine sweeping excitation test, as shown in Figs. 8a and 9a. For the convenience of comparison, the theoretical damping results of composite plates *D* and plate *E* were also obtained, as seen in Figs. 8a and 9a, where the related maximum calculation errors of the first 3 modes of FCTP are also indicated in the same figures. Besides, in order to visualize the effect of degradation time on damping property, the measured and calculated rising rates for the first three modal damping ratios of composite plate *D* and plate *E* at 120 °C and 180 °C are plotted in Figs. 8b and 9b.

It can be seen from Figs. 8a and 9a that the calculation errors of the first three modal damping ratios with considering the degradation effect at different temperatures are less than 5.0%, which further verifies the degradation model. Besides, the damping properties of FCTP at different degradation time points in thermal environment gradually increase compared to the ones at room temperature. Meanwhile, the higher the temperature is, the larger each modal damping ratio is. Taking the experimental results at degradation time point of 2 h as an example, when the temperature rises from 120 to 180 °C, the first modal damping ratio increases from 0.741 to 0.755%. The reason may be the increased energy dissipation capacity of interfacial friction in the composite plate induced by thermal degradation effect, which becomes more severe when the temperature rises.

Author Proof



**Fig. 8** The first three damping ratios and the corresponding rising rate of composite plate *D* obtained by theoretical calculations and experimental tests with varied degradation time at 120 °C



**Fig. 9** The first three damping ratios and the corresponding rising rate of composite plate *E* obtained by theoretical calculations and experimental tests with varied degradation time at 180 °C

892 However, from Figs. 8b and 9b, it can be found  
 893 out that the uptrend of damping properties of FCTP  
 894 declines gradually as the degradation time increases.  
 895 By taking the experimental damping ratios of compos-  
 896 ite plate *D* at 120 °C as an example, when the environ-  
 897 mental temperature is raised from room temperature to  
 898 a higher temperature through the first heating process  
 899 for 2 hours, the rising rates of the first three damp-  
 900 ing ratios reach to 36.2, 21.9 and 16.8%. While at the  
 901 degradation time point of 4h, those values are reduced  
 902 to 15.2, 14.7 and 13.0%, and finally at the degradation  
 903 time point of 6h, are further declined to 5.6, 8.6 and  
 904 4.7%. The reason for this may be that as the degrada-  
 905 tion time continues, the energy dissipation capacity of  
 906 interfacial friction in the composite plate is gradually  
 907 weakened, i.e., there is less and less thermal energy  
 908 being able to be converted into the friction and dissi-

909 pation energy during the degradation process. In addi-  
 910 tion, from a chemical reaction point of view, it can be  
 911 explained by the fact that the polymer matrix of fiber-  
 912 reinforced composites is getting harder and harder due  
 913 to the thermal oxidation effect [49].

## 914 5 Conclusions

915 In this research, a novel dynamic degradation model of  
 916 FCTP subjected to thermal environment with degrada-  
 917 tion time has been established and verified. Also, the  
 918 nonlinear dynamic degradation characteristics of FCTP  
 919 are investigated. It can be discovered that:

920 Due to the stiffness softening behavior and increased  
 921 energy dissipation capacity of interfacial friction of  
 922 fiber-reinforced composites, the higher the temperature



is, the more natural frequencies of FCTP decrease, and the more on the dynamic responses and damping properties increase.

However, due to the complicated effects of thermal degradation on stiffness and material properties, as the degradation time continues, the composite thin plate structures show a slowed downward trend in changing of natural frequencies, while an upward trend in resonant responses and damping behavior becomes less and less steep.

In addition, due to that the actual thermal environment applied on FCTP is quite complicated, which not only covers the constant temperature, but also the heating and cooling stages. Therefore, the future work should consider those above factors, better to establish a more comprehensive model to describe the degradation behavior of FCTP in a quick-change thermal condition.

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**Compliance with ethical standards**

**Conflict of interest** The authors declared no potential conflicts of interest with respect to the research, authorship, and/or publication of this article.

**Appendix**

$$\begin{aligned} \bar{Q}_{11} &= \frac{E_1^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} \cos^4 \theta_k + 2 \frac{\mu_{12}E_1^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} \sin^2 \theta_k \cos^2 \theta_k \\ &+ 4G_{12}^*(\Delta T, t) \sin^2 \theta_k \cos^2 \theta_k + \frac{E_2^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} \sin^4 \theta_k \\ \bar{Q}_{12} &= \left( \frac{E_1^*(\Delta T, t) + E_2^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} - 4G_{12}^*(\Delta T, t) \right) \sin^2 \theta_k \cos^2 \theta_k \\ &+ \frac{\mu_{12}E_1^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} (\sin^4 \theta_k + \cos^4 \theta_k) \\ \bar{Q}_{22} &= \frac{E_1^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} \sin^4 \theta_k + 2 \frac{\mu_{12}E_1^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} \sin^2 \theta_k \cos^2 \theta_k \\ &+ 4G_{12}^*(\Delta T, t) \sin^2 \theta_k \cos^2 \theta_k + \frac{E_2^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} \cos^4 \theta_k \\ \bar{Q}_{16} &= \left( \frac{(1 - \mu_{12})E_1^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} - 2G_{12}^*(\Delta T, t) \right) \sin \theta_k \cos^3 \theta_k \\ &+ \frac{\mu_{12}E_1^*(\Delta T, t) - E_2^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} \sin^3 \theta_k \cos \theta_k \end{aligned}$$

$$\begin{aligned} &+ 2G_{12}^*(\Delta T, t) \sin^3 \theta_k \cos \theta_k \\ \bar{Q}_{26} &= \left( \frac{(1 - \mu_{12})E_1^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} - 2G_{12}^*(\Delta T, t) \right) \sin^3 \theta_k \cos \theta_k \\ &+ \frac{\mu_{12}E_1^*(\Delta T, t) - E_2^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} \sin \theta_k \cos^3 \theta_k \\ &+ 2G_{12}^*(\Delta T, t) \sin \theta_k \cos^3 \theta_k \\ \bar{Q}_{66} &= \frac{(1 - 2\mu_{12})E_1^*(\Delta T, t) + E_2^*(\Delta T, t)}{1 - \mu_{12}\mu_{21}} \sin^2 \theta_k \cos^2 \theta_k \\ &- 2G_{12}^*(\Delta T, t) \sin^2 \theta_k \cos^2 \theta_k \\ &+ G_{12}^*(\Delta T, t) (\sin^4 \theta_k + \cos^4 \theta_k) \end{aligned}$$

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