

1 **Plasma-catalytic synthesis of ammonia over Ru/BaTiO₃-based**
2 **bimetallic catalysts: Synergistic effect from dual-metal active**
3 **sites**

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17

18 **Abstract**

19 Plasma-catalytic synthesis of ammonia (NH₃) was carried out using BaTiO₃
20 supported Ru-M bimetallic catalysts (Ru-M/BaTiO₃, M = Fe, Co and Ni) in a
21 dielectric barrier discharge (DBD) reactor. The NH₃ synthesis performance followed
22 the order of Ru-Ni/BaTiO₃ > Ru/BaTiO₃ > Ru-Co/BaTiO₃ > Ru-Fe/BaTiO₃, with the
23 highest NH₃ concentration (3895 ppm) and energy yield (0.39 g kWh⁻¹) achieved over

24 Ru-Ni/BaTiO₃ at 25 W and 10 W, respectively. To gain insights into the
25 physio-chemical properties of the Ru-M/BaTiO₃ catalysts, comprehensive catalyst
26 characterizations were performed, including X-ray diffraction, N₂ physisorption
27 measurements, X-ray photoelectron spectroscopy (XPS), high-resolution transmission
28 electron microscopy (HRTEM), energy dispersive spectroscopy (EDS), and
29 temperature-programmed desorption of CO₂ and N₂ (CO₂ and N₂-TPD). The results
30 indicated that the loading of Ni enhanced the basicity and N₂ adsorption capacity of
31 the catalyst surface, as well as the density of oxygen vacancy (OV) on the BaTiO₃
32 surface, which facilitated the adsorption and activation of N₂ on BaTiO₃ surface.
33 These effects led to the enhanced NH₃ synthesis, as excited N₂ could be adsorbed on
34 Ru-Ni/BaTiO₃ from plasma region and stepwise hydrogenated to form NH_x species
35 and ultimately NH₃.

36 **Keywords:** plasma catalysis, ammonia synthesis, Ru-based bimetallic catalyst,
37 oxygen vacancy, N₂ activation

38

39 **1. Introduction**

40 Besides being an essential raw material for many chemicals, ammonia (NH₃) has
41 shown great value and environmental benefits as an emerging clean fuel due to its
42 characteristics of carbon-free, high energy density and easy to liquid and store [1].
43 NH₃ can be directly used in fuel cells to generate electricity, and blended with
44 conventional fossil fuels such as gasoline and diesel for combustion to output
45 mechanical power [2, 3]. Using a NH₃ blend-fuel power system can achieve higher
46 thermal efficiency and lower NO_x, SO_x, and CO₂ emissions than a fossil fuel-powered

47 internal combustion engine [4, 5].

48 Currently, the Haber-Bosch process based NH_3 synthesis is the major
49 industrial-scale NH_3 synthesis process. This process is an energy-intensive thermal
50 catalytic reaction that requires the reactants to be maintained within a temperature
51 range of 450–600°C and a pressure range of 10–25 MPa for NH_3 synthesis. The
52 Haber-Bosch process based NH_3 synthesis relies almost entirely on the combustion of
53 fossil fuels to provide the required energy. Due to the harsh reaction conditions, the
54 production of 1 kg of NH_3 results in emissions of 1.25–2.16 kg of CO_2 equivalent due
55 to the harsh reaction conditions [6]. Therefore, the NH_3 synthesis industry consumes
56 approximately 1.8–3% of global energy each year and emits approximately 290
57 million tons of CO_2 [7, 8].

58 Establishing small-scale, decentralized NH_3 production plants to utilize localized
59 and intermittent renewable energy sources is currently the main way to produce
60 "green ammonia". Among various technologies for the synthesis of "green ammonia",
61 e.g. photo-catalysis [9, 10] and electro-catalysis [11, 12], the plasma-catalysis process
62 has received widespread attention. The plasma-catalytic system could be operated
63 under mild catalytic reaction conditions, and is easy to achieve miniaturization and
64 decentralization of NH_3 synthesis equipment. The plasma-catalytic system is
65 characterized by quick start and stop times, which allows it to match intermittent and
66 seasonal renewable energy sources. This enables the conversion and storage of
67 localized renewable energy sources. Furthermore, in the process of plasma-catalytic
68 synthesis of NH_3 , the activation of the $\text{N}\equiv\text{N}$ bond is mainly achieved via collisions

69 with energetic electrons, making NH_3 synthesis thermodynamically favorable even at
70 ambient pressure and room temperature [13, 14].

71 Perovskite oxides (with the general formula ABO_3) have recently emerged as a
72 new and promising class of catalyst support in heterogeneous catalysis due to their
73 exceptional thermal stability, tunable electronic structure, and redox behavior [15].
74 Due to the unique structure of perovskite oxides, which can accommodate variable
75 B-site elements and valences, these catalyst supports are highly susceptible to
76 introducing interface effects, such as strong metal-support interaction (SMSI) and
77 oxygen vacancies (OV). This, in turn, enhances the activity of ABO_3 -supported
78 catalysts for NH_3 synthesis [16, 17]. Moreover, our previous research has
79 demonstrated that the introduction of BaTiO_3 , with its high dielectric constant, into
80 the plasma discharge region could amplify the electric field of the gas phase, thereby
81 facilitating N_2 activation through electron-impact reactions, which represents the
82 primary step in plasma-catalytic NH_3 synthesis [18].

83 The incorporation of metallic catalysts has been demonstrated to enhance the
84 plasma-catalytic performance of NH_3 synthesis, as the metal sites on the catalyst
85 support play a critical role in heterogeneous reactions [19-21]. Various metal-based
86 catalysts have proved to be active for NH_3 synthesis to date. Among them, the
87 Ru-based catalysts are more efficient under milder reaction conditions as they afford
88 more prominent N_2 activation due to the proper binding strength of atomic N and N_2
89 dissociation barrier [22, 23]. However, the utilization of Ru-based catalysts has been
90 limited due to the high cost and low production of Ru. To reduce cost and optimize

91 the performance of Ru-based metallic catalysts, alloying Ru with other metal
92 components has been found effective in adjusting its structural and electronic
93 properties, thereby increasing its ability to activate the $\text{N}\equiv\text{N}$ bond. This approach
94 builds upon knowledge gained from conventional thermal catalytic NH_3 synthesis.
95 For example, previous studies have demonstrated that an alloy of Ru-Fe/MgO resulted
96 in a higher NH_3 yield and better anti-poisoning performance compared to pure Ru and
97 Fe catalysts [24]. Similarly, the Ru-Co/MgAl₂O₄ catalyst was found to be the most
98 active among the Ru-M/MgAl₂O₄ catalysts (M = Fe, Co, Ni, and Mo). When the Ru
99 loading was reduced from 4 wt.% to 2 wt.%, and Co was added at the same 2 wt.%
100 loading in the Ru-Co/MgAl₂O₄ catalyst, its activity was comparable to that of the 4
101 wt.% Ru/MgAl₂O₄ catalyst, suggesting the underlying synergistic effect between
102 bimetallic components [25]. Some studies suggested that the improved performance
103 of Ru-based bimetallic catalysts may be attributed to different surface activities
104 resulting from electron transfer between the metals, as well as favorable kinetics and
105 thermodynamics of N_2 dissociation [24, 26]. Despite these efforts to investigate the
106 mechanisms underlying for thermal-catalytic NH_3 synthesis, the understanding of the
107 reaction mechanisms of plasma-catalytic NH_3 synthesis over Ru-based bimetallic
108 catalysts remains limited.

109 In this study, we investigate the performance of plasma-catalytic NH_3 synthesis
110 over bimetallic Ru-M/BaTiO₃ (M = Fe, Co, and Ni) catalysts in a dielectric barrier
111 discharge (DBD) reactor. To gain insight into the changes in the physicochemical
112 properties of the catalysts after incorporating M, we extensively characterized

113 Ru-M/BaTiO₃ catalysts using various techniques such as X-ray diffraction (XRD), N₂
114 physisorption measurements, X-ray photoelectron spectroscopy (XPS),
115 high-resolution transmission electron microscopy (HRTEM), energy dispersive
116 spectroscopy (EDS), and temperature programmed desorption (CO₂-TPD and
117 N₂-TPD). In addition, we used optical emission spectroscopy (OES) to identify the
118 reactive species during the plasma-catalytic reactions. Based on the results obtained,
119 we propose the mechanisms underlying the enhanced performance of Ru-Ni/BaTiO₃
120 catalysts in plasma-catalytic NH₃ synthesis.

121 **2. Experimental**

122 **2.1 Catalysts preparation and characterization**

123 Bimetallic catalysts of Ru-M/BaTiO₃ (M = Fe, Co and Ni) were prepared using
124 the co-impregnation method. Initially, the nitrates of M and RuCl₃·3H₂O were
125 dissolved in deionized water, then the solution was added to a measured amount of
126 BaTiO₃ powder. The slurry was stirred for 30 min, then heated at 80°C for 3 h, dried
127 at 110°C for 12 h, and calcined at 550°C for 6 h. The catalysts were reduced in a 5
128 vol.% H₂/Ar flow at 550°C and 100 ml min⁻¹ for 3 h, then screened to 40-60 mesh.
129 The total loading amount of active metal for each catalyst was fixed at 3 wt.%, while
130 the molar ratio of Ru to M was 1:1. The catalysts were named Ru-Fe/BaTiO₃,
131 Ru-Co/BaTiO₃, Ru-Ni/BaTiO₃ and Ru/BaTiO₃. The chemicals Ni(NO₃)₂·6H₂O,
132 Co(NO₃)₂·6H₂O, Fe(NO₃)₃·9H₂O, RuCl₃·3H₂O and BaTiO₃ were all of analytical
133 grade (Macklin, China). The purity of all gases (H₂, N₂, and Ar) was 99.999%
134 (Fangxin, China).

135 The XRD patterns of the Ru-M/BaTiO₃ catalysts were obtained using a Rikagu
136 D/max-2000 X-ray diffractometer equipped with Cu-K α radiation. The scanning
137 range was 10° to 80°, and the scanning step size was 0.02°. N₂ physisorption
138 measurements were performed using a Quantachrome Autosorb-1 instrument at 77 K.
139 Before each test, the catalysts were degassed at 250 °C for 3 h to remove impurities.
140 The specific surface area was obtained using the Brunauer-Emmett-Teller (BET)
141 method, and pore volume and diameter distribution were obtained using Barrett
142 Joyner Halenda (BJH) method. High-resolution transmission electron microscopy
143 (HRTEM) and energy-dispersive X-ray spectrometry (EDS) mapping analyses were
144 performed using the FEI Tecnai G2 F20. The particle size distribution of each catalyst
145 was obtained by analyzing 100 particles from the TEM images.

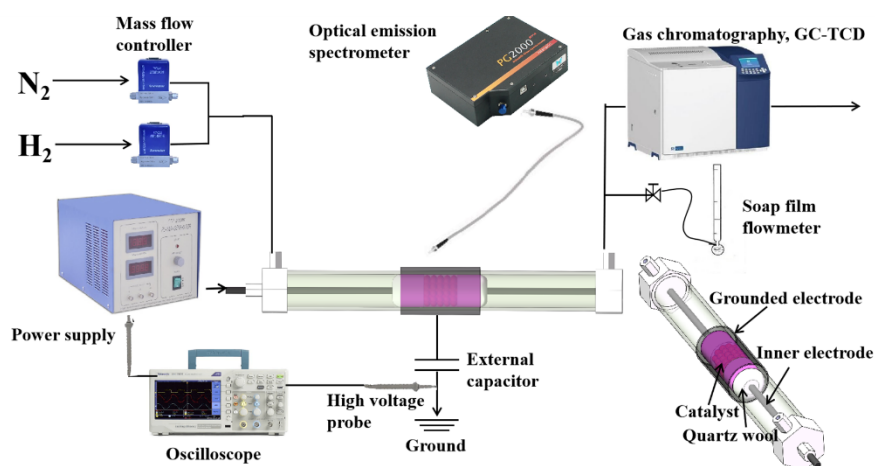
146 Temperature programmed desorption (TPD) experiments were conducted on an
147 AutoChem II 2920 instrument to measure the desorption behavior of CO₂ and N₂.
148 Prior to the test, catalysts were reduced in a H₂/He gas flow (10 vol.% H₂; 50 mL
149 min⁻¹) at 550°C for 2 h and then cooled to 50°C. For CO₂-TPD, the catalysts were
150 treated in a CO₂ gas flow at 50°C and 50 ml min⁻¹ for 30 min to adsorb CO₂, while for
151 N₂-TPD, the catalysts were placed in a N₂ gas flow at 500°C and 50 ml min⁻¹ for 1 h,
152 then they were cooled to 50°C and kept for 30 min to adsorb N₂. The samples were
153 then heated to 900°C at the rate of 10 °C min⁻¹ in a He flow at the flow rate of 40 ml
154 min⁻¹ for both CO₂-TPD and N₂-TPD.

155 X-ray photoelectron spectroscopy (XPS) analysis was performed on a Thermo
156 ESCALAB 250Xi using an Al K α X-ray source. Both low-resolution survey and

157 high-resolution scans were obtained for each catalyst, and the spectra were calibrated
158 with C 1s at a binding energy of 284.8 eV. Prior to the XPS analysis, the catalysts
159 were reduced in H₂/He gas flow (10 vol.% H₂; 50 ml min⁻¹) at 550°C for 2 h and then
160 cooled to ambient temperature. After reduction, the samples were prepared and
161 analyzed in a reduced environment.

162

163 2.2 Experimental setup



164

165 **Fig. 1.** Experimental setup for plasma-catalytic NH₃ synthesis

166

167 A schematic diagram of the experimental system used in this study is shown in
168 Fig. 1. The reactor comprised a quartz tube with an inner diameter of 8 mm and a wall
169 thickness of 2 mm, sealed with two PTFE seals. The high-voltage electrode was a 4
170 mm diameter stainless-steel rod connected to a power source (Suman CTP-2000K,
171 China), while the external electrode was a stainless-steel mesh wrapped around the
172 quartz tube. The Ru-M/BaTiO₃ catalyst (0.4 g, 40–60 meshes) was held in the reactor
173 by quartz wool. High voltage probes (Tektronix TPP0502, USA) were used to

174 measure the voltages across the external capacitor (0.47 μF) and the reactor, and a
175 four-channel digital oscilloscope (Tektronix TDS2024C, USA) was used to record all
176 electrical signals.

177 N_2 and H_2 gases with a purity exceeding 99.999% were sourced from gas
178 cylinders and regulated by mass flow controllers (Sevenstars D07-B, China). The
179 gases were mixed in a chamber before being introduced to the reactor in a 1:1 molar
180 ratio. To measure the concentrations of gas products (including N_2 , H_2 , and NH_3), a
181 two-channel gas chromatography (Fuli 9790II, China) equipped with a thermal
182 conductivity detector (TCD) was used. Optical emission spectra during plasma
183 discharge in the presence of the Ru-M/BaTiO₃ catalysts were captured using an
184 optical spectrometer (Ideaoptics PG2000EX, China).

185 **2.3 Electrical characteristic measurement and parameters calculation.**

186 The plasma discharge power was calculated by the Lissajous method as follows:

$$P_{\text{dis}} (\text{W}) = f \times C_{\text{ext}} \times A \quad (1)$$

187 where C_{ext} is external capacitance (0.47 μF), f is the frequency of AC discharge (9.4 to
188 9.8 kHz), and A is the area of the Lissajous figures (Fig. S1).

189 The energy yield of the plasma-catalytic NH_3 synthesis was calculated using the
190 following equation:

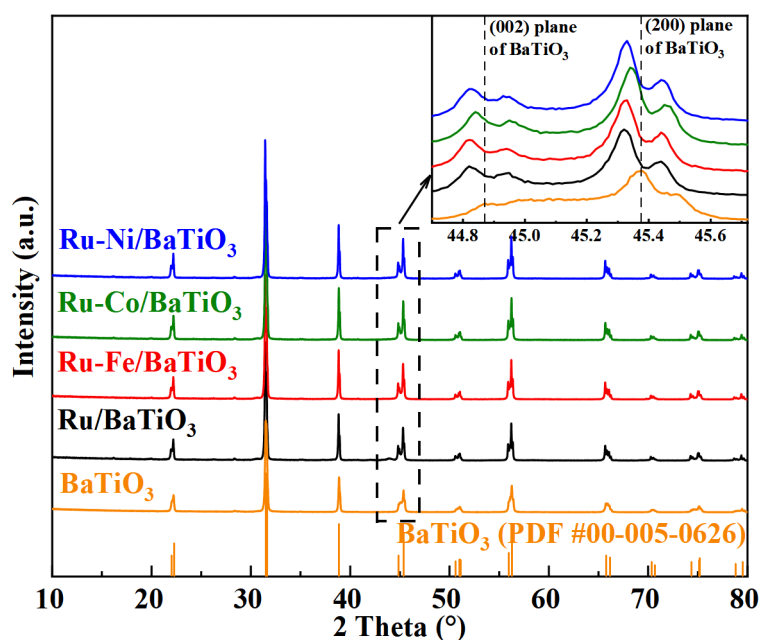
$$191 \quad \text{Energy yield} (\text{g kWh}^{-1}) = \frac{M \times C_{\text{out}} \times Q_{\text{after}}}{P_{\text{dis}}} \quad (2)$$

192 where M denotes the molar mass of NH_3 , C_{out} is the NH_3 concentration measured at
193 the reactor outlet and Q_{after} is the gas flow rate after the reaction measured by a
194 soap-film flowmeter.

195

196 3. Results and discussion

197 3.1. Catalyst characterizations



198

199 **Fig. 2.** XRD patterns for the Ru-M/BaTiO₃ catalysts

200

201 Fig. 2 depicts the XRD patterns of the Ru-M/BaTiO₃ catalysts used in the study.

202 All the catalysts showed well-matched diffraction peaks with the tetragonal phase

203 BaTiO₃ (ICDD PDF #00-005-0626). The diffraction peaks at 2θ of 31.49°, 31.64°,

204 38.88°, 44.85°, 45.37°, 55.95°, and 56.25°, which can be attributed to the (1 0 1), (1 1

205 0), (1 1 1), (0 0 2), (2 0 0), (1 1 2), and (2 1 1) planes of BaTiO₃ in the tetragonal

206 phase, were intense for all the Ru-M/BaTiO₃ catalysts. No diffraction peaks of

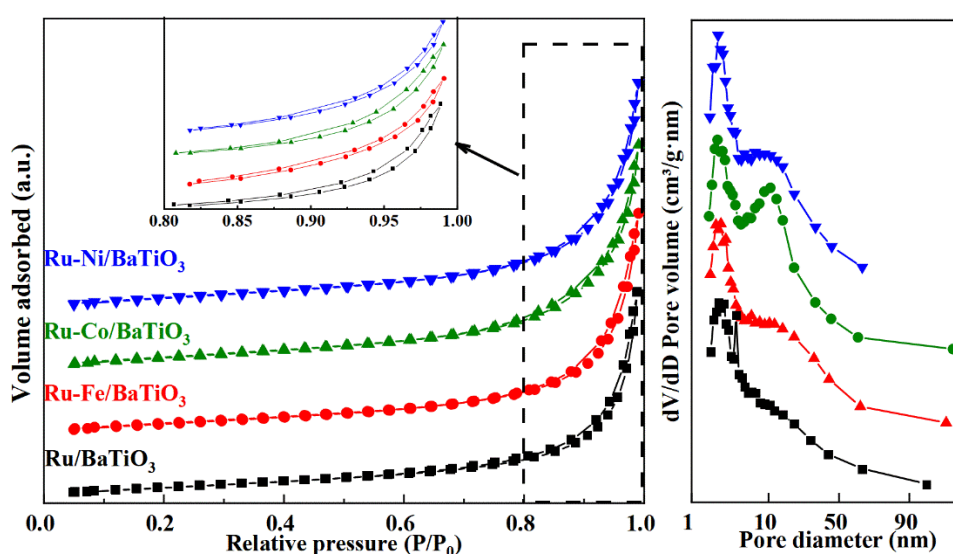
207 metallic Ru, Fe, Co, and Ni were observed in the XRD patterns, probably due to the

208 high dispersion of the metal dopants on the BaTiO₃ support. The presence of the

209 dopant M caused the diffraction peaks of (0 0 2) and (2 0 0) planes to shift to lower

210 angles when compared to BaTiO₃, which could be associated with the lattice

211 expansion of BaTiO₃. This phenomenon might be attributed to the substitution of Ti⁴⁺
 212 in BaTiO₃ by M cation with a larger radius (which is c.a. 0.61, 0.65, 0.61, 0.69 and
 213 0.68 Å for Ti⁴⁺, Fe³⁺, Co²⁺, Ni²⁺ and Ru³⁺, respectively [27-30]), and oxygen vacancy
 214 was generated for electronic compensation, as the dopant of M cation preferentially
 215 substitutes for Ti⁴⁺ ions in BaTiO₃ due to the similarity of their ionic radii in
 216 octahedral coordination [31-33].



217
 218 **Fig. 3.** N₂ adsorption-desorption isotherms and pore diameter distribution of the
 219 Ru-M/BaTiO₃ catalysts
 220

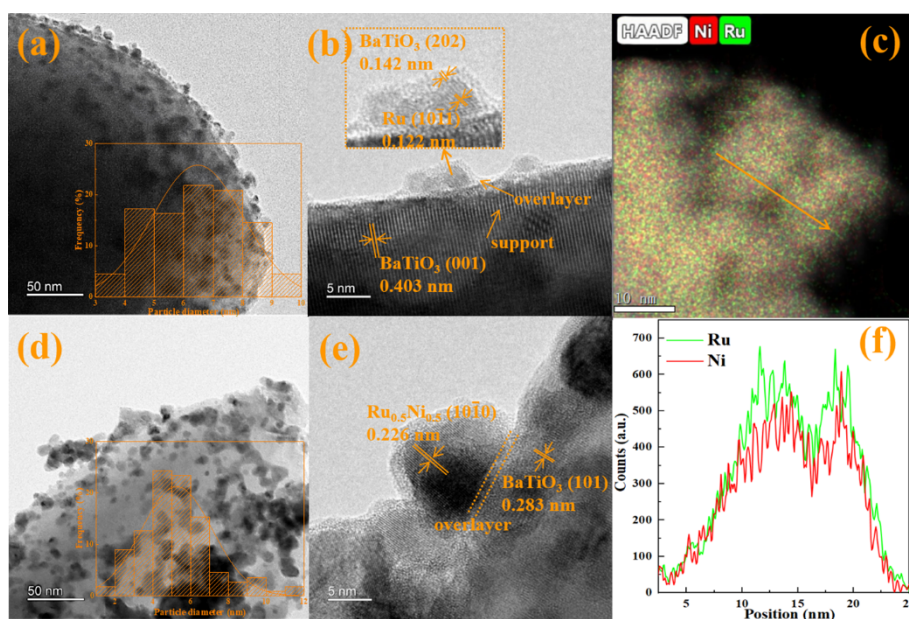
221 As shown in Fig. 3, all Ru-M/BaTiO₃ samples exhibited clear characteristics of
 222 type IV isotherms with H4 hysteresis loops, indicating their mesoporous structure.
 223 Compared to Ru-M/BaTiO₃, the introduction of dopant M resulted in a decrease in the
 224 most probable aperture and average pore size, while the specific surface area slightly
 225 increased. For instance, the most probable aperture was 2.66 nm for Ru/BaTiO₃,
 226 which reduced to 2.26 nm for Ru-Ni/BaTiO₃ and 2.20 nm for Ru-Co/BaTiO₃.
 227 Meanwhile, the specific surface area increased from 2.26 m² g⁻¹ for Ru/BaTiO₃ to
 228 2.62 m² g⁻¹ for Ru-Ni/BaTiO₃. Considering the structural differences among the

229 Ru-M/BaTiO₃ catalysts were not significant, it suggests that the loading of M had a
 230 limited impact on adjusting the framework structure of catalysts.

231 **Table 1.** Textural properties of the Ru-M/BaTiO₃ catalysts

Catalyst	S _{BET} (m ² g ⁻¹)	Total pore volume (cm ³ g ⁻¹)	Average pore size (nm)	The most probable aperture (nm)
Ru-Fe/BaTiO ₃	2.42	0.010	16.05	2.37
Ru-Co/BaTiO ₃	2.57	0.011	15.05	2.20
Ru-Ni/BaTiO ₃	2.62	0.011	15.86	2.26
Ru/BaTiO ₃	2.26	0.010	17.18	2.66

232



233

234 **Fig. 4.** HRTEM for (a and b) Ru/BaTiO₃, (d and e) for Ru-Ni/BaTiO₃; (c) EDS
 235 mapping of Ru-Ni particles on Ru-Ni/BaTiO₃; (f) line scanning of Ru-Ni particles

236

237 High-resolution transmission electron microscopy (HRTEM) was employed to
 238 analyze the catalysts' morphology. As depicted in Fig. 4(a) and (d), the doped metal
 239 particles were predominantly present on the support's surface in a highly disordered

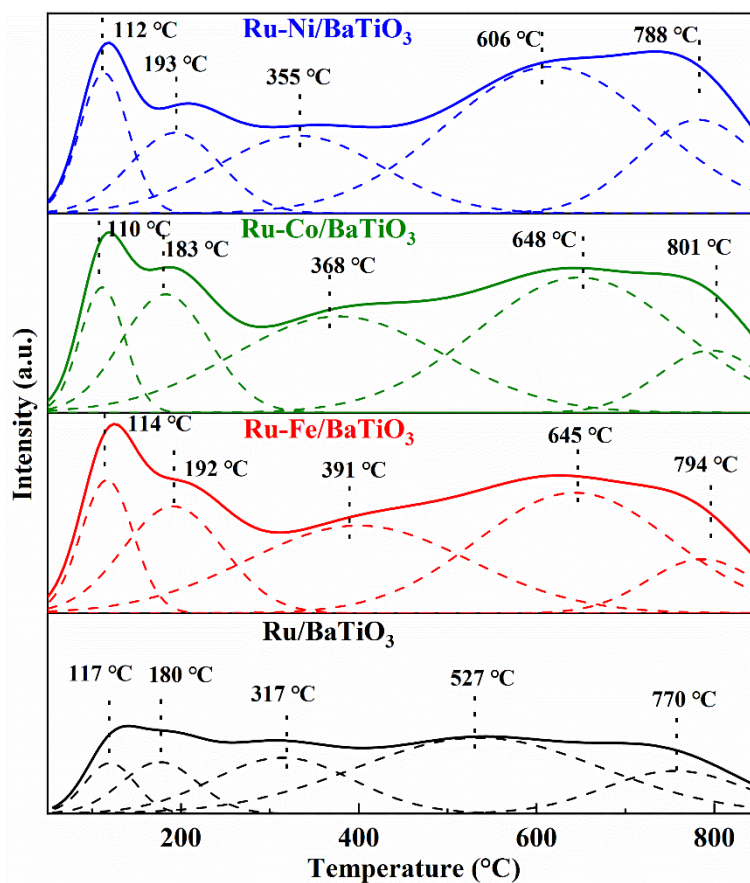
240 state. By using EDS-mapping, the distribution of Ru and Ni elements on the surface
241 of the BaTiO₃ support was investigated, and the results were shown in Fig. S2. The
242 uniform and random distribution of Ru and Ni elements on the surface of the BaTiO₃
243 support was confirmed by comparing spatial features in both the Ru and Ni maps,
244 which is consistent with Fig. 4(d). In comparison to Ru/BaTiO₃, the average size of
245 the metal particle in Ru-Ni/BaTiO₃ is 3.12 nm, which is much smaller than that in
246 Ru/BaTiO₃ (5.14 nm). The metal particle size of Ru-Ni/BaTiO₃ mainly ranged from
247 4–6 nm, whereas it was 6–8 nm for Ru/BaTiO₃. This observation is consistent with
248 the study by Ni et al., who reported that adding Mg reduced the particle size of
249 Ru-Mg particles and improved the dispersion of Ru nanoparticles on the surface of
250 Ru-Mg/AC due to the interaction of metal precursors during the high-temperature
251 reduction process [34].

252 Figs. 4(b) and (e) show the d-spacings of 0.206 and 0.226 nm correspond to the
253 (1 0 $\bar{1}$ 1) plane of Ru (ICDD PDF #00-006-0663) and (1 0 $\bar{1}$ 0) plane of Ru-Ni alloy
254 (ICDD PDF #04-001-2915) in the hexagonal phase, respectively. The structure of
255 Ru-Ni binary particle was further investigated by EDS. As presented in Fig. 4(c), the
256 location color point presenting Ru and Ni elements were uniformly mixed and
257 covered. The line scanning results showing in Fig. 4(e) suggest the intensities of Ru
258 and Ni exhibited the same spatial trend of variation, which confirmed that Ru-Ni
259 existed in the form of a homogeneous alloy.

260 Fig. 4(b) reveals that the Ru particles were doped onto and encapsulated by the
261 BaTiO₃ support, which can be attributed to the SMSI effect [35]. The SMSI effect

262 could alter the electronic properties of active metal species by electron transfer from
263 the support to metal particles that are doped onto the support, and further vary the
264 adsorption strength of reactants on the metal particles [36]. However, in the case of
265 Ru/BaTiO₃, the full encapsulation of the Ru particles had a detrimental effect on their
266 catalytic activity due to the decrease in metal sites. Fig. 4(e) also confirms the
267 occurrence of the SMSI effect in the case of Ru-Ni/BaTiO₃ since an overlayer is
268 found between the Ru-Ni metal particle and the BaTiO₃ support.

269



270

271 **Fig. 5.** CO₂-TPD profiles of the Ru-M/BaTiO₃ catalysts

272

273 The basicity of Ru-M/BaTiO₃ catalysts was evaluated using CO₂-TPD, as

274 presented in Fig. 5. The desorption peaks centered in the low temperature region (50–

275 300°C) correspond to physical adsorption and the adsorption on Brønsted basicity
276 sites of the Ru-M/BaTiO₃ catalysts. Meanwhile, the desorption peaks centered in the
277 medium temperature (300–600°C) and high temperature (> 600°C) regions
278 correspond to the desorption of CO₂ on medium and strong basic sites, respectively,
279 indicating the desorption of CO₂ from Lewis basicity sites [37, 38]. Compared to
280 Ru/BaTiO₃, the introduction of dopant M into the Ru-M/BaTiO₃ catalysts improved
281 the density of both weak and strong basic sites. The increase in the density of weak
282 basic sites in the presence of dopant M is believed to be related to the
283 surface-adsorbed hydroxyl groups[38], while the medium and strong basicity are
284 associated with the chemisorption on oxygen species such as electron-rich O²⁻, O₂²⁻
285 and O⁻ species [39], which are related to CO₂ adsorption on surface OV sites [40]. To
286 better quantify the strength of basicity in the Ru-M/BaTiO₃ catalysts, the CO₂-TPD
287 profiles were deconvoluted and the amount of CO₂ desorbed was listed in Table S1.
288 The results showed that the amount of CO₂ desorbed from medium and strong basic
289 sites of the Ru-M/BaTiO₃ catalysts were 54.2 μmol g⁻¹ for Ru-Ni/BaTiO₃, 50.5 μmol
290 g⁻¹ for Ru-Co/BaTiO₃, 48.2 μmol g⁻¹ for Ru-Fe/BaTiO₃ and 31.9 μmol g⁻¹ for
291 Ru/BaTiO₃, respectively. These findings indicate that there was an increased density
292 of Lewis basic sites on the surface of the Ru-M/BaTiO₃ catalysts, with Ru-Ni/BaTiO₃
293 exhibiting the highest density of Lewis basic sites.

294

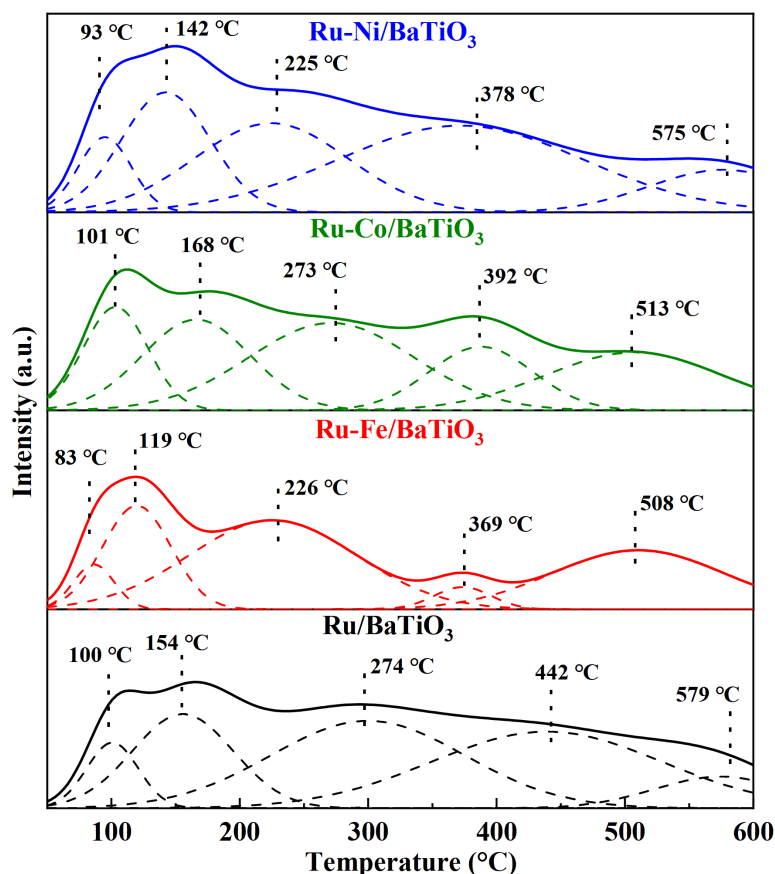


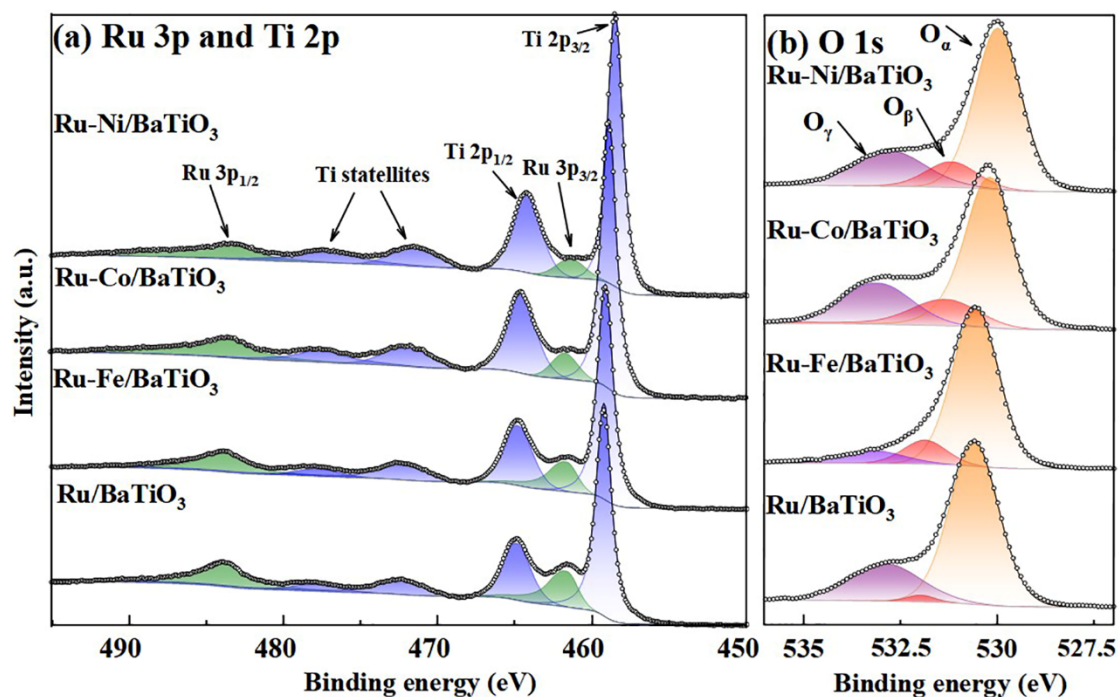
Fig. 6. N₂-TPD profiles for the Ru-M/BaTiO₃ catalysts

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296
297

298 As presented in Fig. 6, the N₂-TPD desorption profiles for the Ru-M/BaTiO₃
 299 catalysts can be deconvoluted into 5 peaks. The first two peaks, centered below 200°C,
 300 correspond to the desorption of physically adsorbed N₂, while the peaks centered
 301 above 200°C can be attributed to the desorption of chemisorbed N₂ [41]. In
 302 comparison to Ru/BaTiO₃, the desorption profile of Ru-Ni/BaTiO₃ catalyst showed a
 303 similar pattern with a significantly higher N₂ desorption amount. However, the
 304 introduction of Fe or Co to the catalyst resulted in the inhibition of both physical and
 305 chemical adsorption of N₂. The amount of N₂ chemisorption, which was determined
 306 by N₂ desorption amount above 200°C, follows the order of Ru-Ni/BaTiO₃ (53.2
 307 μmol g⁻¹) > Ru/BaTiO₃ (48.8 μmol g⁻¹) > Ru-Co/BaTiO₃ (37.9 μmol g⁻¹) >
 308 Ru-Fe/BaTiO₃ (24.1 μmol g⁻¹). Such chemisorption of N₂ is necessary for N₂

309 activation, but the diffusion of some key intermediates, such as N and NH_x radicals
310 that bind too strongly to the metal surface, may be impeded [42].

311



312

313 **Fig. 7.** High-resolution XPS spectra for Ru-M/BaTiO₃ catalysts: (a) Ru 3p and Ti 2p;
314 (b) O 1s

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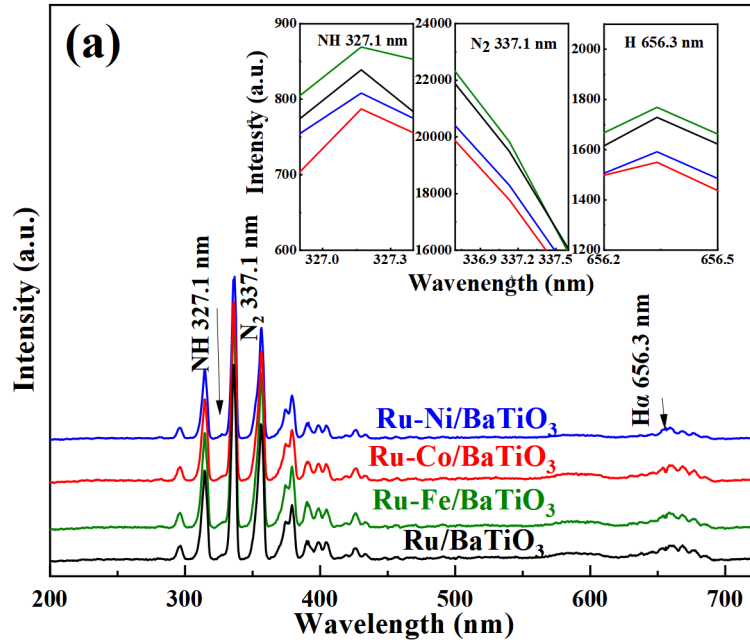
316 Fig. 7(a) presents high-resolution XPS spectra of the Ru 3p and Ti 2p orbits.
317 These spectra were deconvoluted using the Lorentzian-Gaussian function. Generally,
318 the binding energies (BE) of Ru 3p_{1/2} and Ru 3p_{3/2} were centered around 483.7 and
319 461.7 eV, respectively. For Ru/BaTiO₃, the BE of Ru 3p_{1/2} was 483.92 eV, which was
320 lower than the standard BE of Ru 3p_{1/2} at 484.8 eV, indicating that the Ru particles on
321 the BaTiO₃ surface were in an electron-rich state [43]. The BE of Ru 3p_{1/2} for the
322 Ru-M/BaTiO₃ catalysts further decreased compared to the Ru/BaTiO₃ catalyst, for
323 example, the BE of Ru 3p_{1/2} for Ru-Fe/BaTiO₃ was 483.88 eV, while it shifted to
324 483.33 eV for Ru-Ni/BaTiO₃.

325 Regarding the Ti 2p spectra of the Ru-M/BaTiO₃ catalysts, two main peaks from
326 the Ti 2p_{3/2} and Ti 2p_{1/2} orbits at around 459 eV and 464 eV, respectively, were clearly
327 observed along with two satellite peaks. The BE of Ti 2p_{3/2} for Ru/BaTiO₃ was 459.28
328 eV, and it shifted to 458.96 eV and 458.45 eV for Ru-Fe/BaTiO₃ and Ru-Ni/BaTiO₃,
329 respectively. The shift of characteristic peaks in the Ti 2p_{3/2} and Ti 2p_{1/2} orbits can be
330 attribute to the generation of Ti³⁺ and are related to the generation of OV [44, 45].

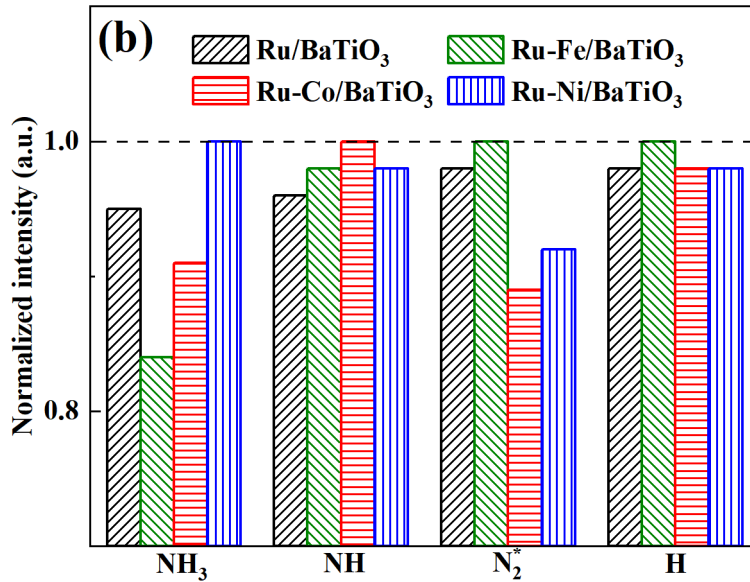
331 Fig. 7(b) presents the high-resolution XPS spectra of the O 1s orbit for the
332 Ru-M/BaTiO₃ samples. The peaks centered around 530.58 eV belong to lattice
333 oxygen species (O_α), which mainly exists in the form of Ba-O-Ti [46]. The peaks
334 centered around 531.90 eV correspond to O_β, which is the oxygen species in the
335 vicinity of the OV, and the peaks centered around 533.37 eV could be attributed to
336 surface adsorbed oxygen species (O_γ) like OH⁻ or O₂²⁻, CO₃²⁻ [33, 47]. The introduction
337 of dopant M led to the left shift of the BE of O_α and an increase in the area of O_β
338 peaks, suggesting an increased density of OV on catalyst surface [48, 49]. The relative
339 OV density of the Ru-M/BaTiO₃ catalysts, defined as O_β/(O_α + O_β + O_γ), was as
340 follows: Ru-Ni/BaTiO₃ (15.35%) > Ru-Fe/BaTiO₃ (14.18%) > Ru-Co/BaTiO₃
341 (7.90%) > Ru/BaTiO₃ (1.73%). This order is in agreement with the order of the
342 binding energy of Ti 2p.

343

344 3.2. Optical emission spectrum diagnosis



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Fig. 8. (a) Optical emission spectra for plasma discharge over Ru-M/BaTiO₃ (plasma discharge power is 15 W and exposure time is 400 ms); (b) normalized relative intensity of reactive species

Fig. 8(a) shows the N₂ second positive system (SPS) indicating the vibrational excitation reaction of N₂. The H_α atomic line at 656.3 nm indicates the dissociation of H₂. The strong intensity of N₂ at 337 nm caused the NH band at 336.1 nm to overlap and become indistinguishable. Therefore, the NH band at 327.1 nm was chosen to determine the relative intensity of the NH bands. The selection of emission lines and

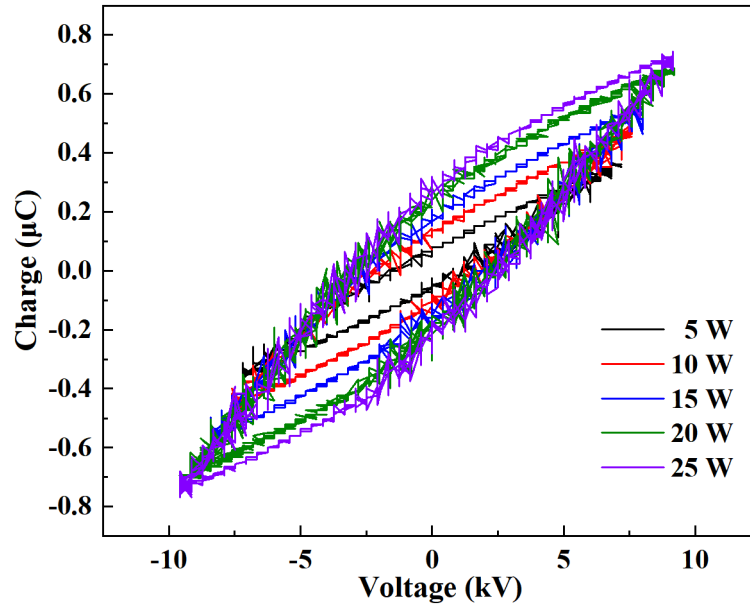
356 corresponding transitions are listed in Table S2. The intensities of the selected
357 emission lines of these reactive species were normalized using the method proposed
358 by Wang et al. [19], and the results are presented in Fig. 8(b).

359 As shown in Fig. 8(b), compared with Ru/BaTiO₃ and Ru-Fe/BaTiO₃, the
360 relative intensity of N₂^{*} emission line was lower in the presence of Ru-Ni/BaTiO₃ and
361 Ru-Co/BaTiO₃. Considering that the generation of N₂^{*} is mainly derived from
362 electron-impact reactions in gas phase, which are a function of electron energy and
363 electric field strength. The lower relative intensity of N₂^{*} suggests a lower degree of
364 vibrational-excitation of N₂, indicating that the electric field was relatively lower for
365 the case of Ru-Ni/BaTiO₃ and Ru-Co/BaTiO₃. For all Ru-M/BaTiO₃ catalysts, the
366 relative intensity of the H atomic line was roughly the same around 0.99. In addition,
367 Ru-Co/BaTiO₃ showed the highest relative intensity of the NH emission line, while
368 Ru-Fe/BaTiO₃ and Ru-Ni/BaTiO₃ exhibit a lower relative intensity of 0.8, indicating a
369 relatively lower concentration of NH radical generated in the plasma discharge region.

370

371 **3.3. Plasma-catalytic NH₃ synthesis over Ru-M/BaTiO₃**

372 **3.3.1 Electrical characteristics of plasma discharge**

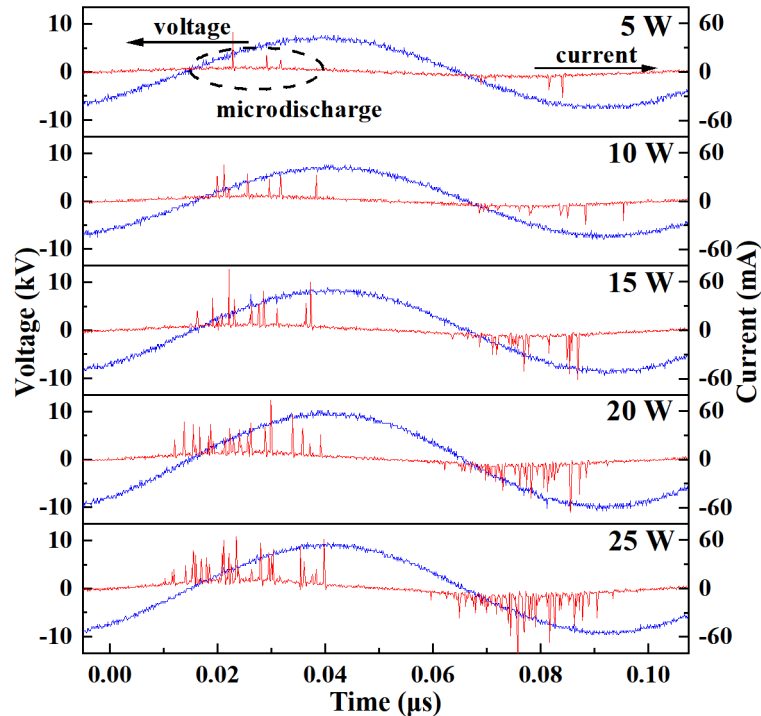


373
374 **Fig. 9.** Lissajous figures of Ru-Ni/BaTiO₃ packed DBD reactor
375

376 Fig. 9 shows that the Lissajous figures have a parallelogram shape, indicating the
377 presence of filamentary discharge in the plasma region [50]. The Lissajous figures
378 were used to calculate several parameters, including total capacitance (C_{cell}), effective
379 capacitance (C_{eff}), transferred charges (Q_{trans}), and breakdown voltage (U_b), using the
380 method described in the supporting information (Section 1). As presented in table S3,
381 increasing the plasma discharge power resulted in a significant increase in C_{eff} , Q_{trans} ,
382 and U_b . For example, when the plasma discharge power increased from 5 W to 20 W,
383 C_{eff} increased from 61.43 pF to 103.75 pF. The increase in C_{eff} suggests that
384 increasing the discharge power promotes the generation of filamentary discharge
385 channels and the spatial expansion of the plasma discharge region [51]. When the
386 dielectric barrier is broken down, gas molecules are ionized and a large number of
387 charged particles are generated, and electrons are directionally moved under the
388 action of the electric field to form filamentary discharge channels [52]. As the plasma
389 discharge power gradually increased from 5 W to 25 W, the U_b increased from 0.27

390 kV to 1.23 kV. The increased U_b resulted in a higher electric field strength in the
 391 plasma discharge region, and the electron escape velocity was accelerated, which
 392 promoted the development of the electron avalanche. As a result, the density of the
 393 filamentary discharge channel increased significantly, and the value of C_{eff} increased
 394 with the increase of discharge power. Considering the higher U_b and C_{eff} , it could be
 395 deduced that the discharge mode changed from partial discharge to uniform
 396 filamentary discharge when plasma discharge power increased from 5 W to 25 W.
 397 Correspondingly, the increased plasma discharge also promoted electron transfer per
 398 half-cycle via plasma discharge, which increased from 0.27 μC (5 W) to 0.97 μC (20
 399 W).

400



401

402 **Fig. 10.** *U-I* profiles of Ru-Ni/BaTiO₃ packed DBD reactor during plasma discharge

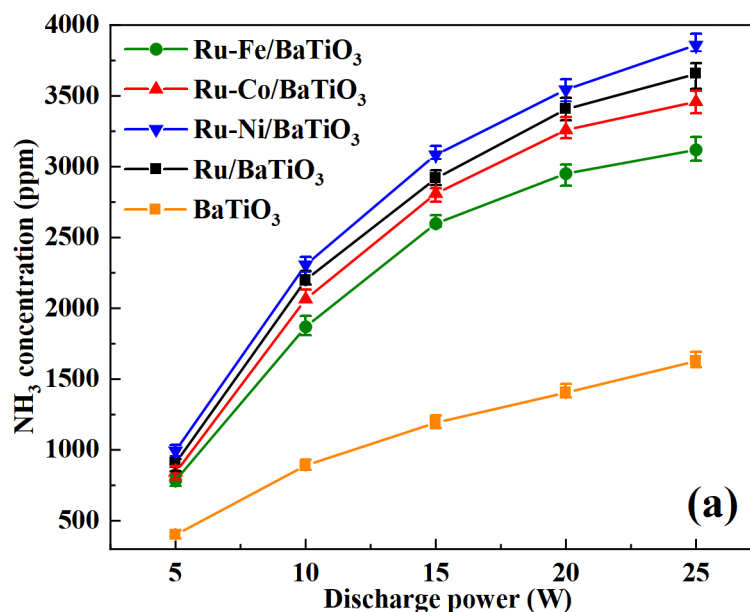
403

404 Fig. 10 presents the electrical signals at different plasma discharge power. Both

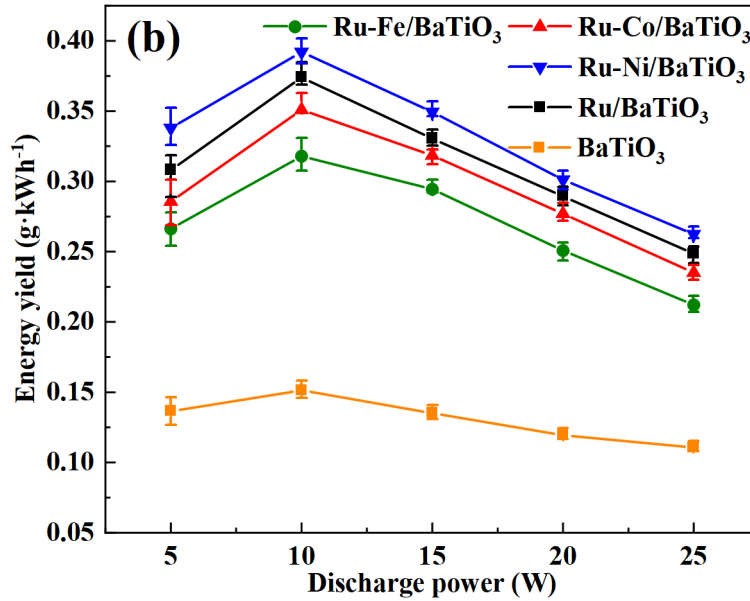
405 voltage and current waveforms exhibit a sinusoidal pattern, but with numerous spikes
406 near the voltage peak, which further confirms the presence of filamentary discharges.
407 The intensity and density of micro-discharges were evaluated by the spikes per
408 discharge cycle [53]. Only a few spikes could be observed at a plasma discharge
409 power of 5 W. Increasing the plasma discharge power enhanced the intensity and
410 density of filamentary discharge, leading to more transferred charges and a higher
411 effective capacitance, as confirmed by the Lissajous figures. This phenomenon
412 confirms that increasing plasma discharge power can promote NH_3 synthesis by
413 expanding the discharge region and increasing the number of filamentary channels.
414 Furthermore, the Lissajous figures overlapped with each other in the presence of
415 different Ru-M/BaTiO₃ catalysts as shown in Fig. S3, indicating that the dopant M has
416 only a limited effect on plasma discharge characteristics.

417

418 3.3.2 Plasma-catalytic NH_3 synthesis performance



419



420

421 **Fig. 11.** Plasma-catalytic NH₃ synthesis performance over Ru-M/BaTiO₃ catalysts (a)
 422 outlet NH₃ concentration; (b) energy yield

423

424 Fig. 11 illustrates the NH₃ synthesis performance of Ru-M/BaTiO₃ (M = Fe, Co
 425 and Ni) catalysts based on the outlet NH₃ concentration at the plasma discharge power
 426 from 5 W to 25 W. It is evident that the NH₃ concentration showed an upward trend
 427 following the increasing plasma discharge power. Among the tested catalysts,
 428 Ru-Ni/BaTiO₃ exhibited the highest NH₃ concentration, followed by Ru/BaTiO₃,
 429 Ru-Co/BaTiO₃, Ru-Fe/BaTiO₃ and BaTiO₃ over the discharge power range. The NH₃
 430 concentration over Ru-Ni/BaTiO₃ increased significantly from 994 ppm at 5 W to
 431 3083 ppm at 15 W, and then gradually to 3895 ppm at 25 W. In contrast, the
 432 introduction of dopant Co and Fe reduced the NH₃ concentration compared to
 433 Ru/BaTiO₃. For all tested catalysts, the energy yield initially increased from 5 W to 10
 434 W and then decreased from 10 W to 25 W, as shown in Fig. 11(b). Specifically, for
 435 Ru-Ni/BaTiO₃, the energy yield increased from 0.34 g kWh⁻¹ at 5 W to 0.39 g kWh⁻¹
 436 at 10 W and then decreased to 0.26 g kWh⁻¹ at 25 W. Similar trends were observed in

437 previous works [54, 55].

438

439 **3.3.3 Enhancement mechanism of Ru-Ni/BaTiO₃**

440 Instead of N₂ directly dissociating in the gas phase through electron-impact, the
441 plasma promotes N₂ dissociation on the catalyst surface by activating N₂ to the
442 vibrational state, which improves the dissociative sticking probability and reduces the
443 dissociation energy barrier of N₂ on the catalyst surface, as reported in previous
444 studies [56, 57]. In this work, we found that the relative concentration of N₂^{*} in the
445 presence of Ru-Ni/BaTiO₃ was lower than that of Ru/BaTiO₃. This is likely due to the
446 high dispersity of the Ru-Ni particles, which facilitates the breakover of plasma
447 discharge current, but is unfavorable for enhancing the electric field [58, 59]. Based
448 on the relatively lower degree of N₂ excitation in the presence of Ru-Ni/BaTiO₃, it can
449 be deduced that the enhancement of NH₃ synthesis performance mainly results from
450 the activation of N₂ on the catalyst surface, followed by hydrogenation through
451 heterogeneous reactions.

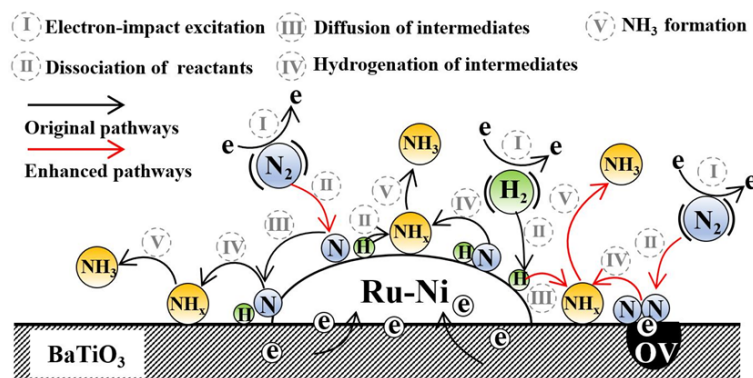
452 The introduction of dopant M caused changes to the lattice structure of BaTiO₃
453 by partially substituting Ti⁴⁺. This led to a distortion of the coordination environment
454 of the B site in BaTiO₃, which facilitated the generation of OV on the support of
455 Ru-M/BaTiO₃. In the present study, the density of OV was significantly increased in
456 the presence of dopant M, with the order being Ru-Ni/BaTiO₃ (15.35%) >
457 Ru-Fe/BaTiO₃ (14.18%) > Ru-Co/BaTiO₃ (7.90%) > Ru/BaTiO₃ (1.73%). The
458 generation of OV on the catalyst surface enhanced the Lewis basicity of the catalysts,

459 which in turn promoted the adjacent active metal into the electron-rich state via the
460 classical SMSI effect [60]. Based on the CO₂-TPD results, the Ru-Ni/BaTiO₃ catalyst
461 had the highest Lewis basic sites of 54.2 μmol g⁻¹, which was 1.70 times higher than
462 that of Ru/BaTiO₃. Correspondingly, the BE of Ru 3p_{1/2} for the Ru-M/BaTiO₃
463 catalysts decreased further compared to the Ru/BaTiO₃ catalyst, indicating the higher
464 density of electron on Ru surface for the case of Ru-Ni/BaTiO₃. Although the
465 generation of OV may not be the only reason for the electron-rich state of Ru sites
466 considering the electron transfer from Ni to Ru, it had a positive effect on improving
467 the electron density of Ru.

468 OV has been demonstrated to act as electron traps that can effectively activate N₂
469 by depleting electrons from the vacancy and accumulating electrons on the adsorbed
470 N₂ [61, 62]. Zhang et al. have also found that higher OV densities on TiO₂ surfaces
471 led to stronger N₂ adsorption and activation capacities, suggesting that chemisorption
472 of N₂ on OV sites [63]. These findings are consistent with the observed enhancement
473 in N₂ chemisorption and increase in OV density on Ru-Ni/BaTiO₃ catalysts.
474 Conventionally, electron transfer occurs in M-N₂ complexes, where transitional metals
475 such as Fe, Mo, Ru, and Co donate their available d-orbital electrons to the π*
476 anti-bonding orbital to activate chemisorbed N₂. Therefore, two types of active sites
477 existed on Ru-Ni/BaTiO₃ and played a role in activating N₂ on the catalyst surface:
478 electron-rich metal sites and OV. The electrons in the metastable state on these active
479 sites can transfer into the anti-bonding orbital of chemisorbed N₂, promoting N₂
480 activation [63, 64]. Consequently, N₂ is converted to NH_x intermediates and

481 ultimately to NH_3 by stepwise hydrogenation [19, 65].

482 During the plasma-catalytic synthesis of NH_3 from N_2 and H_2 , H radicals can
483 also be generated through electron-impact dissociation of H_2 in the gas phase. These
484 H species have a high probability to be adsorbed on the catalyst surface, especially on
485 the active metal sites, resulting in hydrogen poisoning of catalyst [19, 66]. This
486 indicates that abundant H atoms cover the active sites of Ru catalysts, inhibiting the
487 dissociative adsorption of N_2 on the metal surface and thereby limiting NH_3 synthesis
488 [67]. In addition to promoting the activation and dissociation of adsorbed N_2 , OV can
489 also accept H atoms from the Ru surface, leading to the formation of H atoms
490 confined in OV (OV_H). This process helps to suppress hydrogen poisoning and
491 promotes the regeneration of active metal sites [68]. The HRTEM images confirmed
492 that the SMSI effect was successfully induced in Ru/ BaTiO_3 and Ru-Ni/ BaTiO_3 ,
493 which can inhibit hydrogen adsorption on the active metal sites and, consequently,
494 facilitate N_2 dissociation on catalyst surfaces [38]. However, it was observed that the
495 metal particles tend to be fully encapsulated in Ru/ BaTiO_3 , indicating a strong SMSI
496 effect that could decrease the number of active metal sites as larger metal particles are
497 more likely to be encapsulated by the migrated support, which could further limit the
498 accessibility of active sites [69]. By doping with Ni, the overlayer generated by the
499 SMSI effect only occurs at the interfaces between the Ru-Ni alloy and BaTiO_3 support
500 probably due to its smaller size distribution of the metal particles [35]. This way, the
501 SMSI effect can maintain the electron transfer capacity with few active sites being
502 covered.

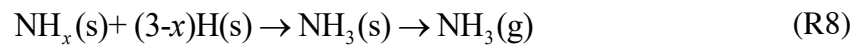
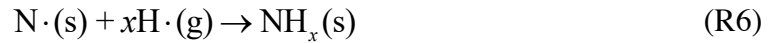
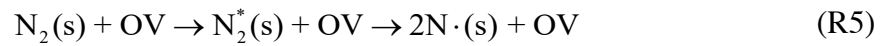
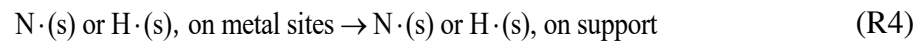
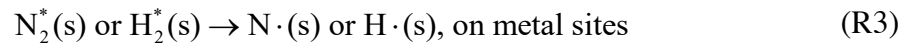
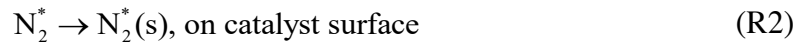
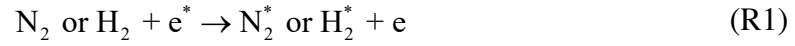


503

504 **Fig. 12.** Proposed mechanisms of plasma-catalytic NH₃ synthesis over Ru-Ni/BaTiO₃
 505 catalyst
 506

507 Based on the preceding discussions, Fig. 12 summarizes the possible
 508 enhancement mechanisms for NH₃ synthesis by plasma-catalysis over Ru-Ni/BaTiO₃.
 509 The superior NH₃ synthesis performance of Ru-Ni/BaTiO₃ over Ru/BaTiO₃ was
 510 mainly attributed to the enhanced surface reactions. The key enhanced reactions are
 511 described below: The NH₃ synthesis process over Ru-Ni/BaTiO₃ commenced with the
 512 electron-impact excitation of reactants (R1). In particular, these reactions
 513 pre-activated N₂ and facilitated its chemisorption on the catalyst surface for further
 514 activation and dissociation (R2). The introduction of Ni promoted Ru-Ni to be in an
 515 electron-rich state, which facilitated the dissociation of reactants on the metal surface
 516 and generated surface-adsorbed N and H species (R3). These N and H species could
 517 diffuse from metal sites to the support (R4). The introduction of Ni also resulted in an
 518 increased density of OV, which was a newly-generated active site for N₂ activation
 519 (R5). Consequently, Ru-Ni/BaTiO₃ significantly promoted N₂ surface-dissociation
 520 reactions, leading to an improvement in the concentration of N reactive species on the
 521 catalyst surface. These adsorbed N reactive species could react with abundant H
 522 radicals in the gas-phase (via E-R mechanism, R6) and on the surface (via L-H

523 mechanism, R7). As a result, more NH_x ($x = 1$ or 2) species could be generated
 524 through the stepwise hydrogenation of N radicals, and NH_3 generation was
 525 accelerated finally (R8).



526

527 **4. Conclusion**

528 Plasma-catalytic NH_3 synthesis was carried out using Ru-based bimetallic
 529 catalysts (Ru-M/BaTiO₃, M = Fe, Co and Ni). The NH_3 synthesis performance was
 530 observed to follow the order of Ru-Ni/BaTiO₃ > Ru/BaTiO₃ > Ru-Co/BaTiO₃ >
 531 Ru-Fe/BaTiO₃, with the highest NH_3 concentration (3895 ppm) and energy yield (0.39
 532 g kWh⁻¹) obtained over Ru-Ni/BaTiO₃ at 25 W and 10 W, respectively.

533 The effect of dopant M in the Ru-M/BaTiO₃ catalysts and its effect on
 534 plasma-catalytic NH_3 synthesis were investigated. The basicity of the catalyst was
 535 found to increase after being loaded with M, while the loading of Ni strengthened the
 536 N_2 adsorption capacity of the catalyst. Compared to Ru/BaTiO₃, the introduction of

537 dopant Ni increases the density of Lewis basic sites and amount of chemisorbed N₂ on
538 the surface of Ru-Ni/BaTiO₃, which were 1.70 times and 1.09 times higher,
539 respectively. The Ru-Ni/BaTiO₃ catalyst also had a smaller metal particle size, which
540 prevented the encapsulation of Ru-Ni particles by the BaTiO₃ support, improving the
541 accessibility of active metal sites for NH₃ synthesis compared to Ru/BaTiO₃. On the
542 one hand, these properties increased the amount of chemisorbed N₂ on the catalyst
543 surface. On the other hand, they increased the electron density on the surface of Ru.
544 These facilitate the activation of N₂ on the metal sites via the electron-donating effect,
545 as well as on the catalyst surface. As a result, N₂ could be stepwise hydrogenated to
546 form NH_x species and ultimately NH₃. This study demonstrated that the incorporation
547 of dopant Ni could be an effective means of further improving catalyst activity in the
548 plasma-catalytic NH₃ synthesis process, as it could act as both an electronic and
549 structural promoter in Ru-based bimetallic catalysts.

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553

554 **References**

- 555 1. R.H. Dolan, J.E. Anderson, T.J. Wallington, Outlook for ammonia as a sustainable
556 transportation fuel. *Sustainable Energy Fuels*, 5(2021): 4830-4841.
- 557 2. D. Erdemir, Dincer I., A perspective on the use of ammonia as a clean fuel: Challenges
558 and solutions. *Int. J. Energy Res.*, 4 (2021): 4827-4834.
- 559 3. D. Zhou, R. Zhou, R. Zhou, B. Liu, T. Zhang, Y. Xian, P.J. Cullen, X. Lu, K. Ostrikov,
560 Sustainable ammonia production by non-thermal plasmas: Status, mechanisms, and
561 opportunities. *Chem. Eng. J.*, 421(2021): 129544.
- 562 4. A. Yousefi, H. Guo, S. Dev, B. Liko, S. Lafrance, Effects of ammonia energy fraction and
563 diesel injection timing on combustion and emissions of an ammonia/diesel dual-fuel
564 engine. *Fuel*, 314(2022): 122723.

- 565 5. Lasocki, J., M. Bednarski, M. Sikora, Simulation of ammonia combustion in dual-fuel
566 compression-ignition engine. *IOP Conf. Ser. Earth Environ. Sci.*, 214(2019): 012081.
- 567 6. A.G. Olabi, M.A. Abdelkareem, M.A. Murisi, N. Shehata, A.H. Alami, A. Radwan, T.
568 Wilberforce, K. Chae, E.T. Sayed, Recent progress in Green Ammonia: Production,
569 applications, assessment; barriers, and its role in achieving the sustainable development
570 goals. *Energy Convers. Manage.*, 277(2023): 116594.
- 571 7. Z. Wan, Y. Tao, J. Shao, Y. Zhang, H. You, Ammonia as an effective hydrogen carrier and
572 a clean fuel for solid oxide fuel cells. *Energy Convers. Manage.*, 228(2021): 113729.
- 573 8. F. Gorky, J.M. Lucero, J.M. Crawford, B.A. Blake, S.R. Guthrie, M.A. Carreon, M.L.
574 Carreon, Insights on cold plasma ammonia synthesis and decomposition using alkaline
575 earth metal-based perovskites. *Catal. Sci. Technol.*, 11(2021): 5109-5118.
- 576 9. Y. Huang, N. Zhang, Z. Wu, X. Xie, Artificial nitrogen fixation over bismuth-based
577 photocatalysts: fundamentals and future perspectives. *J. Mater. Chem. A*, 8(2020):
578 4978-4995.
- 579 10. M.H. Vu, M. Sakar, T.O. Do, Insights into the recent progress and advanced materials for
580 photocatalytic nitrogen fixation for ammonia (NH₃) production. *Catalysts*, 8(2018): 621.
- 581 11. A. Wu, J. Yang, B. Xu, X. Wu, Y. Wang, X. Lv, Y. Ma, A. Xu, J. Zheng, Q. Tan, Y. Peng,
582 Z. Qi, H. Qi, J. Li, Y. Wang, J. Harding, X. Tu, A. Wang, J. Yan, X. Li, Direct ammonia
583 synthesis from the air via gliding arc plasma integrated with single atom electrocatalysis.
584 *Appl. Catal., B*, 299(2021): 120667.
- 585 12. A. Chen, B. Xia, Ambient dinitrogen electrocatalytic reduction for ammonia synthesis. *J.*
586 *Mater. Chem. A*, 7(2019): 23416-23431.
- 587 13. L.R. Winter, B. Ashford, J. Hong, A.B. Murphy, J.G. Chen, Identifying surface reaction
588 intermediates in plasma catalytic ammonia synthesis. *ACS Catal.*, 10(2020):
589 14763-14774.
- 590 14. A. Vojvodic, A.J. Medford, F. Studt, F.A. Pedersen, T.S.Khan, T. Bligaard, J.K. Ngrskov,
591 Exploring the limits: A low-pressure, low-temperature Haber-Bosch process. *Chem. Phys.*
592 *Lett.*, 598(2014): 108-112.
- 593 15. H. Zhu, P. Zhang, S. Dai, Recent Advances of lanthanum-based perovskite oxides for
594 catalysis. *ACS Catal.*, 5(2015): 6370-6385.
- 595 16. H. Fang, D. Liu, Y. Luo, Y. Zhou, S. Liang, X. Wang, B. Lin, L. Jiang, Challenges and
596 opportunities of Ru-based catalysts toward the synthesis and utilization of ammonia. *ACS*
597 *Catal.*, 12(2022): 3938-3954.
- 598 17. V.S. Marakatti, E.M. Gaigneaux, Recent advances in heterogeneous catalysis for
599 ammonia synthesis. *Chemcatchem*, 12(2020): 5838-5857.
- 600 18. J. Liu, X. Zhu, X. Hu, X. Tu, Plasma-assisted ammonia synthesis in a packed-bed
601 dielectric barrier discharge reactor: roles of dielectric constant and thermal conductivity
602 of packing materials. *Plasma Sci. Technol.*, 24(2022): 025503.
- 603 19. Y. Wang, M. Craven, X. Yu, J. Ding, P. Bryant, J. Huang, X. Tu, Plasma-enhanced
604 catalytic synthesis of ammonia over a Ni/Al₂O₃ catalyst at near-room temperature:
605 Insights into the importance of the catalyst surface on the reaction mechanism. *ACS*
606 *Catal.*, 9(2019): 10780-10793.
- 607 20. L. Dou, Y. Liu, Y. Gao, J. Li, X. Hu, S. Zhang, K. Ostrikov, T. Shao, Disentangling
608 metallic cobalt sites and oxygen vacancy effects in synergistic plasma-catalytic CO₂/CH₄

- 609 conversion into oxygenates. *Appl. Catal. B*, 318(2022): 121830.
- 610 21. X. Hu, X. Zhu, X. Wu, Y. Cai, X. Tu, Plasma-enhanced NH₃ synthesis over activated
611 carbon-based catalysts: Effect of active metal phase. *Plasma Processes Polym.*, 17(2020):
612 2000072.
- 613 22. Y. Gong, H. Li, C. Li, X. Bao, H. Hosono, J. Wang, Insight into rare-earth-incorporated
614 catalysts: The chance for a more efficient ammonia synthesis. *J. Adv. Ceram.*,
615 11(2022): 1499-1529.
- 616 23. T. Wang, F.A. Pedersen, Achieving industrial ammonia synthesis rates at near-ambient
617 conditions through modified scaling relations on a confined dual site. *Proc. Natl. Acad.*
618 *Sci. U. S. A.*, 118(2021): e2106527118.
- 619 24. K.K. Ghuman, K. Tozaki., M. Sadakiyo, S. Kitano, T. Oyabe, M. Yamauchi, Tailoring
620 widely used ammonia synthesis catalysts for H and N poisoning resistance. *Phys. Chem.*
621 *Chem. Phys.*, 21(2019): 5117-5122.
- 622 25. Z. Yang, W. Guo, J. Lin, D. Liao, Supported catalysts with Ru-M (M = Fe, Co, Ni, Mo)
623 bimetallic active centers for ammonia synthesis. *Chin. J. Catal.*, 27(2006): 378-380.
- 624 26. M.J. Banisalman, M.C. Kim, S.S. Han, Origin of enhanced ammonia synthesis on Ru-Co
625 catalysts unraveled by density functional theory. *ACS Catal.*, 12(2022): 1090-1097.
- 626 27. L. Ju, T. Sabergharesou, K.G. Stamplecoskie, M. Hegde, T. Wang, N.A. Combe, H. Wu,
627 P.V. Radovanovic, Interplay between size, composition, and phase transition of
628 canocrystalline Cr³⁺-doped BaTiO₃ as a path to multiferroism in perovskite-type oxides. *J.*
629 *Am. Chem. Soc.*, 134(2012): 1136-1146.
- 630 28. M. Arshad, W. Khan, M. Abushad, M. Nadeem, S. Husain, A. Ansari, V.K. Chakradhary,
631 Correlation between structure, dielectric and multiferroic properties of lead free Ni
632 modified BaTiO₃ solid solution. *Ceram. Int.*, 46(2020): 27336-27351.
- 633 29. C. Calle, J.A. Alonso, L. Martínez, M. Retuerto, H. García, M.T. Fernández, Ru-Ru
634 metal-metal bonding in the chains of edge-sharing octahedra of NdMn_{1.5}Ru_{0.5}O₅: A
635 neutron powder diffraction and magnetic study. *Eur. J. Inorg. Chem.*, 5(2010): 781-789.
- 636 30. Y. Hamasaki, T. Shimizu, S. Yasui, T. Taniyama, M. Itoh, Evidence of ferroelectricity in
637 ferrimagnetic kappa-Al₂O₃-type In_{0.25}Fe_{1.75}O₃ films. *Appl. Phys. Lett.*, 16(2016): 162901.
- 638 31. W. Peng, L. Li, S. Yu, P. Yang, K. Xu, Dielectric properties, microstructure and charge
639 compensation of MnO₂-doped BaTiO₃-based ceramics in a reducing atmosphere. *Ceram.*
640 *Int.*, 47(2021): 29191-29196.
- 641 32. J. Wang, J. Leroy, G. Niu, G.S. Giron, B. Gautier, B. Vilquin, N. Barrett, Chemistry and
642 structure of BaTiO₃ ultra-thin films grown by different O₂ plasma power. *Chem. Phys.*
643 *Lett.*, 592(2014): 206-210.
- 644 33. K. Xu, G. Zhu, H. Xu, Y. Zhao, K. Jiang, X. Zhang, H. Yin, M. Shangguan, L. Wan, T.
645 Huang, The colossal permittivity effect on BaTiO₃ induced by different sinter atmosphere.
646 *Appl. Phys. A: Mater. Sci. Process.*, 128(2022): 1044.
- 647 34. J. Ni, R. Wang, F. Kong, T. Zhang, J. Lin, B. Lin, K. Wei, Highly efficient Ru-Ba/AC
648 catalyst promoted by magnesium for ammonia synthesis. *Chin. J. Catal.*, 32(2011):
649 436-439.
- 650 35. Y. Guo, S. Mei, K. Yuan, D. Wang, H. Liu, C. Yan, Y. Zhang, Low-temperature CO₂
651 methanation over CeO₂-supported Ru single atoms, nanoclusters, and nanoparticles
652 competitively tuned by strong metal-support interactions and H-spillover effect. *ACS*

- Catal., 8(2018): 6203-6215.
- 654 36. M.W. Arshad, Y. You, Y. Kim, L. Heo, S. Kim, Effect of the support material of IrRu
655 catalysts on CO-assisted NO_x reduction. Chem. Eng. J., 455(2023): 140911.
- 656 37. H. Ronduda, M. Zybert, W. Patkowski, A. Tarka, P. Jodtowski, L. Kepinski, A. Sarnecki,
657 D. Moszynski, W. R. Pilecka, Tuning the catalytic performance of Co/Mg-La system for
658 ammonia synthesis via the active phase precursor introduction method. Appl. Catal. A,
659 598(2020): 117553.
- 660 38. Z. Wang, B. Liu, J. Lin, Highly effective perovskite-type BaZrO₃ supported Ru catalyst
661 for ammonia synthesis. Appl. Catal. A, 458(2013): 130-136.
- 662 39. J. Hao, F. Cai, J. Wang, Y. Fu, J. Zhang, Y. Sun, The effect of oxygen vacancy of
663 alkaline-earth metal Sr doped Sm₂Zr₂O₇ catalysts in the oxidative coupling of methane.
664 Chem. Phys. Lett., 771(2021): 138562.
- 665 40. D. Yu, J. He, T. Xie, J. Yang, J. Wang, J. Xie, H. Shi, Z. Gao, B. Xiang, D.D. Dionysiou,
666 boosting catalytic activity of SrCoO_{2.52} perovskite by Mn atom implantation for advanced
667 peroxymonosulfate activation. J. Hazard. Mater., 442(2023): 130085.
- 668 41. H. Ma, Z. Shi, Q. Li, S. Li, Preparation of graphitic carbon nitride with large specific
669 surface area and outstanding N₂ photofixation ability via a dissolve-regrowth process. J.
670 Phys. Chem. Solids, 99(2016): 51-58.
- 671 42. A.R. Singh, J.H. Montoya, B.A. Rohr, C. Tsai, A. Vovodic, J.K. Narskov, Computational
672 design of active site structures with improved transition-state scaling for ammonia
673 synthesis. ACS Catal., 8(2018): 4017-4024.
- 674 43. Y. Zhan, C. Zhou, F. Jin, C. Chen, L. Jiang, Ru/TiO₂ catalyst for selective hydrogenation
675 of benzene: Effect of surface hydroxyl groups and spillover hydrogen. Appl. Surf. Sci.,
676 525(2020): 146627.
- 677 44. Z. Han, C. Choi, S. Hong, T. W, Y. Soo, Y. Lung, J. Qiu, Z. Sun, Activated TiO₂ with
678 tuned vacancy for efficient electrochemical nitrogen reduction. Appl. Catal. B, 257(2019):
679 117896.
- 680 45. H. Chen, S. Tosoni, G. Pacchioni, Adsorption of ruthenium atoms and clusters on anatase
681 TiO₂ and tetragonal ZrO₂(101) surfaces: A comparative DFT study. J. Phys. Chem. C,
682 119(2015): 10856-10868.
- 683 46. J.L. Clabel, I.T. Awan, G. Lozano, M.A. Pereira-da-Silva, R.A. Romano, V.A.G. Rivera,
684 S.O. Ferreira, E. Marega, Understanding the electronic properties of BaTiO₃ and Er³⁺
685 doped BaTiO₃ films through confocal scanning microscopy and XPS: the role of oxygen
686 vacancies. Phys. Chem. Chem. Phys., 22(2020): 15022-15034.
- 687 47. J. Zhu, L. Song, J. Jiang, Y. Xiong, N. Zhang, X. Li, H. Ye, S. Chen, H. Ju, D. Liu, Y. Lin,
688 W. Ye, C. Wang, Q. Xu, Oxide defect engineering enables to couple solar energy into
689 oxygen activation. J. Am. Chem. Soc., 138(2016): 8928-8935.
- 690 48. H. Han, S. Jin, S. Park, Y. Kim, D. Jang, M.H. Seo, W.B. Kim, Plasma-induced oxygen
691 vacancies in amorphous MnO_x boost catalytic performance for electrochemical CO₂
692 reduction. Nano Energy, 79(2021): 105492.
- 693 49. H. Zhou, M. Wang, F. Wang, Oxygen-vacancy-mediated catalytic methanation of
694 lignocellulose at temperatures below 200 degrees C. Joule, 5(2021): 3031-3044.
- 695 50. F.A. Herrera, G.H. Brown, P. Barboun, N. Turan, P. Mehta, W.F. Schneider, J.C. Hicks,
696 D.B. Go, The impact of transition metal catalysts on macroscopic dielectric barrier

- 697 discharge (DBD) characteristics in an ammonia synthesis plasma catalysis reactor. *J. Phys.*
698 *D: Appl. Physics*, 52(2019): 224002.
- 699 51. X. Tu, H.J. Gallon, M.V. Twigg, P.A. Gorry, J.C. Whitehead, Dry reforming of methane
700 over a Ni/Al₂O₃ catalyst in a coaxial dielectric barrier discharge reactor. *J. Phys. D: Appl.*
701 *Phys.*, 44(2011): 274007.
- 702 52. D.P. Subedi, U.M. Joshi, C.S. Wong, Dielectric barrier discharge (DBD) plasmas and
703 their applications, *Plasma Sci. Technol. for Emerging Econ* (2017). Springer, Singapore.
- 704 53. B.S. Patil, A.S.R. Kaathoven, F.J.J. Peeters, N. Cherkasov, J. Lang, Q. Wang, V. Hessel,
705 Deciphering the synergy between plasma and catalyst support for ammonia synthesis in a
706 packed dielectric barrier discharge reactor. *J. Phys. D: Appl. Phys.*, 53(2020): 144003.
- 707 54. P. Peng, Y. Li, Y. Cheng, S. Deng, P. Chen, R. Ruan, Atmospheric pressure ammonia
708 synthesis using non-thermal plasma assisted catalysis. *Plasma Chem. Plasma Process.*
709 36(2016): 1201-1210.
- 710 55. X. Zhu, J. Liu, X. Hu, Z. Zhou, X. Li, W. Wang, R. WU, X. Tu. Plasma-catalytic
711 synthesis of ammonia over Ru-based catalysts: Insights into the support effect. *J. Energy*
712 *Inst.* 102(2022): 240-246.
- 713 56. K.H.R. Rouwenhorst, H.G.B. Burbach, D.W. Vogel, J.N. Paulí, B. Geerdink, L. Lefferts,
714 Plasma-catalytic ammonia synthesis beyond thermal equilibrium on Ru-based catalysts in
715 non-thermal plasma. *Catal. Sci. Technol.*, 11(2021): 2834-2843.
- 716 57. K.H.R. Rouwenhorst, L. Lefferts, On the mechanism for the plasma-activated N₂
717 dissociation on Ru surfaces. *J. Phys. D: Appl. Phys.*, 54(2021): 393002.
- 718 58. N. Naude, F. Massines, Influence of the surface conductivity on the stability of a glow
719 dielectric-barrier discharge. *IEEE Trans. Plasma Sci.*, 36(2008): 1322-1323.
- 720 59. R. Pribyl, P. Stastny, M. Pazderka, J. Kellar, Z.K. Tucekova, M. Zemanek, M. Trunec, M.
721 Cernak, Properties of MgAl₂O₄ doped alumina barrier layers for dielectric barrier
722 discharge. *J. Phys. D: Appl. Phys.*, 53(2020): 505202.
- 723 60. X. Liu, M. Liu, Y. Luo, C. Mou, S.D. Lin, H. Cheng, J. Chen, J. Lee, T. Lin, Strong
724 metal-support interactions between gold nanoparticles and ZnO nanorods in CO oxidation.
725 *J. Am. Chem. Soc.*, 134(2012): 10251-10258.
- 726 61. H. Li, J. Shang, Z. Ai, L. Zhang, Efficient visible light nitrogen fixation with BiOBr
727 nanosheets of oxygen vacancies on the exposed {001} facets. *J. Am. Chem. Soc.*,
728 137(2015): 6393-6399.
- 729 62. S. Wang, X. Hai, X. Ding, K. Chang, Y. Xiang, X. Meng, Z. Yang, H. Chen, J. Ye,
730 Light-switchable oxygen vacancies in ultrafine Bi₅O₇Br nanotubes for boosting
731 solar-driven nitrogen fixation in pure water. *Adv. Mater.*, 29(2017): 1701774.
- 732 63. G. Zhang, X. Yang, C. He, P. Zhang, H. Mi, Constructing a tunable defect structure in
733 TiO₂ for photocatalytic nitrogen fixation. *J. Mater. Chem. A*, 8(2020): 334-341.
- 734 64. X. Li, S. Ma, F. Luo, Z. Yang, Engineering oxygen vacancies in Co₃O₄ to boost p-d
735 hybridization for hydrogeneration of nitrogen and oxygen. *Appl. Surf. Sci.*, 600(2022):
736 154115.
- 737 65. P. Peng, Y. Li, Y. Cheng, S. Deng, P. Chen, R. Ruan, Atmospheric pressure ammonia
738 synthesis using non-thermal plasma assisted catalysis. *Plasma Chem. Plasma Process.*,
739 36(2016): 1201-1210.
- 740 66. J. Sun, Q. Chen, X. Zhao, H. Lin, W. Qin, Kinetic investigation of plasma catalytic

- 741 synthesis of ammonia: insights into the role of excited states and plasma-enhanced
742 surface chemistry. *Plasma Sources Sci. Technol.*, 31(2022): 094009.
- 743 67. J. Huang, M. Yuan, X. Li, Y. Wang, M. Li, J. Li, Z. You, Inhibited hydrogen poisoning for
744 enhanced activity of promoters-Ru/Sr₂Ta₂O₇ nanowires for ammonia synthesis. *J. Catal.*
745 389(2020): 556-565.
- 746 68. J. Huang, J. Pan, Z. You, X. Jiang, Enhanced ammonia synthesis activity of Ru/Ba₅Ta₄O₁₅
747 catalyst by reduction of Ba₅Ta₄O₁₅ with CaH₂. *Int. J. Hydro. Energy*, 47(2022):
748 28019-28024.
- 749 69. Z. Luo, G. Zhao, H. Pan, W. Sun, Strong metal-support interaction in heterogeneous
750 catalysts. *Adv. Energy Mater.*, 12(2022): 2201395.
- 751